



Pontifícia Universidade Católica do Rio Grande do Sul

FACULDADE DE ENGENHARIA

PROGRAMA DE PÓS-GRADUAÇÃO EM ENGENHARIA E TECNOLOGIA DE MATERIAIS

# **DESIGN DE NOVOS MATERIAIS POLIMÉRICOS PARA CAPTURA DE CO<sub>2</sub>**

**FRANCIELE LONGARAY BERNARD**

QUÍMICA INDUSTRIAL E LICENCIADA

MESTRE EM ENGENHARIA E TECNOLOGIA DE MATERIAIS

**TESE PARA A OBTENÇÃO DO TÍTULO DE DOUTOR EM ENGENHARIA E  
TECNOLOGIA DE MATERIAIS**

**Porto Alegre**

**Junho, 2017**



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ORIENTADOR: PROF(a). DR(a). SANDRA MARA DE OLIVEIRA EINLOFT

CO-ORIENTADOR: Prof. Dr. FELIPE DALLA VECCHIA

Tese realizada no Programa de Pós-Graduação em Engenharia e Tecnologia de Materiais (PGETEMA) da Pontifícia Universidade Católica do Rio Grande do Sul, como parte dos requisitos para a obtenção do título de Doutor em Engenharia e Tecnologia de Materiais.

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**CANDIDATA: FRANCIELE LONGARAY BERNARD**

Esta Tese de Doutorado foi julgada para obtenção do título de DOUTOR EM ENGENHARIA E TECNOLOGIA DE MATERIAIS e aprovada em sua forma final pelo Programa de Pós-Graduação em Engenharia e Tecnologia de Materiais da Pontifícia Universidade Católica do Rio Grande do Sul.

*Sandra Oliveira Einloft*

\_\_\_\_\_  
**DRA. SANDRA MARA OLIVEIRA EINLOFT - ORIENTADORA**

*Felipe Dalla Vecchia*  
\_\_\_\_\_  
**DR. FELIPE DALLA VECCHIA - CO-ORIENTADOR**

**BANCA EXAMINADORA**

*Carlos Arthur Ferreira*

\_\_\_\_\_  
**DR. CARLOS ARTHUR FERREIRA - DO PPGE3M - UFRGS**

*Ofélia de Queiroz Fernandes Araújo*

\_\_\_\_\_  
**DRA. OFÉLIA DE QUEIROZ FERNANDES ARAÚJO - ESCOLA DE QUÍMICA DA  
UNIVERSIDADE FEDERAL DO RIO DE JANEIRO, DEPARTAMENTO DE ENGENHARIA  
QUÍMICA - UFRJ**

*Rubem Mário F. Vargas*

\_\_\_\_\_  
**DR. RUBEM MÁRIO FIGUEIRÓ VARGAS - DO PGETEMA/FENG - PUCRS**

**PUCRS**

Campus Central  
Av. Ipiranga, 6681 - Prédio 30 - Sala 103 - CEP: 90619-900  
Telefone: (51) 3353.4059 - Fax: (51) 3320.3625  
E-mail: engenharia.pg.materiais@puccrs.br

*"Cada sonho que você deixa pra trás, é um pedaço do seu futuro que deixa de existir"*

(Steve Jobs)

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## LISTA DE SÍMBOLOS

[bmpyrr][Cl]	Cloreto de 1-butil-1-metilpirrolidínio
[Bmim] <sup>+</sup>	Cátion 1 butil - 3 – metilimidazólio
[bmim] [PF <sub>6</sub> ]	Hexafluorofosfato de 1-butil-3-metilimidazólio
[BMIM] [TF <sub>2</sub> N]	Bis(trifluorometilsulfonil)imida de 1-butil-3-metilimidazólio
[bmim] [BF <sub>4</sub> ]	1-butil-3-metilimidazolio tetrafluoroborato
[BF <sub>4</sub> ] <sup>-</sup>	Ânion Tetrafluoroborato
DEA	Dietanolamina
GEE	Gases Efeito Estufa
LIIs	Líquidos iônicos
MEA	Monoetanolamina
MDEA	Metil dietanolamina
Mn	Massa molar ponderal média
OTf	Ânion Triflato (CF <sub>3</sub> SO <sub>3</sub> <sup>-</sup> )
OTs	Ânion Tosilato (H <sub>3</sub> CC <sub>6</sub> H <sub>4</sub> SO <sub>3</sub> <sup>-</sup> )
PLIs	Poli (líquidos iônicos)
P[VBTMA][BF <sub>4</sub> ]	Poli [p-(vinilbenzil)-trimetilamônio tetrafluoroborato]
P[MATMA][BF <sub>4</sub> ]	Poli [2- (metacrilóiloxi)etil-trimetilamônio tetrafluoroborato]
P[VBTMA][BF <sub>4</sub> ]	Poli [p-(vinilbenzil)-trimetilamônio tetrafluoroborato]
P[VBBI][BF <sub>4</sub> ]	Poli [p-(vinilbenzil)-3-butilimidazólio tetrafluoroborato]
P[VBTEA][PF <sub>6</sub> ]	Poli [4-(vinilbenzil)-triethylamônio hexafluorofosfato]
P[MABI][BF <sub>4</sub> ]	Poli [2- (metacrilóiloxi)etil-3-butilimidazólio tetrafluoroborato]
(PVBIT)	Poli[1-p-(vinilbenzil)-3-butilimidazolio tetrafluoroborato]
(PVBIIH)	Poli[1-p-(vinilbenzil)-3-butilimidazolio Hexafluorofosfato]
(PBIMT)	Poli [2-(metacrilóiloxi)etil-3-butilimidazolio tetrafluoroborato]
[TF <sub>2</sub> N] <sup>-</sup>	Ânion bis(trifluorometilsulfonil)imida
TBAB	Brometo de tetra-n-butilamônio
TBPB	Brometo de tetrabutílfosfônio

## RESUMO

BERNARD, FRANCIELE. L. **Design de Novos Materiais Poliméricos Para Captura de CO<sub>2</sub>**. Porto Alegre. 2017. Tese. Programa de Pós-Graduação em Engenharia e Tecnologia de Materiais, PONTIFÍCIA UNIVERSIDADE CATÓLICA DO RIO GRANDE DO SUL.

O desenvolvimento de materiais de baixo custo e com design inteligente para a separação de CO<sub>2</sub> de gases exaustos é um dos principais focos de pesquisa na atualidade. A absorção química utilizando soluções aquosas de aminas como solventes tem sido extensivamente estudada e utilizada na indústria há décadas. No entanto, a degradação do solvente, a corrosão de equipamentos e o elevado consumo energético necessário para etapa de regeneração são os principais problemas relacionados ao uso destes solventes. Poli (líquidos iônicos) (PLIs) e aminas suportadas têm sido apontados como materiais promissores para captura de CO<sub>2</sub>. PLIs combinam as características úteis dos líquidos iônicos (LIs) tais como a seletividade ao CO<sub>2</sub>, propriedades ajustáveis, estabilidade química e térmica com as propriedades dos polímeros. A adsorção usando aminas suportadas em sólidos oferece vantagens, como por exemplo: a facilidade de manuseio, estabilidade térmica, eliminação do processo de corrosão dos equipamentos, baixo consumo de energia na etapa de regeneração, baixo custo operacional e cinética rápida. Neste trabalho, foram sintetizados PLIs baseados em polímeros sintéticos (poliuretano) e naturais (celulose extraída da casca de arroz). A fim de comparar aminas e polieeteraminas, estas também foram suportadas em celulose extraída da casca de arroz. Os materiais obtidos foram caracterizados por FTIR, RMN, DRX, TGA, DSC, GPC e AFM e MEV-FEG, PTGA ou célula de decaimento de pressão. Estudos com simulações computacionais também foram realizados com o propósito de investigar a interação do CO<sub>2</sub> e os grupos funcionais presentes nos materiais sintetizados. Dentre todos os materiais sintetizados neste trabalho, os PLIs celulósicos CL-TBA e [CeEt<sub>3</sub>N][PF<sub>6</sub>] e com o PLI base PU-imida HPIL-02-TBA demonstraram maior afinidade pelo CO<sub>2</sub>, aliado a boa estabilidade térmica e reciclabilidade.

Palavras-Chaves: Poli (líquidos iônicos), dióxido de carbono, aminas suportadas, celulose.

## ABSTRACT

BERNARD, FRANCIELE. L **Design of New Polymeric Materials for CO<sub>2</sub> Capture** Porto Alegre. PhD Thesis. Graduation Program in Materials Engineering and Technology, PONTIFICAL CATHOLIC UNIVERSITY OF RIO GRANDE DO SUL.

The development of low-cost materials with intelligent design to be used in CO<sub>2</sub> separation from flue gases is one of the main research focuses in this field. Absorption using amines solutions as solvents have been extensively studied and used in industry for decades. However, solvent degradation, equipment corrosion, and the high regeneration energy penalty are the main problems in this process. Poly(ionic liquids) (PILs) or supported amines has been pointed as promising materials for CO<sub>2</sub> capture. PILs combine the useful features of ionic liquid (IL), such as affinity for CO<sub>2</sub>, thermal and chemical stability and adjustable properties, allied to the intrinsic polymer properties. Adsorption using amine supported in solid sorbents offers advantages such as easy handling, thermal stability, elimination of corrosion processes, low regeneration energy, lower operating costs and fast kinetics. In this work, synthetic based (polyurethane) as well natural based (cellulose from rice husk) poly(ionic liquids) were synthesized. Aiming to compare amines and polyetheramines, both were also supported in cellulose extracted from rice husk. The obtained materials were characterized by FTIR, NMR, XDR, TGA, DSC, GPC, AFM, MEV-FEG and PTGA or decay pressure cell in order to evaluate their structure, thermal stability, morphology and sorption capacity. Computational simulation was also performed in order to investigate the interaction of CO<sub>2</sub> with functional groups present in the synthesized materials. Among all synthesized materials cellulosic PILs CL-TBA e [CeIEt<sub>3</sub>N][PF<sub>6</sub>] and PU-imide based PIL HPIL-02-TBA presented higher CO<sub>2</sub> affinity allied to good thermal stability and reusability.

Key-words: Poly(ionic liquids), carbon dioxide, supported amines, cellulose.

## 1. INTRODUÇÃO

O nível crescente da concentração de dióxido de carbono ( $\text{CO}_2$ ) na atmosfera, decorrente da produção e utilização de combustíveis fósseis, tem gerado grande preocupação da sociedade devido à sua implicação no aquecimento global (MONTZKA; DLUGOKENCKY; BUTLER, 2011; SONGOLZADEH et al., 2014; BRUHN; NAIMS; OLFE-KRÄUTLEIN, 2016; YAUMI; ABU BAKAR; HAMEED, 2017). Diferentes estratégias têm sido consideradas e adotadas por diversos países com o objetivo de redução da emissão do  $\text{CO}_2$ . Dentre elas destacam-se a melhoria da eficiência energética, o aumento do uso de combustíveis de baixa emissão de carbono, energias renováveis ou energia nuclear, aplicação de abordagens da geoengenharia e tecnologias de captura de carbono (GARCES et al., 2013; LEUNG; CARAMANNA; MAROTO-VALER, 2014; TOME; MARRUCHO, 2016).

Dentre estas diferentes estratégias a captura de carbono é considerada crucial, pois reduz a emissão de dióxido de carbono de grandes fontes de emissão (TOME; MARRUCHO, 2016). As tecnologias comerciais mais empregadas para captura são a pós-combustão, pré-combustão e oxi-combustão (MONDAL; BALSORA; VARSHNEY, 2012; SONGOLZADEH et al., 2014; KETZER; IGLESIAS; EINLOFT, 2015). Além do ponto de vista ambiental, a captura de carbono pode fornecer  $\text{CO}_2$  para diversas aplicações industriais, tais como a recuperação avançada de petróleo, produção de uréia, fertilizantes, gelo seco ou até mesmo o uso como solvente no estado supercrítico (SONGOLZADEH et al., 2014; BRUHN; NAIMS; OLFE-KRÄUTLEIN, 2016).

Absorção química com o uso de soluções aquosas de aminas é o processo mais empregado comercialmente em plantas de captura de  $\text{CO}_2$  (BACHELOR; TOOCHINDA, 2012; ARIAS et al., 2016). Embora, este método seja considerado economicamente viável para separação de  $\text{CO}_2$  de diversas correntes de misturas



gasosas (RIVERA-TINOCO; BOUALLOU, 2010; BACHELOR; TOOCHINDA, 2012; RAMDIN et al., 2014), a utilização de soluções aquosas de aminas como solvente oferece uma série de inconvenientes operacionais, sendo as mais preocupantes a corrosão de equipamentos, volatilidade, degradação e alto consumo de energia (BACHELOR; TOOCHINDA, 2012; YAUMI; ABU BAKAR; HAMEED, 2017). Desta forma, a obtenção de materiais de baixo custo para a captura tornou-se um dos principais focos de pesquisa neste campo (ZULFIQAR; SARWAR; MECERREYES, 2015a).

Os sorventes empregados na captura de CO<sub>2</sub> precisam ser econômicos e operacionais. Para tal, o ideal seria que apresentassem alta capacidade de sorção com elevada seletividade, estabilidade, resistência química e térmica, baixo consumo de energia, cinética de sorção/dessorção rápida e baixo custo (YAUMI; ABU BAKAR; HAMEED, 2017). No entanto, é muito raro que um sorvente cumpra todos os critérios. Assim, os melhores sorventes são aqueles que podem capturar o CO<sub>2</sub> de forma efetiva e econômica (YAUMI; ABU BAKAR; HAMEED, 2017). O aperfeiçoamento ou desenvolvimento de Poli (líquidos iônicos) (PLIs) ou aminas suportadas em materiais sólidos têm sido relatado em diversos trabalhos, como opções promissoras para separação de CO<sub>2</sub> de diversos tipos de correntes gasosas (WATABE; YOGO, 2013; ZULFIQAR; SARWAR; MECERREYES, 2015b; QIAN; TEXTER; YAN, 2017; YAUMI; ABU BAKAR; HAMEED, 2017).

A captura de CO<sub>2</sub> utilizando aminas suportadas em materiais sólidos requer menos energia para a etapa de regeneração quando comparadas com soluções aquosas de aminas, pois não existe a necessidade de água no processo. Além disso, também demonstram maior estabilidade térmica e resistência a contaminantes presentes nos gases de combustão (BACHELOR; TOOCHINDA, 2012; YU; HUANG; TAN, 2012; WATABE; YOGO, 2013; YAUMI; ABU BAKAR; HAMEED, 2017). No entanto, o alto custo dos suportes comerciais é um desafio que ainda precisa ser superado para a consolidação do uso destes materiais em plantas de captura (BACHELOR; TOOCHINDA, 2012).

Poli(líquidos iônicos) ou líquidos iônicos polimerizados (PLIs) são uma subclasse de polieletrólitos que apresentam uma espécie de líquido iônico em cada

unidade de repetição, conectados através de uma estrutura polimérica para formar uma estrutura macromolecular (ZULFIQAR; SARWAR; MECERREYES, 2015a; TOME; MARRUCHO, 2016; QIAN; TEXTER; YAN, 2017). Estes compostos combinam os benefícios da utilização dos líquidos iônicos (LI) com as propriedades dos polímeros (YUAN; ANTONIETTI, 2011; TOME; MARRUCHO, 2016; QIAN; TEXTER; YAN, 2017). Outra característica importante destes materiais deve-se ao fato da sorção e dessorção de CO<sub>2</sub> ser completamente reversível e ocorrer de forma mais rápida do que nos líquidos iônicos (TANG et al., 2005a; ZULFIQAR; SARWAR; MECERREYES, 2015a).

Neste contexto, este trabalho visou à síntese e avaliação do potencial para captura de CO<sub>2</sub> de diferentes tipos de Poli (líquidos iônicos), baseados em polímeros sintéticos (poliuretano e poliuretano-imida) e naturais (celulose extraída da casca de arroz). Também foi avaliado a solubilidade do CO<sub>2</sub> em aminas e polieteramias suportadas em celulose extraída da casca de arroz, com o propósito de produzir adsorventes químicos de baixo custo e contribuir para a redução da geração de resíduos agrícolas.

## **2. OBJETIVOS**

Este trabalho tem como objetivo principal a síntese e a avaliação da capacidade de sorção de CO<sub>2</sub> de diferentes materiais poliméricos que contenham líquidos iônicos ou aminas em sua estrutura.

### **2.1. Objetivos Específicos**

- Sintetizar poli (líquidos iônicos) aniônicos baseados em polímeros sintéticos (poliuretano ou poliuretano-imida);
- Extrair celulose a partir da casca de arroz por meio de métodos químicos
- Sintetizar poli (líquidos iônicos) catiônicos e aniônicos base celulose;
- Investigar a influência dos grupos funcionais presentes nos materiais sintetizados sobre a sorção de CO<sub>2</sub>, utilizando dados experimentais e simulação computacional.

### 3. REVISÃO BIBLIOGRÁFICA

#### 3.1. Tecnologias para mitigação da emissão de CO<sub>2</sub>

A emissão de gases de efeito estufa de origem antropogênica, especialmente o CO<sub>2</sub>, resultam de um amplo conjunto de atividades humanas, principalmente aquelas associadas com a geração de energia na área de transportes, ambientes urbanos e setores industriais (MONTZKA; DLUGOKENCKY; BUTLER, 2011; SONGOLZADEH et al., 2014; WANG et al., 2015; BRUHN; NAIMS; OLFE-KRÄUTLEIN, 2016; YAUMI; ABU BAKAR; HAMEED, 2017). As emissões de CO<sub>2</sub> para atmosfera continuam aumentando em ritmo acelerado e tem provocado alterações profundas no ecossistema (NOAA – NATIONAL ORGANIC & ATMOSPHERIC ADMINISTRATION, 2017; YAUMI; ABU BAKAR; HAMEED, 2017). A Figura 3.1, apresenta um gráfico que demonstra a taxa de crescimento da concentração de CO<sub>2</sub> em ppm/ano, medido em Mauna Loa no Havaí.

Estima-se que a concentração atmosférica global de dióxido de carbono tenha aumentado desde 1750 (pré- revolução industrial) de um valor de cerca de 280 ppm para 379 ppm em 2005, com um aumento da temperatura média global da terra de 0,6 °C para 1,0 °C dentro deste período (YAUMI; ABU BAKAR; HAMEED, 2017). As projeções do Painel Internacional de Alterações Climáticas (IPCC) indicam que, até 2100, a concentração atmosférica global de CO<sub>2</sub> atinja 570 ppm com um aumento médio de temperatura de 1,9°C (YAUMI; ABU BAKAR; HAMEED, 2017). Nos dias atuais, Janeiro de 2017, a concentração alcançou valores próximos a 406,13 ppm (Figura 3.1), com tendência de aumento, devido à enorme demanda global por combustíveis fósseis (NOAA – NATIONAL ORGANIC & ATMOSPHERIC ADMINISTRATION, 2017; YAUMI; ABU BAKAR; HAMEED, 2017).

Logo, do ponto de vista ambiental, reduzir os níveis crescentes de CO<sub>2</sub> na atmosfera torna-se essencial para mitigar os impactos causados pelo aquecimento global e as mudanças climáticas imprevisíveis (CHENG et al., 2015; ZULFIQAR; SARWAR; MECERREYES, 2015b; TOME; MARRUCHO, 2016).

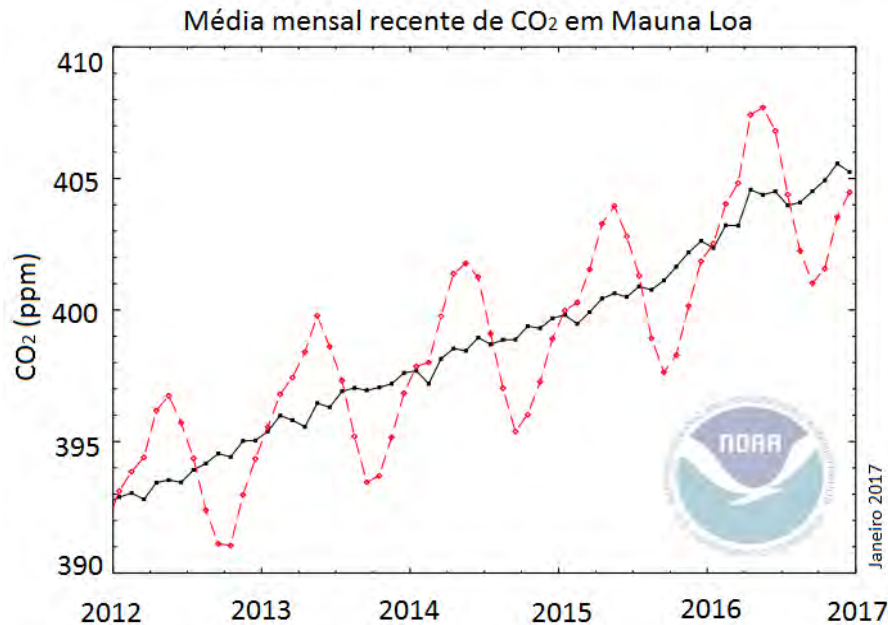


Figura 3.1. Aumento da concentração de CO<sub>2</sub> e as mudanças climáticas. Fonte: (NOAA – NATIONAL ORGANIC & ATMOSPHERIC ADMINISTRATION, 2017). A linha vermelha tracejada representa os valores médios mensais e a linha preta representa o mesmo, após correção para a média do ciclo sazonal.

O uso de tecnologias de captura de carbono tem sido indicado como um meio capaz para mitigar as emissões de CO<sub>2</sub> de grandes fontes geradoras de energia (TOME; MARRUCHO, 2016). As rotas tecnológicas para captura de CO<sub>2</sub> mais discutidas atualmente para instalações que utilizam combustíveis fósseis ou biomassa são: Pós - combustão, Pré - combustão e Oxi - combustão (Figura 3.2) (FIGUEROA et al., 2008; OLAJIRE, 2010; MONDAL; BALSORA; VARSHNEY, 2012; SONGOLZADEH et al., 2014; KETZER; IGLESIAS; EINLOFT, 2015; YAUMI; ABU BAKAR; HAMEED, 2017).

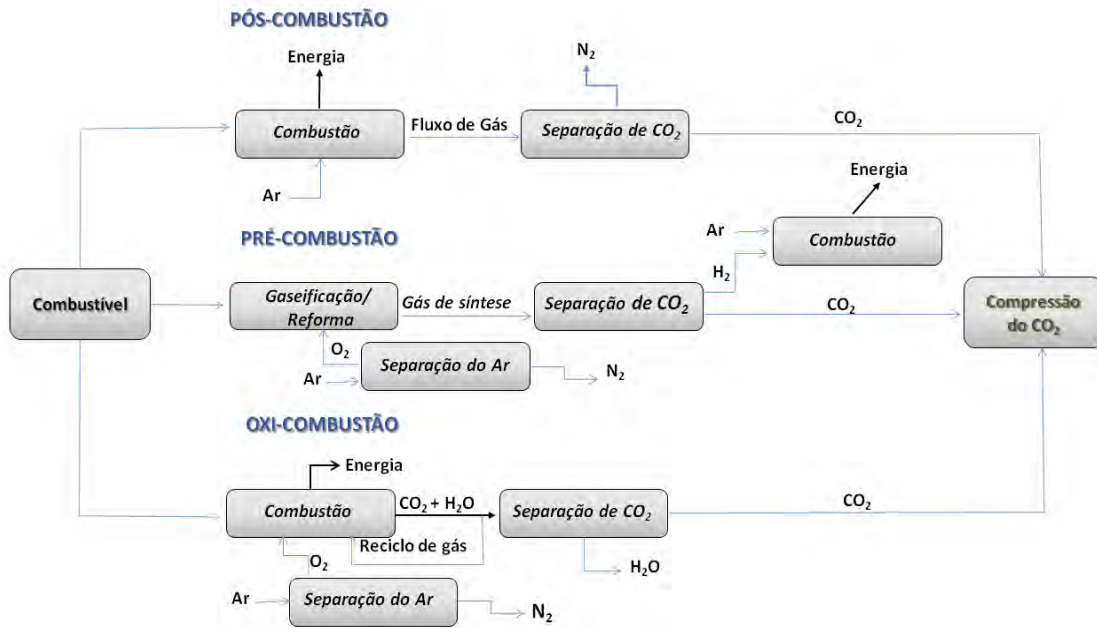


Figura 3.2. Fluxograma das tecnologias de captura de CO<sub>2</sub>. Fonte: Adaptação (FIGUEROA et al., 2008).

A remoção do dióxido de carbono de gases produzidos e emitidos para a atmosfera em processos de queima de combustível fóssil ou biomassa é normalmente realizada pela tecnologia de Pós-combustão. Os gases de combustão, normalmente apresentam baixas concentrações de CO<sub>2</sub> (3 - 15 % para usinas de carvão, 7 - 8 % para usinas de gás (por volume)) (MONDAL; BALSORA; VARSHNEY, 2012; SONGOLZADEH et al., 2014) e saem à pressão atmosférica e a temperaturas entre 47°C e 127°C (SONGOLZADEH et al., 2014). A condição desfavorável dos gases de combustão (pressão e concentração baixas) para captura de CO<sub>2</sub> é apresentada como a principal desvantagem desta tecnologia (FIGUEROA et al., 2008; OLAJIRE, 2010; SONGOLZADEH et al., 2014) e a sua principal vantagem está relacionada ao uso de processos de baixo custo (SONGOLZADEH et al., 2014). Processos de absorção química se adaptam melhor a esta tecnologia (MONDAL; BALSORA; VARSHNEY, 2012), soluções aquosas de aminas (monoetanolamina (MEA), dietanolamina (DEA) e metildietanolamina (MDEA) destacam-se como os solventes mais empregados (YANG et al., 2008; BACHELOR; TOOCHINDA, 2012; RAMDIN et al., 2014; KETZER; IGLESIAS; EINLOFT, 2015; YAUMI; ABU BAKAR; HAMEED, 2017). Contudo, o uso deste tipo de solvente oferece algumas desvantagens operacionais, dentre as quais podem-se destacar o elevado consumo de energia (necessário para a etapa de regeneração), a corrosão

de equipamentos causada pela formação de subprodutos corrosivos, a necessidade de absorvedores de grande volume, além de custos operacionais adicionais devido à substituição de solventes decorrente da degradação (YANG et al., 2008; BACHELOR; TOOCHINDA, 2012; RAMDIN et al., 2014; SEO et al., 2014; KETZER; IGLESIAS; EINLOFT, 2015; YAUMI; ABU BAKAR; HAMEED, 2017). Outros processos de separação também podem ser empregados, como por exemplo: separação por membranas e adsorção em sólidos (OLAJIRE, 2010; BACHELOR; TOOCHINDA, 2012).

A captura de CO<sub>2</sub> de gases produzidos no sistema de pré-combustão ocorre a partir da produção de uma mistura gasosa contendo principalmente CO e H<sub>2</sub> (gás de síntese). Esta mistura é produzida pela reação do ar ou oxigênio com vapor, o monóxido de carbono produzido é colocado em contato com vapor de água resultando na produção de mais hidrogênio e CO<sub>2</sub> (teores na faixa de 15 a 60 % vol) a alta pressão. Finalmente o dióxido de carbono é separado e o hidrogênio pode ser usado como combustível (FIGUEROA et al., 2008; OLAJIRE, 2010; MARKEWITZ et al., 2012; SONGOLZADEH et al., 2014; KETZER; IGLESIAS; EINLOFT, 2015). Embora o principal inconveniente do sistema de pré-combustão esteja relacionado com o alto custo necessário para a instalação, a alta pressão e a concentração de CO<sub>2</sub> permitem a diminuição dos custos nas etapas de carga e compressão, bem como o emprego de processos de separação de CO<sub>2</sub> que necessitem de baixo consumo de energia na etapa de regeneração (ex, absorventes e adsorventes físicos) (FIGUEROA et al., 2008; OLAJIRE, 2010).

Na oxi-combustão, ao invés do ar, usa-se oxigênio praticamente puro (pureza >95%) para realização da combustão. Inicialmente o O<sub>2</sub> é separado do ar, em seguida o combustível é queimado com O<sub>2</sub> e uma mistura de gás de combustão rico em CO<sub>2</sub> e vapor de água é obtida e reciclada. Os gases de combustão contém CO<sub>2</sub> com teor superior a 80% vol., assim a água pode ser facilmente removida por condensação (FIGUEROA et al., 2008; OLAJIRE, 2010; MARKEWITZ et al., 2012; SONGOLZADEH et al., 2014; KETZER; IGLESIAS; EINLOFT, 2015). A maior vantagem desta tecnologia é a alta pressão parcial do CO<sub>2</sub> que permite sua fácil separação dos gases exaustos (KETZER; IGLESIAS; EINLOFT, 2015). No entanto,

esta tecnologia necessita de grande consumo de  $O_2$ , o qual tem custo elevado para produção (FIGUEROA et al., 2008; OLAJIRE, 2010; SONGOLZADEH et al., 2014).

### 3.2. Líquidos iônicos e seu potencial para captura de $CO_2$

Os Líquidos iônicos (LIs) são uma classe de sais orgânicos, formados pela combinação de cátions orgânicos e ânions inorgânicos ou orgânicos, que possuem temperatura de fusão inferior a  $100\text{ }^\circ\text{C}$  (WILKES, 2002; FIGUEROA et al., 2008; PRIVALOVA et al., 2012; BABAMOHAMMADI; SHAMIRI; AROUA, 2015; SHAPLOV et al., 2016). Estes compostos exibem propriedades únicas como a capacidade de não inflamar, pressão de vapor extremamente baixa e boa estabilidade térmica (WILKES, 2002; FIGUEROA et al., 2008; HASIB-UR-RAHMAN; SIAJ; LARACHI, 2010; RAMDIN; DE LOOS; VLUGT, 2012; QIAN; TEXTER; YAN, 2017). Além disso, são versáteis e menos prejudiciais ao meio-ambiente do que os solventes orgânicos convencionais. A grande variabilidade estrutural de cátions e ânions permite a criação de compostos com características e propriedades específicas para uma determinada aplicação (BABAMOHAMMADI; SHAMIRI; AROUA, 2015; DAI et al., 2016; TOME; MARRUCHO, 2016). A Figura 3.3 mostra as estruturas de alguns cátions e ânions comumente utilizados para obtenção de LIs.

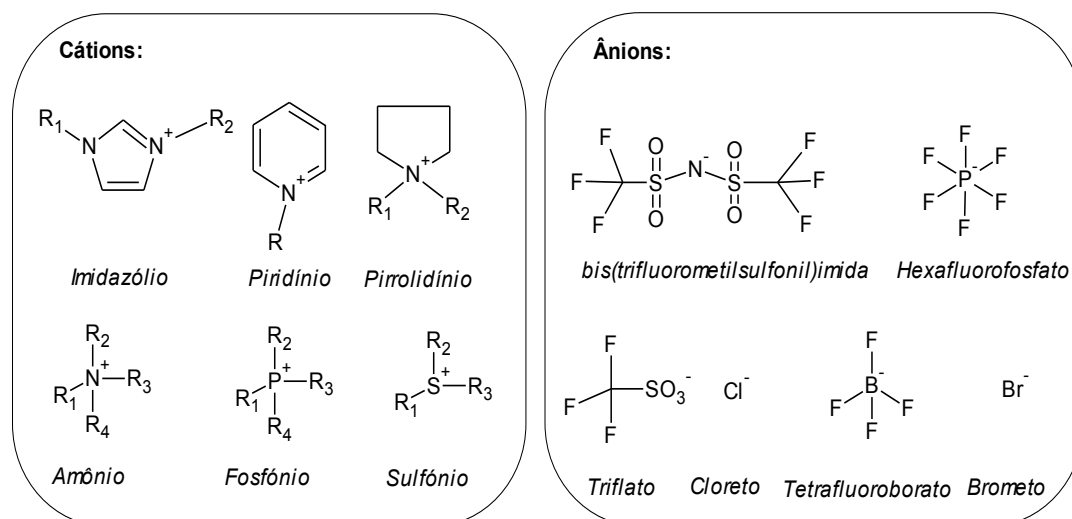


Figura 3.3. Exemplo de estruturas típicas de cátions e ânions comumente empregados em sínteses de líquidos iônicos. Fonte: Baseado em M. Hasib-ur-Rahman et al.,2010 (HASIB-UR-RAHMAN; SIAJ; LARACHI, 2010).



Em geral, os LIs são divididos em duas grandes categorias: Líquidos Iônicos à Temperatura Ambiente (RTILs: do inglês "*Room Temperature Ionic Liquids*") considerados como os LIs tradicionais ou convencionais e, os líquidos iônicos específicos (TSILs: do inglês "*task-specific ionic liquids*"), isto é, que apresentam grupos funcionais reativos em sua estrutura (HASIB-UR-RAHMAN; SIAJ; LARACHI, 2010; PRIVALOVA et al., 2012; BABAMOHAMMADI; SHAMIRI; AROUA, 2015; TOME; MARRUCHO, 2016).

Os LIs vem demonstrando potencial para captura de gases ácidos, como por exemplo, o CO<sub>2</sub> (HASIB-UR-RAHMAN; SIAJ; LARACHI, 2010; BABAMOHAMMADI; SHAMIRI; AROUA, 2015; TOME; MARRUCHO, 2016). Isto se deve principalmente às suas propriedades e características peculiares aliado ao fato da grande maioria dos LIs demonstrarem absorção seletiva para o CO<sub>2</sub> em uma mistura de gases (CADENA et al., 2004; ANTHONY et al., 2005; BABAMOHAMMADI; SHAMIRI; AROUA, 2015; TOME; MARRUCHO, 2016). Entretanto, existem alguns empecilhos que precisam ser superados para a sua ampla utilização em plantas de captura de CO<sub>2</sub>, como por exemplo, o alto custo de produção em comparação com as soluções aquosas de aminas, a elevada viscosidade e a baixa taxa de absorção/dessorção de CO<sub>2</sub> (HASIB-UR-RAHMAN; SIAJ; LARACHI, 2010; KENARSARI et al., 2013). Avanços recentes sugerem que estas dificuldades podem ser superadas com o uso de líquidos iônicos polimerizados (ZULFIQAR; SARWAR; MECERREYES, 2015a; SADEGHPOUR; YUSOFF; AROUA, 2016).

Líquidos iônicos tradicionais ou convencionais são os mais investigados para utilização em processo de absorção de gases, em especial para o CO<sub>2</sub>. É possível que este interesse esteja relacionado com o fato desta categoria de LIs apresentar comportamento típico de solvente físico, ou seja, a solubilidade dos gases tende a aumentar com o aumento da pressão e o processo de sorção/dessorção seja reversível (ANTHONY et al., 2005; HASIB-UR-RAHMAN; SIAJ; LARACHI, 2010; PRIVALOVA et al., 2012; RAMDIN; DE LOOS; VLUGT, 2012; BABAMOHAMMADI; SHAMIRI; AROUA, 2015).

Estudos sobre a interação do CO<sub>2</sub> com LIs tradicionais tem evidenciado que o ânion desempenha um papel chave na dissolução de CO<sub>2</sub>, enquanto que o cátion

tem um papel secundário. Porém, em menor proporção, modificações no cátion envolvendo grupos alquil podem influenciar a capacidade de absorção de CO<sub>2</sub>. Trabalhos relatam também que a fluoração do ânion e do cátion pode contribuir significativamente para o aumento da solubilidade do CO<sub>2</sub> nestes compostos (CADENA et al., 2004; ANTHONY et al., 2005; RAMDIN; DE LOOS; VLUGT, 2012; TOME; MARRUCHO, 2016).

### **3.3. Poli (líquidos iônicos) desenvolvidos para captura de CO<sub>2</sub>**

Poli (líquidos iônicos) ou líquidos iônicos poliméricos (PLIs) apresentam uma espécie de líquido iônico (LI) conectados através de uma estrutura polimérica para formar uma estrutura macromolecular (YUAN; MECERREYES; ANTONIETTI, 2013; ZULFIQAR; SARWAR; MECERREYES, 2015a; SHAPLOV et al., 2016; QIAN; TEXTER; YAN, 2017). Estes polímeros funcionais tornaram-se uma nova plataforma, extremamente versátil e ajustável para o desenvolvimento de uma grande variedade de sorventes para captura de CO<sub>2</sub> (TANG et al., 2009; HASIB-UR-RAHMAN; SIAJ; LARACHI, 2010; ZULFIQAR; SARWAR; MECERREYES, 2015a; TOME; MARRUCHO, 2016) combinando características notáveis dos líquidos iônicos (LIs), em especial, a possibilidade de variar suas propriedades físicas e químicas, por meio da combinação de cátions e ânions (YUAN; ANTONIETTI, 2011; TOME; MARRUCHO, 2016), com as propriedades dos polímeros (estabilidade mecânica, processabilidade e design macromolecular) (YUAN; ANTONIETTI, 2011; PRIVALOVA et al., 2012; TOME et al., 2013; YUAN; MECERREYES; ANTONIETTI, 2013; TOME; MARRUCHO, 2016).

Outra característica que deve ser ressaltada nestes materiais é o fato do processo de sorção e dessorção do CO<sub>2</sub> ser completamente reversível e ocorrer de forma mais rápida do que nos líquidos iônicos tradicionais (TANG et al., 2005a). Além disso, também tem sido reportado que a sorção de CO<sub>2</sub> em PLIs é seletiva (TANG et al., 2005a).

O método de síntese está diretamente ligado ao tipo de poli(líquidos iônico) que se deseja obter (YUAN; MECERREYES; ANTONIETTI, 2013). Em geral, a

síntese dos PLIs pode ser realizada por meio de duas rotas principais: a primeira consiste na polimerização direta de monômeros de LIs (TOME; MARRUCHO, 2016) e a segunda envolve basicamente reações de condensação, que podem ser conduzidas através da modificação de polímeros (SHAPLOV et al., 2016; TOME; MARRUCHO, 2016) ou via condensação do respectivo monômero (SHAPLOV et al., 2016). PLIs com diferentes estruturas podem ser produzidos, e certamente o tipo de rota selecionada pode afetar diretamente as características, propriedades e o custo destes materiais, acarretando em vantagens e desvantagens (YUAN; MECERREYES; ANTONIETTI, 2013).

### 3.3.1. Poli (líquidos iônicos) obtidos por polimerização radicalar direta

Segundo a ideia dos LIs, a grande maioria dos PLIs sintetizados via polimerização radicalar direta é do tipo catiônico, ou seja, a estrutura da cadeia principal apresenta porções catiônicas enquanto que a parte móvel (contra - íon) é a aniônica (MECERREYES, 2011). O imidazólio é o cátion mais utilizado, podendo ser inserido na estrutura da cadeia principal do polímero a partir de estruturas como vinil, estireno, metacrilato entre outros (MECERREYES, 2011). O método de reação por troca aniônica de um monômero catiônico contendo contra-ânion do tipo haleto (Figura 3.4) é o mais empregado para obtenção de polications (MECERREYES, 2011; SADEGHPOUR; YUSOFF; AROUA, 2016). Em geral, esta rota utiliza monômeros de LIs, Azobisisobutironitrila (AIBN) e solvente dimetilformamida (DMF) (SADEGHPOUR; YUSOFF; AROUA, 2016).

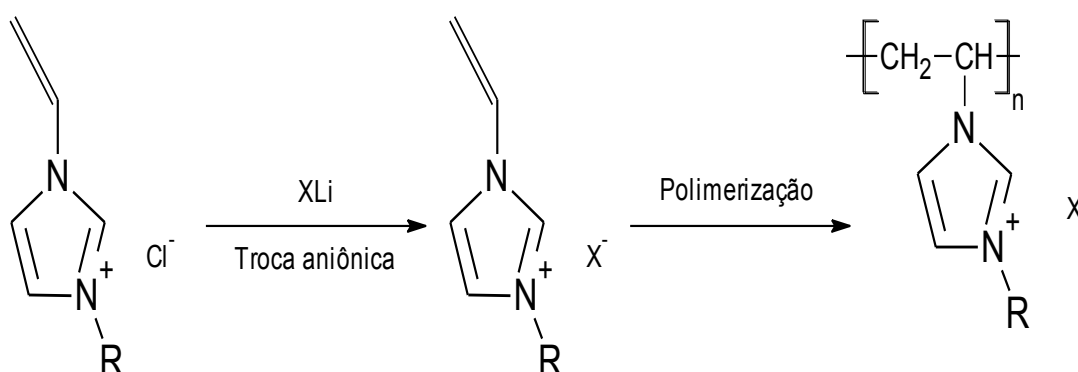


Figura 3.4. Rota sintética geral para obtenção de PLIs via reação por troca aniônica. Adaptação Mecerreyes, 2011 (MECERREYES, 2011).

Os primeiros estudos sobre a solubilidade de  $\text{CO}_2$  em PLIs obtidos via síntese radicalar foram reportados no ano de 2005 por Tang e colaboradores (TANG et al., 2005a, 2005b, 2005c, 2005d). Estes trabalhos despertaram um enorme interesse da comunidade científica e alavancaram as pesquisas sobre a utilização de PLIs para captura de  $\text{CO}_2$ .

Inicialmente Tang e co-autores investigaram o desempenho de três PLIs base imidazólio: poli[1-p-(vinilbenzil)-3-butilimidazólio tetrafluoroborato] (PVBIT), poli[1-p-(vinilbenzil)-3-butilimidazólio hexafluorofosfato] (PVBIH) e poli [2-(metacrilóiloxi)etil-3-butilimidazólio tetrafluoroborato] (PBIMT). Surpreendentemente, como pode ser verificado na Figura 3.5, os PLIs exibiram capacidade de sorção superior à do líquido iônico 1-butil-3-metilimidazólio tetrafluoroborato ([bmim][BF<sub>4</sub>]) e de seus monômeros correspondentes (TANG et al., 2005a). Os PLIs sintetizados por Tang et. al., 2005 (TANG et al., 2005a) necessitaram de apenas alguns minutos para alcançar 90% da sua capacidade de sorção de  $\text{CO}_2$  e cerca de 30 minutos para atingir o equilíbrio, enquanto que para o [bmim][BF<sub>4</sub>] foi necessário mais de 400 min para que a sua capacidade de sorção total fosse atingida (Figura 3.5).

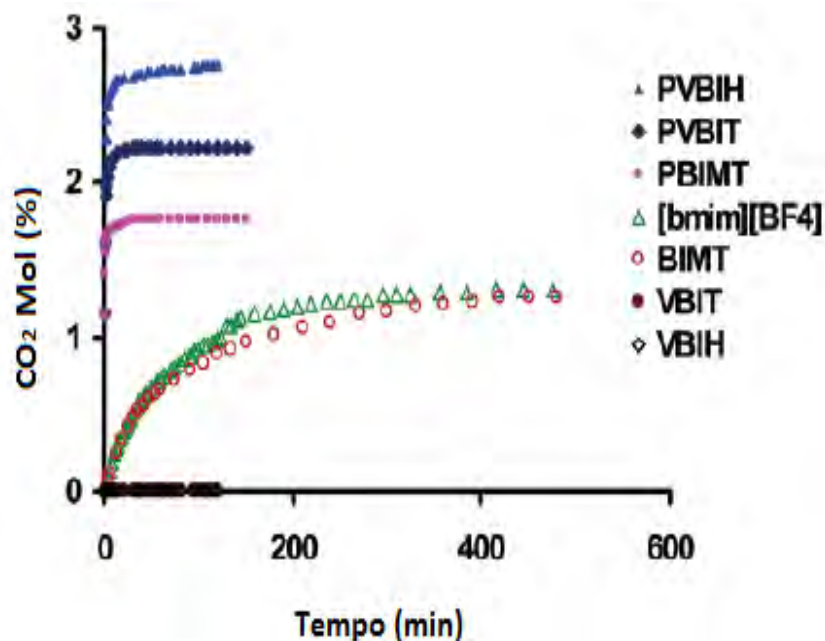


Figure 3.5: Sorção de  $\text{CO}_2$  ao longo do tempo para os PLIs: PVBIH, PVBIT, PBIMT e seus respectivos monômeros (VBIT, VBIH, BIMT). Condições: (0,79 bar e 22 °C). Fonte: Tang et. al., 2005 (TANG et al., 2005a).

Ainda neste trabalho (TANG et al., 2005a), os PLIs PVBIT e PBIMT foram submetidos a quatro ciclos de sorção/dessorção (Figura 3.6a) e o PLI PVBIT foi avaliado quanto a sua seletividade (Figura 3.6b). Os resultados experimentais de sorção/dessorção apresentados na Figura 3.6a não demonstraram variação na capacidade de sorção de  $\text{CO}_2$  dos PLIs PVBIT e PBIMT após a realização de cada ciclo. Isto sugere que a dessorção ocorreu de forma completa e que o processo é completamente reversível (TANG et al., 2005a). Também foi verificado que o PLI PVBIT é seletivo para  $\text{CO}_2$ , pois não foi observado nenhum ganho de massa quando este PLI foi exposto a  $\text{N}_2$  ou  $\text{O}_2$  (TANG et al., 2005a) (Figura 3.6b).

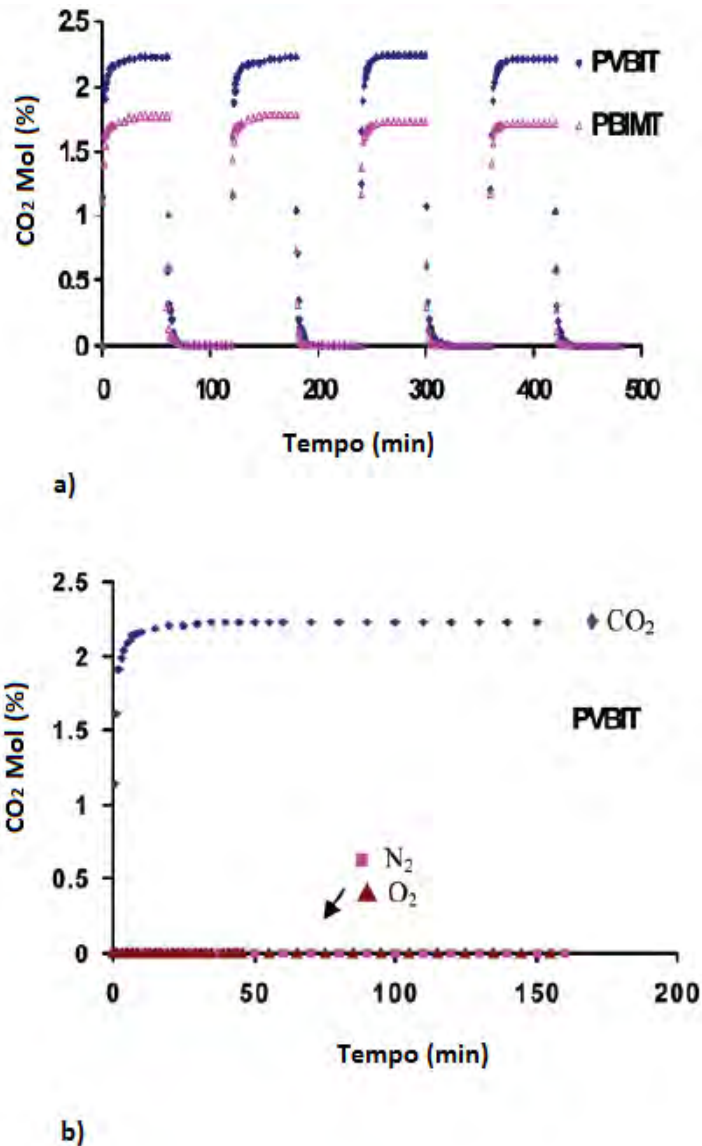


Figura 3.6. a) Ciclos de sorção (592,3 mmHg e 22 °C /dessorção com vácuo em PVBIT e PBIMT e b) Sorção dos Gases:  $\text{CO}_2$ ,  $\text{O}_2$  e  $\text{N}_2$  em PVBIT .Condições : 592,3 mmHg e 22 °C.

Diversos estudos têm sido realizados sobre o uso de PLIs catiônicos sintetizados via polimerização radicalar direta de monômeros em captura de CO<sub>2</sub> (TANG et al., 2005a, 2005b, 2005d, 2009; SIMONS et al., 2010; BHAVSAR; KUMBHARKAR; KHARUL, 2012; CARLISLE et al., 2013; YU et al., 2014). Resultados experimentais de sorção de CO<sub>2</sub> demonstraram que além da pressão e temperatura (BLASIG et al., 2007) vários aspectos podem influenciar na solubilidade do CO<sub>2</sub> em PLIs. Neste cenário, o tipo de cátion, ânion e a estrutura da cadeia principal merecem destaque (ZULFIQAR; SARWAR; MECERREYES, 2015a; SADEGHPOUR; YUSOFF; AROUA, 2016; TOME; MARRUCHO, 2016; QIAN; TEXTER; YAN, 2017).

A maior parte do PLIs desenvolvidos para uso em captura e separação de CO<sub>2</sub> tem utilizado metacrilato (TANG et al., 2005b, 2005d, 2009; BARA et al., 2007; TOMÉ et al., 2015) e estireno (TANG et al., 2005d; BLASIG et al., 2007; BARA JASON E. , HATAKEYAMA EVAN S. , GIN DOUGLAS L., 2008; SIMONS et al., 2010) como estrutura da cadeia principal. No entanto, outras estruturas também têm sido exploradas, tais como o vinil (CARLISLE et al., 2013) e o dialildimetilamônio (BHAVSAR; KUMBHARKAR; KHARUL, 2012; TOMÉ et al., 2013, 2015).

Testes experimentais de sorção de CO<sub>2</sub> revelaram que a utilização do estireno ao invés de metacrilato pode aumentar a solubilidade de CO<sub>2</sub> em PLIs, provavelmente devido a sua estrutura mais rígida (TANG et al., 2005b, 2009). Por exemplo, o poli [p-(vinilbenzil)-trimetilamônio tetrafluorborato] (P[VBTMA][BF<sub>4</sub>]) com estrutura principal formada por estireno mostrou capacidade de sorção de CO<sub>2</sub> superior ao poli [2-(metacrilóiloxi)etil-trimetilamônio tetrafluorborato] (P[MATMA][BF<sub>4</sub>]) desenvolvido a partir de metacrilato (TANG et al., 2005b, 2005d, 2009). Comportamento similar foi observado em PLIs de cátion imidazólio, onde o poli [p-(vinilbenzil)-3-butilimidazólio tetrafluorborato] (P[VBBI][BF<sub>4</sub>]) demonstrou capacidade de sorção cerca de 20% (2.27 mol%) superior ao poli [2-(metacrilóiloxi)etil-3-butilimidazólio tetrafluorborato] (P[MABI][BF<sub>4</sub>]) (1.80 mol%) (TANG et al., 2005b).

A permeabilidade e seletividade ao CO<sub>2</sub> destes materiais também podem sofrer influência considerável da estrutura da cadeia principal empregada em sua síntese. Normalmente, PLIs à base de estireno e metacrilato tendem a ser mais seletivos para CO<sub>2</sub> que os a base de vinil (BARA et al., 2007; CARLISLE et al., 2013). Além disso, a permeabilidade do CO<sub>2</sub> em PLIs com a estrutura formada por estireno tende a ser mais elevada que os sintetizados com metacrilato e vinil (BARA et al., 2007; CARLISLE et al., 2013).

A influência da reticulação de PLIs catiônicos sobre a capacidade de sorção de CO<sub>2</sub> também tem sido avaliada (TANG et al., 2009; YU et al., 2014). Yu et al, 2014 (YU et al., 2014) sintetizou o PLI N,N-metilenobisacrilamida (MBA) reticulado - poli [4-(vinilbenzil)-triethylamônio hexafluorofosfato] (MBA-reticulado- P[VBTEA][PF<sub>6</sub>]) e comparou seus resultados de sorção de CO<sub>2</sub> com o poli [4-(vinilbenzil)-triethylamônio hexafluorofosfato] (P[VBTEA][PF<sub>6</sub>]). Conforme pode-se observar na Figura 3.7, o PLI (MBA-reticulado- P[VBTEA][PF<sub>6</sub>]) demonstrou capacidade de sorção de CO<sub>2</sub> superior ao (P[VBTEA][PF<sub>6</sub>]). Este comportamento está relacionado com o aumento da área específica e formação de poros promovida pela reticulação (YU et al., 2014).

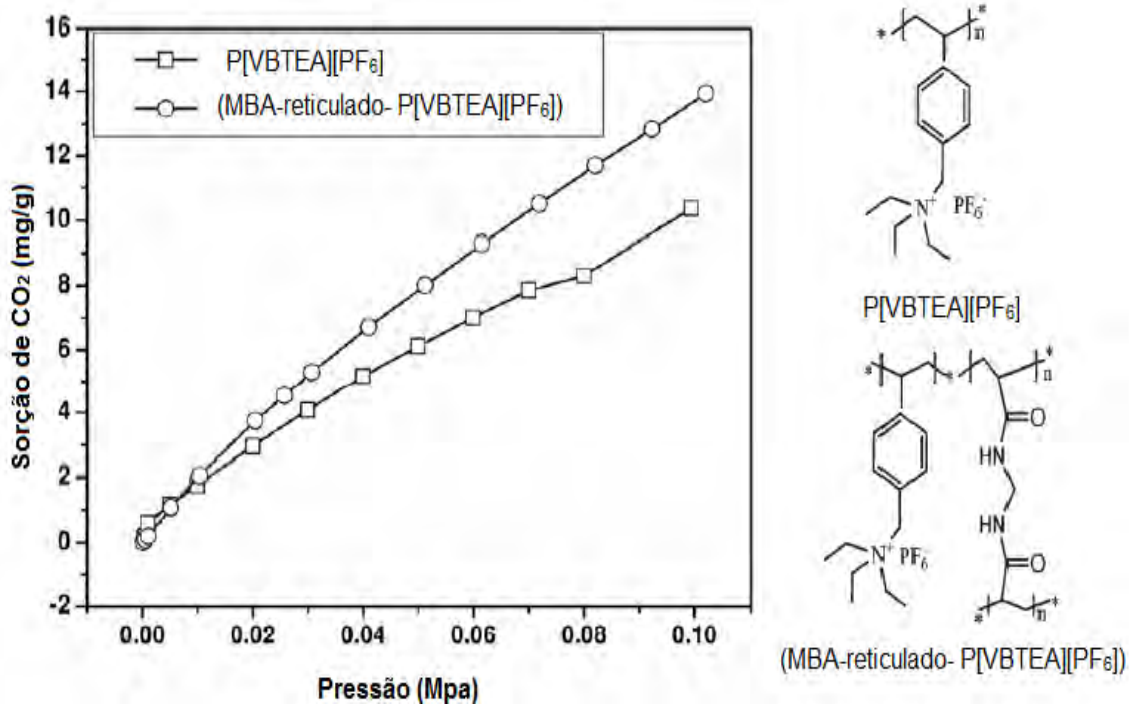


Figura 3.7. Isotherma de sorção de CO<sub>2</sub> do (MBA-reticulado- P[VBTEA][PF<sub>6</sub>]) e P[VBTEA][PF<sub>6</sub>] a 25°C avaliando a influência da reticulação na capacidade de sorção de CO<sub>2</sub>. Adaptação: Yu et al., 2014 (YU et al., 2014).

Comportamento distinto foi observado por Tang e colaboradores, 2009 (TANG et al., 2009) com a inserção de 5% de reticulação no poli [1-(p-vinilbenzil)-trimetilamônio tetrafluoroborato] (P[VB-TMA]-[BF<sub>4</sub>]). Neste caso, a rigidez decorrente da reticulação dificultou a interação do CO<sub>2</sub> com o PLI, e, conseqüentemente reduziu os valores de sorção de CO<sub>2</sub> (TANG et al., 2009).

Ao contrário do observado em líquidos iônicos convencionais, nos PLIs o ânion desenvolve um papel secundário sobre a sorção de CO<sub>2</sub> enquanto que o cátion desempenha o papel principal (TANG et al., 2005d, 2009). Estudos de simulação atomística (Figura 3.8 e 3.9) utilizando o policátion (1-vinil-3-butylimidazólio) [VBIM]<sup>+</sup> e diferentes ânions foram conduzidos por Fang et al., 2013 com o objetivo de compreender a interação entre o CO<sub>2</sub> e os PLIs catiônicos.

A Figura 3.8 ilustra as funções de distribuição radial  $g(r)$  do CO<sub>2</sub> no LI [BMIM] [TF<sub>2</sub>N] e no poli [VBIM] [TF<sub>2</sub>N] (FANG; LUO; JIANG, 2013). A tendência de distribuição radial em torno do [TF<sub>2</sub>N]<sup>-</sup> do LI ou PLI é praticamente idêntica. No entanto, uma diferença é notada entre a distribuição radial obtida com o poli [VBIM]<sup>-</sup> e para o [BMIM]<sup>-</sup>. Para o LI [BMIM] [TF<sub>2</sub>N] tanto o cátion quanto o ânion demonstram uma interação estreita com o CO<sub>2</sub>, três picos acentuados são verificados, sendo dois picos para os átomos da extremidade do cátion [BMIM]<sup>+</sup> (C1 e C8) um pico para o átomo C1 do ânion [TF<sub>2</sub>N]<sup>-</sup>. Em contraste, no poli [VBIM] [TF<sub>2</sub>N]<sup>-</sup> a distribuição radial  $g(r)$  do CO<sub>2</sub> é mais intensa para o átomo C1 do policátion. Isto sugere que o CO<sub>2</sub> interage mais fortemente com o átomo C1 do policátion do que com os outros átomos do cátion ou ânion. O átomo C1 do policátion está localizado na extremidade da cadeia polimérica, isto lhe confere elevada mobilidade e conseqüentemente grande volume livre disponível para acomodar o CO<sub>2</sub>, enquanto que nesta configuração o átomo C8 não está acessível ao CO<sub>2</sub> (FANG; LUO; JIANG, 2013).



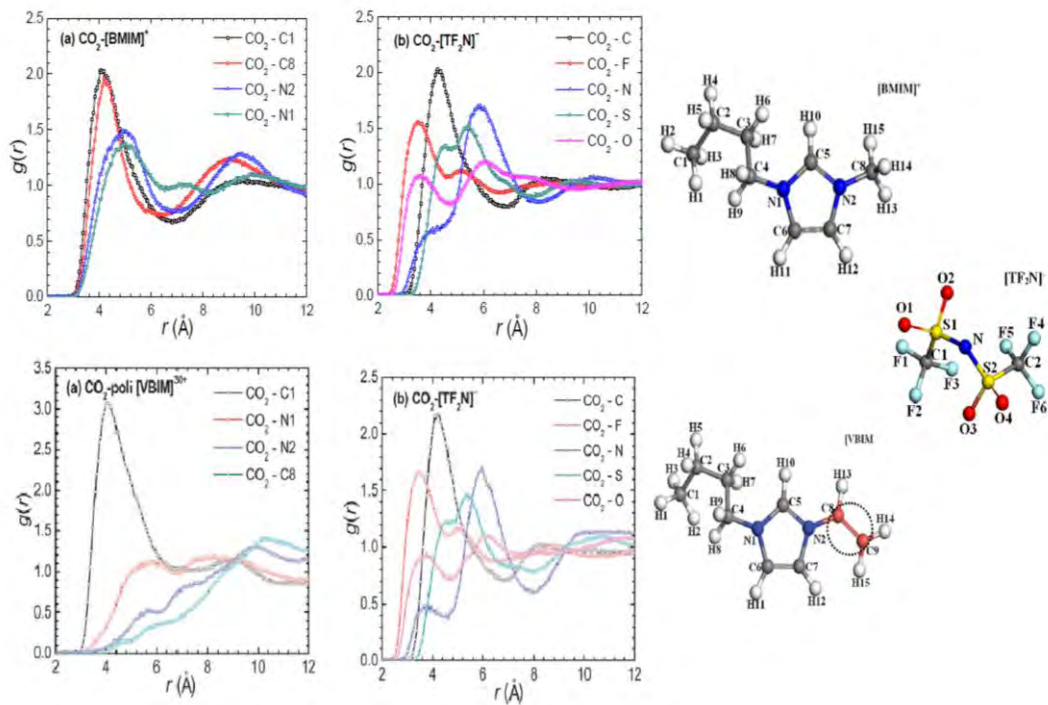


Figura 3.8. Funções de distribuição radial em [BMIM] [TF<sub>2</sub>N] e poli [VBIM] [TF<sub>2</sub>N]. Fonte: (FANG; LUO; JIANG, 2013)

A avaliação da distribuição radial  $g(r)$  de CO<sub>2</sub> nos poli [VBIM] [PF<sub>6</sub>] e [VBIM] [Cl] (Figura 3.9), demonstrou comportamento semelhante ao observado no poli [VBIM] [TF<sub>2</sub>N] (FANG; LUO; JIANG, 2013). O pico  $g(r)$  do CO<sub>2</sub> mostrou-se mais elevado em torno do átomo C1 do polycátion. Este resultado corrobora a teoria de que o CO<sub>2</sub> possui maior afinidade pela cadeia lateral do polycátion do que pelos átomos presentes na estrutura do ânion (FANG; LUO; JIANG, 2013).

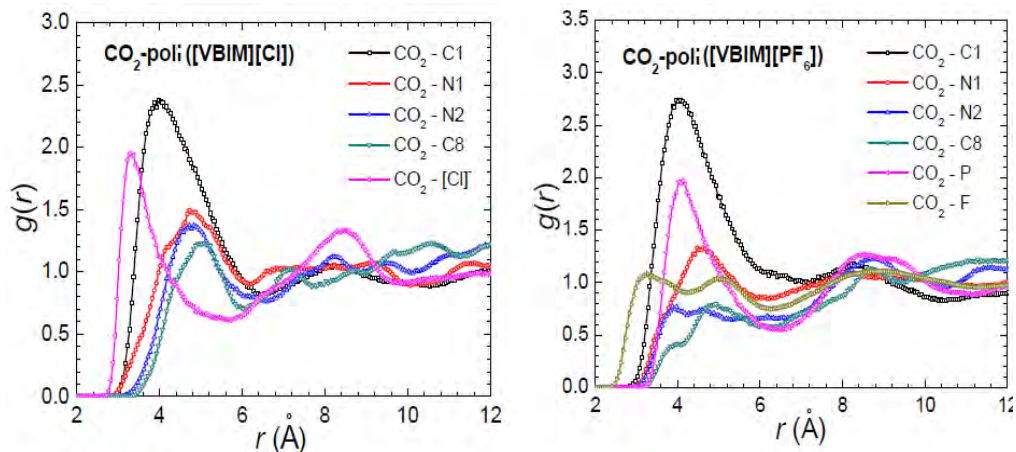


Figura 3.9. Funções de distribuição radial em poli [VBIM] [Cl] e poli [VBIM] [PF<sub>6</sub>]. Fonte: (FANG; LUO; JIANG, 2013)

Estudos sobre o efeito do tipo de cátion ou modificações na estrutura da cadeia lateral deste elemento, sobre a solubilidade do CO<sub>2</sub> têm sido relatados na literatura (TANG et al., 2005b, 2005c, 2005d, 2009, BARA et al., 2007, 2008; BARA JASON E. , HATAKEYAMA EVAN S. , GIN DOUGLAS L., 2008; SIMONS et al., 2010). Normalmente, PLIs contendo o cátion tetralquilamônio demonstram capacidade de sorção de CO<sub>2</sub> superior a PLIs com o cátion imidazólio (TANG et al., 2005b). Este fato possivelmente está relacionado à alta densidade de carga positiva do cátion tetralquilamônio que resulta em uma forte interação com o CO<sub>2</sub>, em contraste a carga positiva deslocalizada do cátion imidazólio (TANG et al., 2005b).

Testes experimentais de sorção de CO<sub>2</sub> realizados em PLIs base estireno, contendo o ânion BF<sub>4</sub> e diferentes cátions (Amônio, Piridínio, Fosfônio, Imidazólio) também verificaram este comportamento, visto que a sorção de CO<sub>2</sub> diminuiu na seguinte ordem: Amônio > Piridínio > Fosfônio > Imidazólio (TANG et al., 2009). Além disso, assim como observado em LIs convencionais, a solubilidade de CO<sub>2</sub> é incrementada com o aumento da pressão (TANG et al., 2009).

A capacidade de sorção do CO<sub>2</sub> pode ser fortemente afetada por modificações na cadeia alquílica do cátion (TANG et al., 2005c, 2005d, 2009; QIAN; TEXTER; YAN, 2017). O aumento do comprimento da cadeia alquílica do cátion pode representar um obstáculo, devido ao surgimento do efeito estérico que dificulta a sorção do CO<sub>2</sub> (TANG et al., 2005c, 2009; QIAN; TEXTER; YAN, 2017).

Estudos mostraram que a introdução de cadeias longas no cátion amônio ou imidazólio comprometem a capacidade de sorção do CO<sub>2</sub> (TANG et al., 2005c, 2005d). A substituição do grupo metil por etil e butil no cátion amônio reduziu a capacidade de sorção de CO<sub>2</sub> em cerca de 53% e 70 %, respectivamente (TANG et al., 2005d). Comportamento similar foi encontrado para o cátion imidazólio, no qual a troca do grupo metil por butil provocou uma redução na solubilidade do CO<sub>2</sub> de aproximadamente 26 % (TANG et al., 2005c).

Em contrapartida, o aumento do comprimento da cadeia alquílica ou a introdução de substituintes, ramificados, cíclicos ou polares no cátion pode promover

melhorias na seletividade e permeabilidade ao  $\text{CO}_2$  em correntes gasosas de  $\text{CO}_2/\text{N}_2$  e  $\text{CO}_2/\text{CH}_4$  (BARA et al., 2007, 2008; HORNE et al., 2014). Bara et al., 2007 (BARA et al., 2007) avaliaram o efeito do incremento da cadeia alquílica no cátion imidazólio de PLIs base estireno e acrilato sobre a permeabilidade de  $\text{CO}_2$  (Figura 3.10).

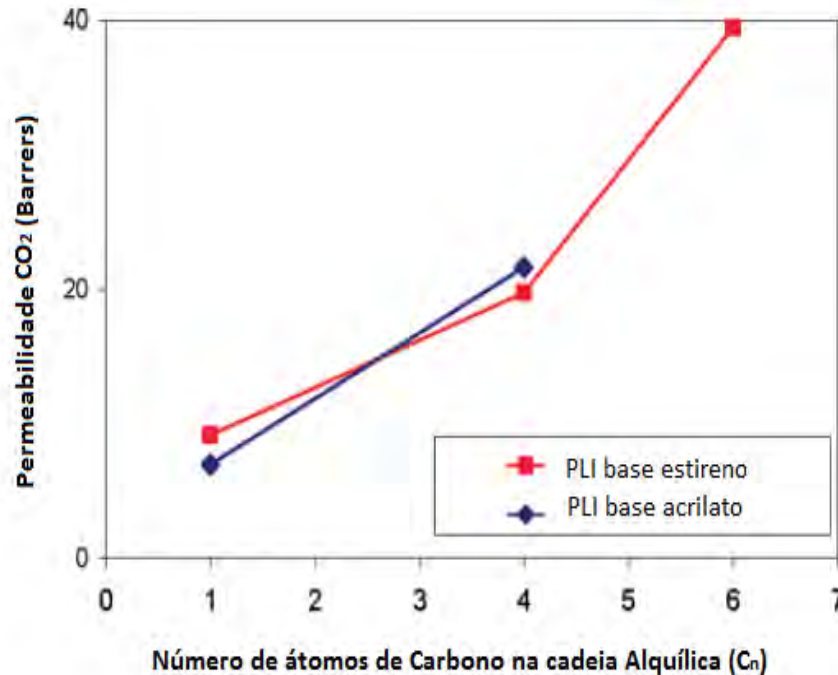


Figura 3.10. Tendências da permeabilidade de  $\text{CO}_2$  em PLIs base estireno e acrilato. Barras de erro dentro de símbolos. Fonte: Bara et al., 2007

De acordo com os resultados apresentados na Figura 3.10 observar-se que o aumento do comprimento da cadeia alquílica do cátion tende a elevar drasticamente a permeabilidade ao  $\text{CO}_2$ . Este comportamento pode ser consequência do aumento do volume livre na rede polimérica, devido a um empacotamento ineficiente promovido pelas cadeias laterais alquílicas (BARA et al., 2007).

O efeito dos substituintes alquil na cadeia lateral do cátion imidazólio, sobre a permeabilidade e seletividade ao  $\text{CO}_2$  em uma mistura de gases  $\text{CO}_2/\text{CH}_4$  (50/50 vol.%), foi investigado por Simons et al., 2010 (SIMONS et al., 2010). A Figura 3.11 mostra a permeabilidade e seletividade ao  $\text{CO}_2$  em função da pressão para os PLIs sintetizados com o ânion bis(trifluorometilsulfonil)imida ( $\text{TF}_2\text{N}$ ) e diferentes substituintes alquila na cadeia lateral do cátion. C1 refere-se ao PLI com o grupo metil como substituinte, C4 ao PLI com o n-butil e C6 ao n-hexil.

Simons et al., 2010 observaram um incremento na permeabilidade dos PLIs com o aumento pressão parcial de CO<sub>2</sub> (Figura 3.11a). Em todas as pressões testadas, o substituinte mais longo (C6) apresentou maior permeabilidade ao CO<sub>2</sub>, enquanto que o substituinte com cadeia mais curta (C1) demonstrou permeabilidade menor, devido a interações iônicas mais fortes dentro do material. O PLI contendo os grupos n-butil (C4) e n-hexil (C6) apresentaram comportamento semelhante e permeabilidades comparáveis (Figura 3.11a).

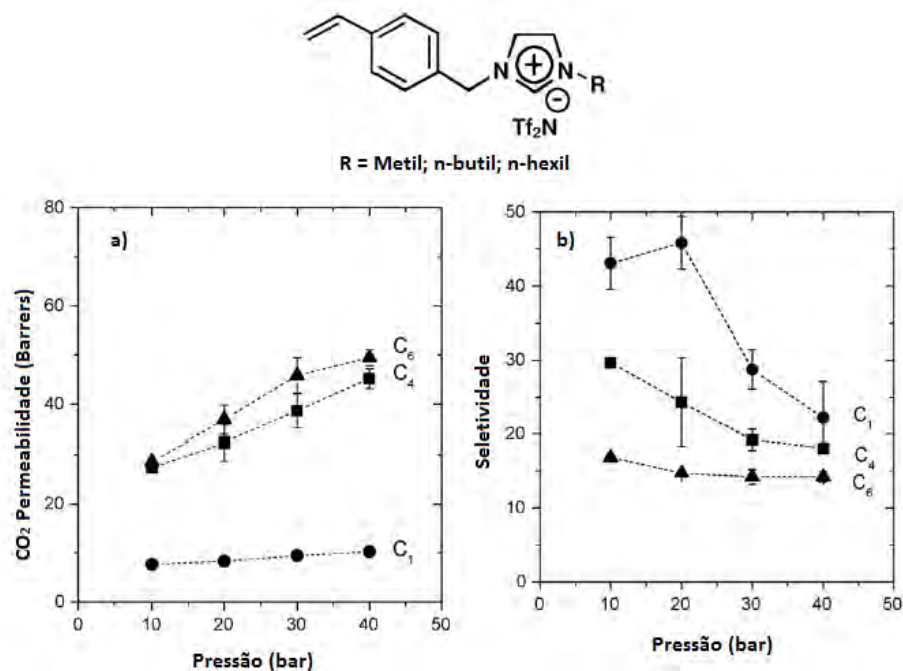


Figura 3.11. a) Permeabilidade ao CO<sub>2</sub> b) seletividade CO<sub>2</sub>/CH<sub>4</sub>, em função da Pressão. Condições: alimentação CO<sub>2</sub>/CH<sub>4</sub> (50/50 vol.%) e temperatura de 20 °C. Adaptação Simons (SIMONS et al., 2010).

A Figura 3.11b mostra também que a seletividade depende fortemente do comprimento da cadeia lateral do cátion, além de sofrer influência da pressão de operação. Observa-se que a seletividade dos PLIs sintetizados com cadeias laterais mais longas (C4 e C6) reduz com o acréscimo da pressão. Isto pode estar relacionado com o fato de que o CO<sub>2</sub> interage preferencialmente com as cargas iônicas, ao contrário do CH<sub>4</sub> que possui grande afinidade pelas cadeias alquílicas, devido à maior hidrofobicidade, aliado ao fato de que substituintes mais longos possuem maior propensão ao inchamento (SIMONS et al., 2010).

O PLI sintetizado com uma cadeia lateral curta (C1) exibiu comportamento divergente dos demais PLIs (Figura 3.11b). Em pressões de até 20 bar, verificou-se um aumento da seletividade proporcionada pela restrição do inchamento do polímero com a incorporação de uma cadeia mais curta. Porém, em pressões superiores uma forte queda na seletividade foi notada, em consequência do aumento do inchamento e da permeabilidade ao CH<sub>4</sub> em relação ao CO<sub>2</sub> (SIMONS et al., 2010).

A introdução de grupos polares como, por exemplo -OR ou -CN na cadeia lateral do cátion foi investigada pelo grupo de pesquisa de Noble (BARA et al., 2007, 2008; BARA JASON E. , HATAKEYAMA EVAN S. , GIN DOUGLAS L., 2008). A Figura 3.12 ilustra as estruturas dos PLIs estudados e a Tabela 3.1 mostra os resultados de permeabilidade e seletividade encontrados para estes PLIs.

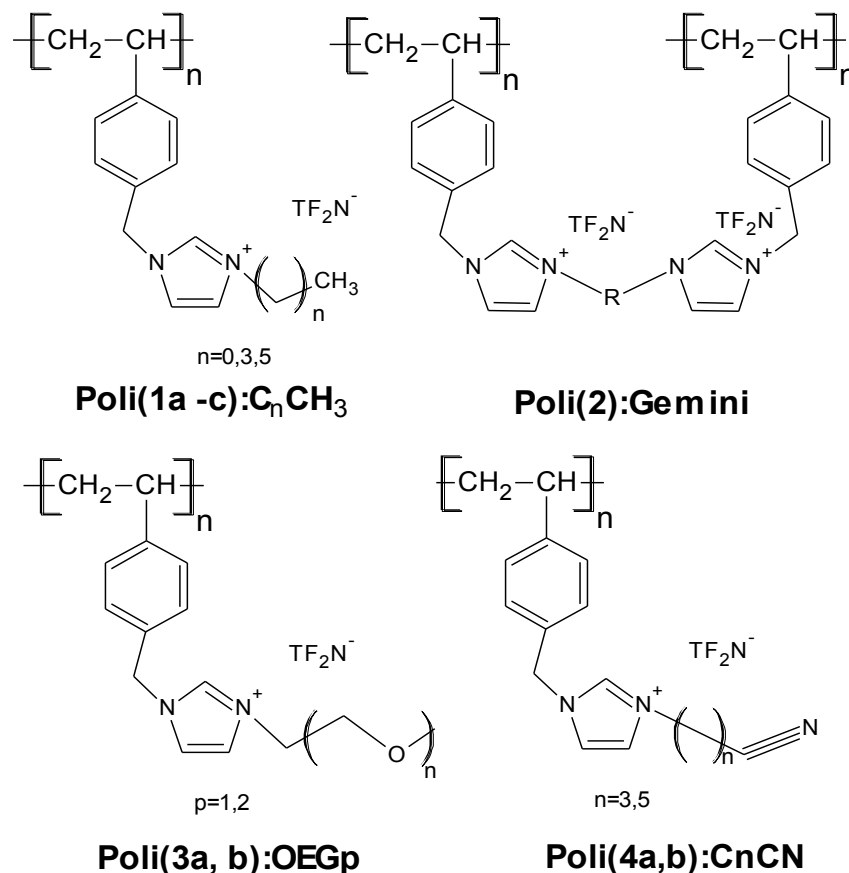


Figura 3.12: Estrutura das diferentes classes de PLIs sintetizadas pelo grupo de Noble Fonte: Bara et al., 2008 (BARA JASON E. , HATAKEYAMA EVAN S. , GIN DOUGLAS L., 2008).

Membranas de PLIs contendo grupos funcionais éter (R-O-R') na cadeia lateral do cátion imidazólio exibiram elevada permeabilidade ao CO<sub>2</sub> em comparação a PLIs que contém o grupo nitrila (-CN) e substituintes alquila. Além disso, a inserção de grupos funcionais éter também proporcionou valores de seletividade superiores ou similares aos alcançados com o aumento do comprimento da cadeia alquílica ou a introdução de grupos do tipo nitrila (-CN) (Tabela 3.1). Estes resultados demonstram que a inserção de grupos éter em PLIs pode ser uma abordagem interessante para melhorar a permeabilidade e seletividade destes materiais em processos de separação de CO<sub>2</sub>.

Tabela 3.1. Valores de permeabilidade ao CO<sub>2</sub> e seletividade (CO<sub>2</sub>/N<sub>2</sub>; CO<sub>2</sub>/CH<sub>4</sub>) obtidas para as classes de PLIs estudadas pelo grupo de Noble. Fonte: Bara et al., 2008 (BARA JASON E. , HATAKEYAMA EVAN S. , GIN DOUGLAS L., 2008).

Membranas	Permeabilidade ao CO <sub>2</sub> *	CO <sub>2</sub> /N <sub>2</sub>	CO <sub>2</sub> /CH <sub>4</sub>
Poli(1a -c):C <sub>n</sub> CH <sub>3</sub>	9-32	28-32	17-39
Poli(2):Gemini	4	22-28	27-32
Poli(3a, b):OEG <sub>p</sub>	16-22	41-44	29-33
Poli(4a,b):C <sub>n</sub> CN	4-8	37-40	30-37

\* Permeabilidade ao CO<sub>2</sub> em Barrers. 1 Barrers = 10<sup>-10</sup> cm<sup>3</sup>(STP)cm/cm<sup>2</sup> s cm Hg.

Apesar da natureza do ânion exercer papel secundário sobre a afinidade do CO<sub>2</sub> com PLIs catiônicos, conforme mostrado no estudo de Fang et al., 2013 (Figura 3.8 e 3.9), a escolha do tipo de ânion é um fator que não pode ser desconsiderado na busca de PLIs catiônicos eficientes para separação de CO<sub>2</sub> (TANG et al., 2005d; CARLISLE et al., 2010; BHAVSAR; KUMBHARKAR; KHARUL, 2012; ZULFIQAR; SARWAR; MECERREYES, 2015b; SADEGHPOUR; YUSOFF; AROUA, 2016). O uso de vários tipos de contra ânions inorgânicos (Br<sup>-</sup>, BF<sub>4</sub><sup>-</sup>, PF<sub>6</sub><sup>-</sup>, Tf<sub>2</sub>N<sup>-</sup> etc.), e orgânicos (Sac<sup>-</sup>, Bz<sup>-</sup>, Ac<sup>-</sup> etc.) tem sido relatados na literatura para obtenção de PLIs catiônicos (TANG et al., 2005a, 2005b, 2005d, 2009; CARLISLE et al., 2010; BHAVSAR; KUMBHARKAR; KHARUL, 2012).

Estudos realizados com policátions de base amônio e imidazólio mostraram que diferentemente do observado em líquidos iônicos convencionais, a presença de

átomos fluorados em seus ânions não promove necessariamente um impacto direto na melhora da sorção de  $\text{CO}_2$  (AKI et al., 2004; CADENA et al., 2004; TANG et al., 2005c, 2005d, 2009). Este comportamento pode ser evidenciado pelos resultados de sorção de  $\text{CO}_2$  (Figura 3.13) obtidos para PLIs de base amônio, sintetizados com diferentes tipos de contra ânions inorgânicos fluorados ( $\text{PF}_6^-$ ,  $\text{BF}_4^-$ , e  $\text{Tf}_2\text{N}^-$ ) e com o ânion orgânico não fluorado  $\text{Sac}^-$ .

Conforme pode-se observar na Figura 3.13, a capacidade de sorção de  $\text{CO}_2$  obtida para o PLI contendo o ânion  $\text{TF}_2\text{N}^-$  (2,85 mol%) foi muito menor que a encontrada para os PLIs com ânion  $\text{PF}_6^-$  (10,66 mol%), e  $\text{BF}_4^-$  (10,22 mol%) e semelhante a alcançada com o ânion não fluorado  $\text{Sac}$  (2,67 mol%) (TANG et al., 2005d).

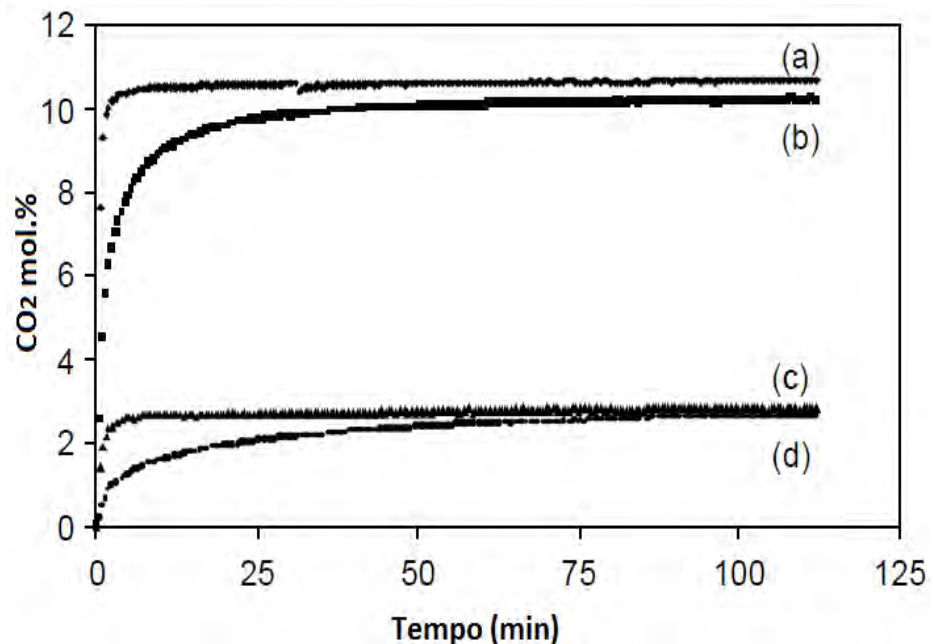


Figura 3.13. Isoterma de sorção de  $\text{CO}_2$  em PLIs base amônio, sintetizados com diferentes ânions ( $22^\circ\text{C}$ ;  $0,79\text{ bar}$ ): (a) P[VBTMA][ $\text{PF}_6^-$ ]; (b) P[VBTMA][ $\text{BF}_4^-$ ]; (c) P[VBTMA][ $\text{TF}_2\text{N}^-$ ]; (d) P[VBTMA][ $\text{Sac}^-$ ].

Fonte: Tang et al., 2005 (TANG et al., 2005d).

Uma série de policátions a base de dialildimetilamônio e amônio com diferentes ânions carboxilatos ( $\text{Ac}^-$ ,  $\text{TFAc}^-$ ,  $\text{HFB}^-$ ,  $\text{BZ}^-$ ), sulfonatos ( $\text{MS}^-$ ,  $\text{PTS}^-$ ,  $\text{TFMS}^-$ ), sulfonamidas ( $\text{TF}_2\text{N}^-$ ), e inorgânicos ( $\text{Cl}^-$ ,  $\text{NO}_3^-$ ,  $\text{BF}_4^-$ ) ilustrados na Figura 3.14, foram investigados quanto a sua capacidade de sorção de  $\text{CO}_2$  e seletividade (BHAVSAR; KUMBHARKAR; KHARUL, 2012). Os valores de coeficiente de solubilidade ( $\text{SCO}_2$ )

e seletividade ( $\text{SCO}_2/\text{SH}_2$ ;  $\text{SCO}_2/\text{SN}_2$ ) obtidos a 20 atm e 35°C podem ser vistos na Tabela 3.2.

Ao analisarmos o PLI P[DADMA] na Tabela 3.2 fica evidente que o ânion carboxilato AC proporcionou o maior coeficiente de solubilidade ( $\text{SCO}_2$ ) (4,0) e seletividade de  $\text{CO}_2$  em relação ao  $\text{H}_2$  ( $\text{SCO}_2/\text{SH}_2$  de 42,6) e  $\text{N}_2$  ( $\text{SCO}_2/\text{SN}_2$  de 114,3)(BHAVSAR; KUMBHARKAR; KHARUL, 2012). Observa-se ainda que o coeficiente de solubilidade ( $\text{SCO}_2$ ) obtido para os PLIs com ânions carboxilatos aumenta na seguinte ordem  $[\text{TFAc}] < [\text{HFB}] < [\text{Bz}] < [\text{Ac}]$  de acordo com a tendência de incremento da sua basicidade ( $\text{pK}_a$  do ácido conjugado = 4,8; 4,2; 0,4; 0,0 respectivamente). Este comportamento é semelhante ao observado em líquidos iônicos convencionais, onde o aumento da basicidade do ânion tende a elevar a sorção de  $\text{CO}_2$  (BHAVSAR; KUMBHARKAR; KHARUL, 2012).

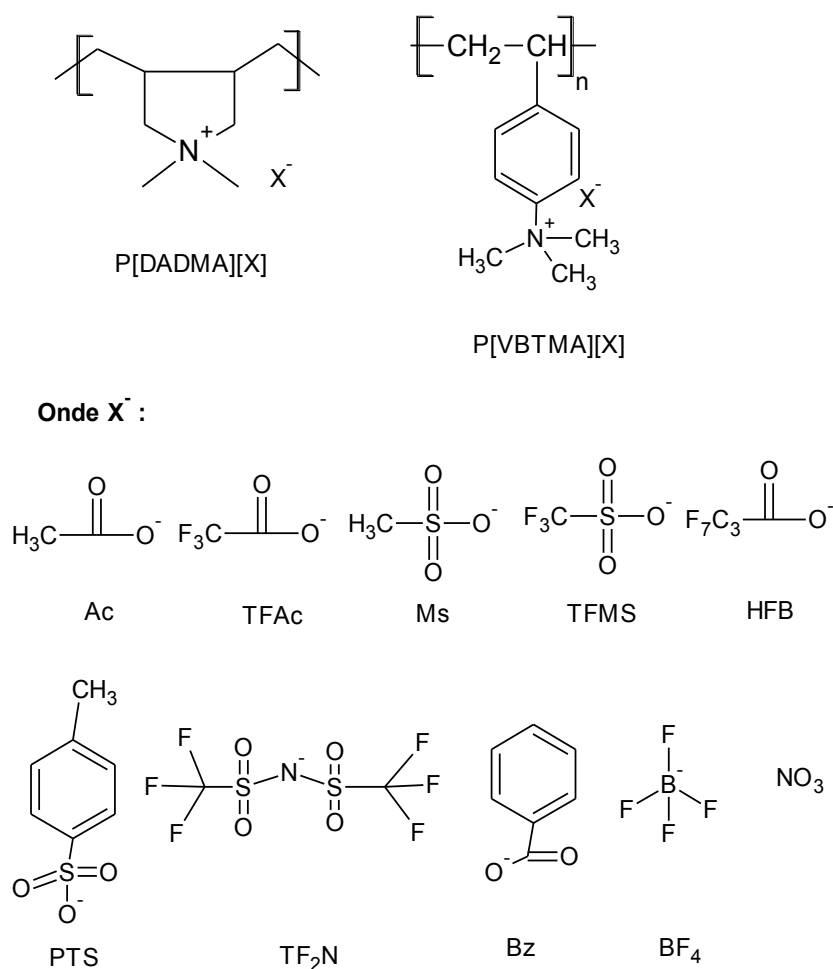


Figura 3.14. Estrutura de policátions e ânions utilizados no estudo de sorção e seletividade. Fonte: Bhavsar et al.,2012 (BHAVSAR; KUMBHARKAR; KHARUL, 2012).



No trabalho desenvolvido por Bhavsar et al., 2012 (BHAVSAR; KUMBHARKAR; KHARUL, 2012) verificou-se também que para PLIs com ânions sulfonados a basicidade do ânion não pode ser considerada como a única responsável pela melhora da seletividade e da capacidade de sorção de  $\text{CO}_2$ , outras propriedades como, por exemplo, a densidade e volume livre fracionário (VLF) também podem influenciar. O  $\text{SCO}_2$  seguiu a seguinte variação  $\text{TFMS}^- > \text{PTS}^- \approx \text{MS}^-$  e os valores de basicidade (pKa do ácido conjugado) foram:  $\text{TFMS}^- = -13$ ,  $\text{PTS}^- = -2,8$ ,  $\text{MS}^- = -2,0$  (BHAVSAR; KUMBHARKAR; KHARUL, 2012). Em ânions inorgânicos tais como  $\text{Cl}^-$ ,  $\text{NO}_3^-$ ,  $\text{BF}_4^-$  o aumento da massa molar do ânion pode contribuir para o incremento no  $\text{SCO}_2$ ,  $\text{SCO}_2/\text{SH}_2$  e  $\text{SCO}_2/\text{SN}_2$  (ver Tabela 3.2) (BHAVSAR; KUMBHARKAR; KHARUL, 2012)

Tabela 3.2. Coeficiente de solubilidade ( $\text{SCO}_2$ ) e seletividade ( $\text{SCO}_2/\text{SH}_2$ ;  $\text{SCO}_2/\text{SN}_2$ ) a 20 atm, expresso em  $\text{cm}^3(\text{STP})/\text{cm}^3$  polímero. Fonte: Bhavsar et al., 2012 (BHAVSAR; KUMBHARKAR; KHARUL, 2012).

PLIs	$\text{SCO}_2$	$\text{SCO}_2/\text{SH}_2$	$\text{SCO}_2/\text{SN}_2$
P[DADMA][Cl]	0,12	1,6	2,3
P[DADMA][Ac]	4,00	42,6	114,3
P[DADMA][TFAc]	0,63	2,7	3,5
P[DADMA][MS]	0,25	2,4	1,4
P[DADMA][TFMS]	0,96	2,4	2,5
P[DADMA][PTS]	0,25	3,3	7,6
P[DADMA][HFB]	1,03	18,7	11,8
P[DADMA][Bz]	1,37	19,7	41,5
P[DADMA][ $\text{NO}_3$ ]	0,42	3,6	4,8
P[DADMA][ $\text{TF}_2\text{N}$ ]	0,73	8,2	12,8
P[DADMA][ $\text{BF}_4$ ]	0,86	10,8	11,5
P[VBTMA][Cl]	0,67	-	4,5
P[VBTMA][Ac]	3,16	-	37,6
P[VBTMA][ $\text{BF}_4$ ]	1,45	-	14,6

### 3.3.2. Poli (líquidos iônicos) obtidos via polimerização por condensação/modificação

Recentemente tem aumentado o interesse de pesquisadores pela obtenção de PLIs via polimerização por condensação/modificação (SHAPLOV et al., 2016). Este método envolve basicamente reações de condensação, que podem ser conduzidas por meio da modificação de polímeros comerciais ou via condensação do respectivo monômero (BHAVSAR et al., 2014; SHAPLOV et al., 2016; TOME; MARRUCHO, 2016). Entretanto, na literatura, poucas são as classes de PLIs conhecidas sintetizadas por esta via para captura de CO<sub>2</sub>: poliuretanos (PU) (MAGALHAES et al., 2014; FERNÁNDEZ et al., 2016), polibenzimidazol (PBI) (BHAVSAR et al., 2014; KUMBHARKAR; BHAVSAR; KHARUL, 2014), poliimidazóis (SHAPLOV et al., 2016) e a base de imidazólio contendo grupos éster e hidroxilas simétricas (XIONG et al., 2012).

A influência do ânion sobre a cinética e a sorção de CO<sub>2</sub> foi investigada em PLIs catiônicos base imidazólio ilustrados na Figura 3.14 (XIONG et al., 2012). O ânion PF<sub>6</sub> proporcionou uma capacidade de sorção superior quando comparada ao ânion BF<sub>4</sub> (Figura 3.15), possivelmente isto está relacionado à morfologia rugosa do PLI-PF<sub>6</sub> que pode promover uma área superficial maior e conseqüentemente uma sorção de CO<sub>2</sub> mais rápida e elevada (XIONG et al., 2012). Observa-se ainda na Figura 3.15, que, embora o tempo necessário para atingir o equilíbrio de sorção com os PLIs tenha sido inferior apenas ao alcançado com os LIs [bmim] [PF<sub>6</sub>] (a) e [EEIM][PF<sub>6</sub>] (c), a capacidade de sorção destes PLIs foi superior a uma série de LIs (XIONG et al., 2012).

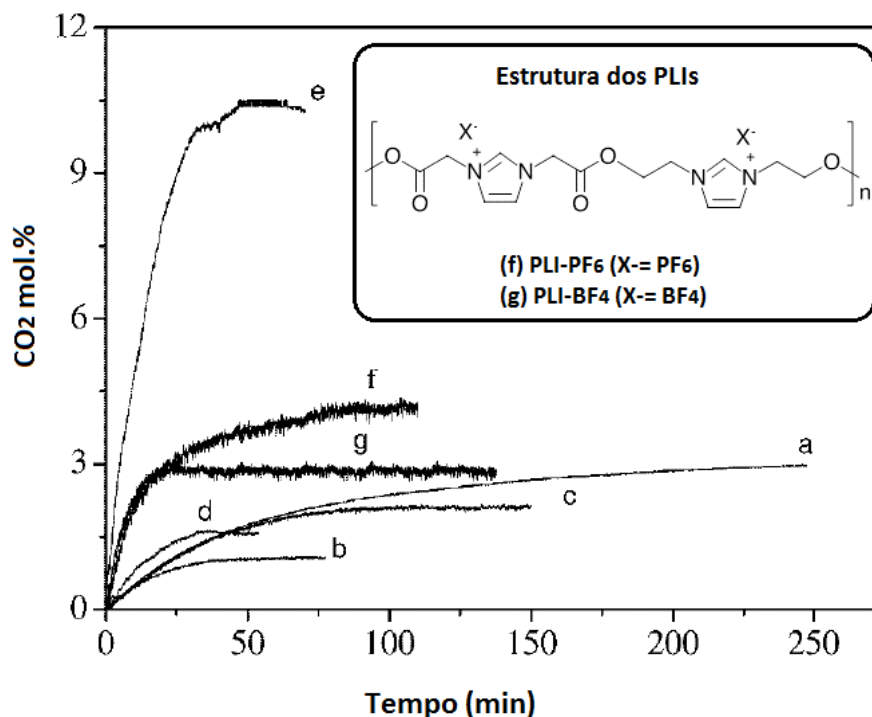


Figura 3.15. Cinética de sorção de CO<sub>2</sub> dos LIs e PLIs (25°C 648 mmHg): (a) [bmim] [PF<sub>6</sub>]; (b) [EEIM][BF<sub>4</sub>]; (c) [EEIM][PF<sub>6</sub>]; (d)[HHIM][BF<sub>4</sub>]; (e)[HHIM][PF<sub>6</sub>]; (e) PLI-PF<sub>6</sub> e (f) PLI-BF<sub>4</sub>. Fonte: Xiong, 2012 (XIONG et al., 2012)

O efeito do ânion e a estrutura do cátion sobre a solubilidade, seletividade e permeabilidade ao CO<sub>2</sub> também tem sido alvo de estudos em PLIs catiônicos baseados em polibenzimidazol (Figura 3.16) (BHAVSAR et al., 2014; KUMBHARKAR; BHAVSAR; KHARUL, 2014). Os coeficientes de solubilidade de CO<sub>2</sub> [S(CO<sub>2</sub>)] e os resultados de permeabilidade e seletividades dos PLIs [TMPBI] [X] evidenciaram que o ânion possuiu um forte efeito sobre as propriedades de sorção de CO<sub>2</sub>. A inserção do ânion TF<sub>2</sub>N no PLI [TMPBI] [X] propiciou uma permeabilidade ao CO<sub>2</sub> mais elevada quando comparado ao BF<sub>4</sub>. De acordo com Kumbharkar et al., 2014 (KUMBHARKAR; BHAVSAR; KHARUL, 2014) o ânion TF<sub>2</sub>N proporciona maior volume livre na rede polimérica, devido a sua elevada mobilidade em comparação aos outros ânions (Ac<sup>-</sup>, BF<sub>4</sub><sup>-</sup> e I<sup>-</sup>), o que resulta em uma melhora na difusão do gás.

Estudos desenvolvidos por Bhavsar et al 2014 (BHAVSAR et al., 2014) indicaram que modificações na estrutura do cátion também podem impactar na

solubilidade do  $\text{CO}_2$ , visto que a sorção de [TMPBI-BuI]  $[\text{BF}_4]$  foi superior a alcançada com os PLIs [TMPBI-I]  $[\text{BF}_4]$  e [DMABPBI]  $[\text{BF}_4]$  (Figura 3.16). Neste estudo, a forte influência do ânion sobre a sorção de  $\text{CO}_2$  também foi observada. A troca do ânion iodeto (I) por tetrafluoroborato ( $\text{BF}_4$ ) no PLIs [TMPBI-Iso] [X] e [DMABPBI] [X] promoveu o aumento da sorção de  $\text{CO}_2$  (BHAVSAR et al., 2014).

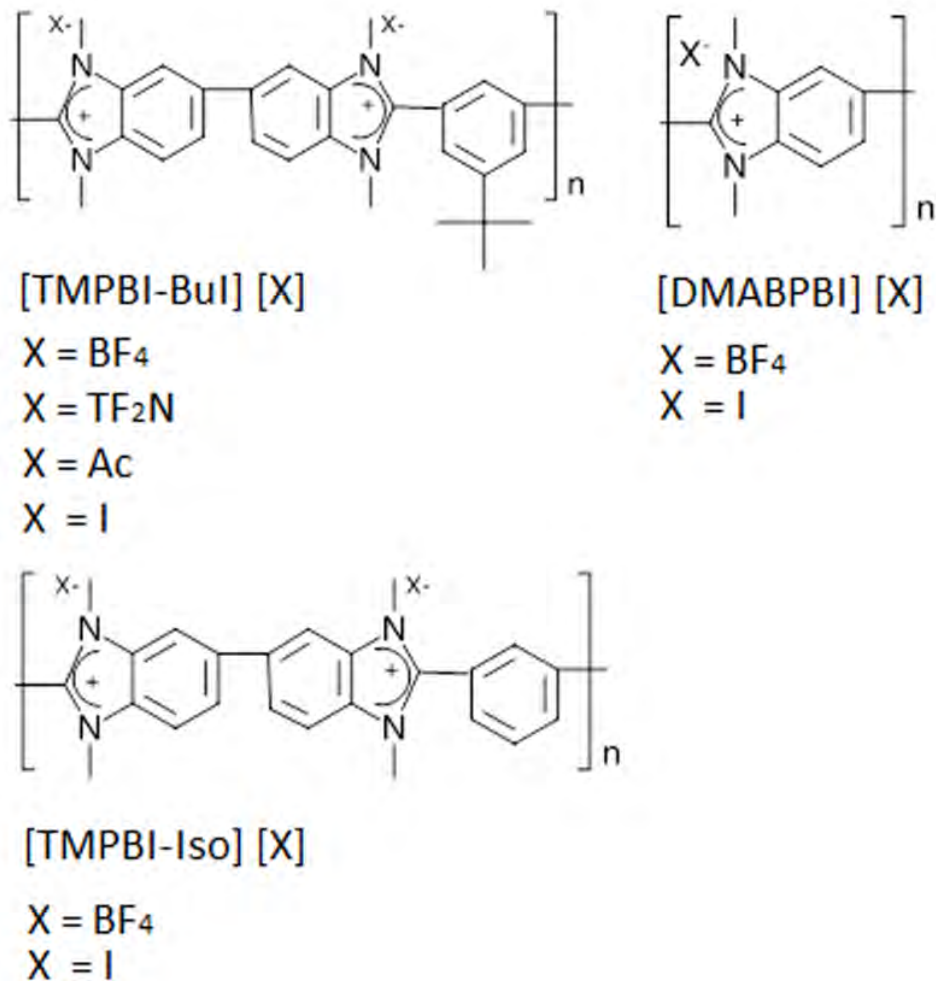


Figura 3.16. PLIs catiônicos baseados em polibenzimidazol. Fonte: Bhavsar et al., 2014 (BHAVSAR et al., 2014).

As isotermas de sorção de  $\text{CO}_2$  obtidas para o PLIs base [TMPBI-BuI] [X] são apresentadas na Figura 3.17. Os resultados mostram que o aumento da sorção de  $\text{CO}_2$  destes PLIs seguiu a ordem crescente de basicidade do ânion ( $\text{pK}_a$  valor dos seus ácidos conjugados  $[\text{Tf}_2\text{N}] -4 < [\text{BF}_4] -0,44 < [\text{Ac}] 4,75$ ) e que todos os PLIs [TMPBI] [X] apresentaram desempenho de sorção superior (cerca de duas a três

vezes maior) ao de membranas convencionais utilizadas, tais como a polissulfona (PSF) e policarbonato (PC)(KUMBHARKAR; BHAVSAR; KHARUL, 2014).

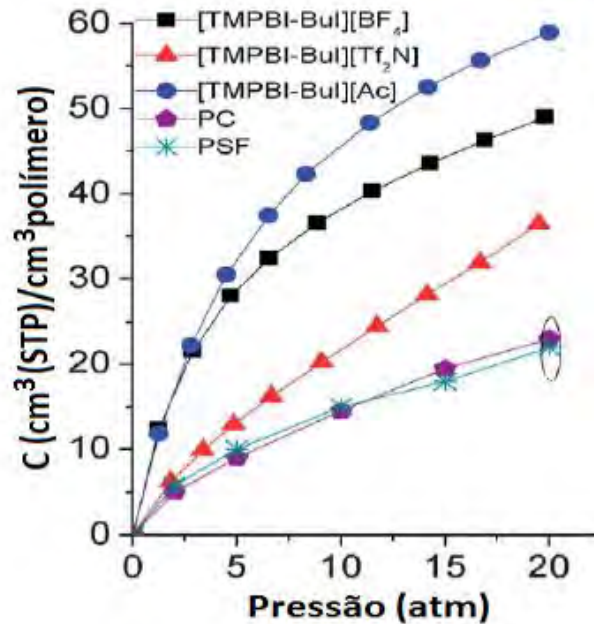


Figura 3.17. Isoterma de sorção de CO<sub>2</sub> a 35°C obtida para o PLI [TMPBI] [X] com diferentes ânions em comparação ao polímeros comerciais (polissulfona (PSF) e policarbonato (PC)).Fonte: Kumbharkar et al., 2014 (KUMBHARKAR; BHAVSAR; KHARUL, 2014)

O efeito da modificação estrutural da cadeia do cátion em PLIs catiônicos base poliimidazoles sobre a separação do CO<sub>2</sub> foi estudado. Foram testados a introdução de dois diferentes grupos (benzimidazole e azabicyclo[2.2.2]octano) no cátion. Membranas que continham grupos azabicyclo[2.2.2]octano demonstraram permeabilidade mais elevada ao CO<sub>2</sub>. O incremento na permeabilidade com a presença do grupo azabicyclo[2.2.2]octano, pode estar associado ao fato deste ser mais volumoso que o benzimidazol, contribuindo assim para o desenvolvimento de uma membrana menos empacotada, com menor resistência à difusão de gás e consecutivamente maior permeabilidade aos gases (SHAPLOV et al., 2016).

A síntese e avaliação do desempenho de sorção de CO<sub>2</sub> de PLIs catiônicos e aniônicos base poliuretano também tem sido reportada na literatura (MAGALHAES et al., 2014; FERNÁNDEZ et al., 2016). Estes estudos mostraram que PLIs aniônicos apresentam capacidade de sorção de CO<sub>2</sub> mais elevada em comparação aos

catiônicos (MAGALHAES et al., 2014). O acréscimo da quantidade de grupos éter através do aumento da massa molar de 1000 para 2000 do polieter poli (tetrametileno glicol) (PTMG), empregado na síntese do PLI proporcionou uma elevação nos valores de sorção de CO<sub>2</sub> a 20 bar em cerca de 71%. Grupos éter possuem uma forte afinidade com o CO<sub>2</sub> e o aumento da distância entre os sítios iônicos pode promover a melhora na interação com o CO<sub>2</sub> e consecutivamente o aumento da solubilidade de CO<sub>2</sub> (MAGALHAES et al., 2014).

Ao contrário do esperado, a introdução de estruturas aromáticas no ânion do PLI aniônico através do uso do diisocianato m-tetrametilxileno diisocianato (TMXDI) ao invés de 1,6-hexametileno (HDI) não promoveu uma melhora significativa na solubilidade do CO<sub>2</sub>. Em baixas pressões (5 bar), o PLI sintetizado com HDI apresentou uma capacidade de sorção 19% maior que a obtida com TMXDI na estrutura, em pressões mais elevadas (20 bar) a capacidade de sorção de CO<sub>2</sub> do PLI contendo a estrutura aromática foi levemente maior (cerca de 2%) que a atingida com a estrutura linear. A análise deste comportamento sugere que em baixas pressões, grupos aromáticos podem dificultar a interação do CO<sub>2</sub> com os grupos polares / iônicos da cadeia polimérica (MAGALHAES et al., 2014).

O poliânion sintetizado com PTMG de massa molar 2000 e HDI teve seus valores de sorção de CO<sub>2</sub> comparados com o PLI [VBTMA][BF<sub>4</sub>] sintetizado via polimerização radicalar por Tang et al., 2005 (TANG et al., 2005b), solução aquosa de metildietanolamina (MDEA) 50 wt% e com o LI 1.butil-3-metilimidazólio bis(trifluorometil)-sulfonamida [bmim] [TF<sub>2</sub>N]. Em 20 bar e 25°C, a capacidade de sorção de CO<sub>2</sub> atingida com o [bmim] [TF<sub>2</sub>N] e MDEA foi de apenas 56% e 51% respectivamente, da alcançada com este PLI aniônico. A capacidade de sorção de CO<sub>2</sub> deste PLI aniônico também foi maior que a alcançada com o PLI [VBTMA][BF<sub>4</sub>] (cerca de 77% maior)(MAGALHAES et al., 2014).

Normalmente, em LIs tradicionais a introdução de cadeias aquílicas ramificadas no cátion promove melhora na solubilidade de CO<sub>2</sub> (PENNLIN et al., 2008). No entanto, em PLIs aniônicos base PU esta tendência não foi observada, os resultados experimentais de sorção de CO<sub>2</sub> alcançados com o PLI de contra cátion

[bmim]<sup>+</sup>(linear) foram levemente superiores aos obtidos com PLI contendo o contração [dmbmim]<sup>+</sup>(ramificado). Este comportamento pode estar relacionado com o impedimento estérico causado pela ramificação (MAGALHAES et al., 2014).

Em outro trabalho (FERNÁNDEZ et al., 2016), extensores de cadeia (dietanolamina (DEA) hidrazina (HYD) e etilenodiamina (EDA)) foram usados com o objetivo de aumentar a massa molar de PLIs aniônicos base PU. A massa molar ponderal média (Mn) dos PLIs obtida variou de 2413 a 10121 g/mol. A técnica de emulsificação múltipla também foi empregada para formação de micropartículas (μP) e nanopartículas (NP) de PLIs.

O comparativo dos resultados de sorção de CO<sub>2</sub> entre os PLIs sintetizados por Rojas et al., 2016 (FERNÁNDEZ et al., 2016) evidenciaram que o aumento da massa molar pode favorecer a solubilidade do CO<sub>2</sub>, assim como a introdução de grupos amidas na estrutura do ânion. Verificou-se ainda que a capacidade de solubilizar o CO<sub>2</sub> baseia-se na estrutura, porém a morfologia e tamanho da partícula favorecem a sorção de CO<sub>2</sub>. A capacidade de sorção de CO<sub>2</sub> das μP e NP foram similares a dos PLIs, no entanto, a formação das partículas proporcionou um aumento significativo na taxa de sorção. Os PLIs necessitaram de aproximadamente 6,7h para atingir 80% da sua capacidade total de sorção de CO<sub>2</sub>. A baixa taxa de sorção pode estar relacionada com o fato de sua sorção ser principalmente superficial, em decorrência da sua morfologia lisa e não porosa (FERNÁNDEZ et al., 2016). Para a NP de PLI, o tempo fundamental para alcançar 82% do total de CO<sub>2</sub> capturado foi de apenas 50 min (4,936 mg/g a 25°C e 4 bar) (FERNÁNDEZ et al., 2016).

Poli (líquidos iônicos) aparecem como materiais promissores para captura de CO<sub>2</sub>. Esta revisão apresentou uma visão geral sobre as tecnologias de captura de CO<sub>2</sub> e abordou de forma ampla o desenvolvimento de PLIs para captura de CO<sub>2</sub>, explorando os principais aspectos da produção de PLIS, que afetam a capacidade de sorção, permeabilidade ou seletividade ao CO<sub>2</sub>. Em resumo, os estudos mostraram que a área específica porosidade, massa molar, aumento da pressão e redução da temperatura tendem a aumentar a sorção de CO<sub>2</sub>. Fatores como cátion,

ânion, introdução de grupos polares e modificações estruturais podem promover melhorias na sorção, seletividade e permeabilidade ao CO<sub>2</sub>. Ultimamente, o interesse por PLIs obtidos via polimerização por condensação/modificação tem aumentado. Este comportamento está relacionado com o fato desta rota permitir o uso de polímeros comerciais e o uso de uma quantidade menor de LIs no processo de síntese, reduzindo assim o custo da produção de PLIs.

Neste cenário, este trabalho sintetizou e avaliou a capacidade de sorção de CO<sub>2</sub> de diferentes poli (líquidos iônicos) baseados em poliuretano, poliuretano-imida e celulose. Também buscou investigar o potencial para captura de CO<sub>2</sub> de aminas e polieeteraminas suportadas em celulose.



## 4. PROCEDIMENTOS EXPERIMENTAIS E RESULTADOS

No presente trabalho foram sintetizados quatro tipos diferentes de poli(líquidos iônicos) baseados em polímeros sintéticos (PU e PU-imida) e na celulose, polímero natural extraído da casca de arroz. Todos os PLIs foram obtidos via polimerização por condensação/modificação, sendo de dois tipos (aniônico e catiônico) os produzidos a partir da celulose e apenas de um tipo (aniônico) os baseados em PU e PU-imida. Neste estudo, buscou-se correlacionar as propriedades mecânicas, térmicas, morfológicas e a capacidade de sorção de CO<sub>2</sub> com a estrutura de cada PLI produzido. Visando uma alternativa a estes PLIs, aminas e polieteraminas foram suportadas em celulose extraída da casca de arroz.

Nesta seção, em forma de artigos, serão apresentados os métodos, resultados e discussões pertinentes a esta tese, divididos da seguinte forma: Poli(líquidos iônicos) aniônicos base poliuretano (Capítulo I); Poli(líquidos iônicos) aniônicos base poliuretano-imida (capítulo II); Poli(líquidos iônicos) aniônicos celulósicos (capítulo III); Poli(líquidos iônicos) catiônicos celulósicos (capítulo IV) e aminas e polieteraminas suportadas em celulose (capítulo V). A Figura 4.1 mostra uma representação esquemática dos capítulos juntamente com os assuntos abordados. Na seção 5, será apresentado um comparativo dos principais resultados obtidos para os melhores materiais de cada grupo.

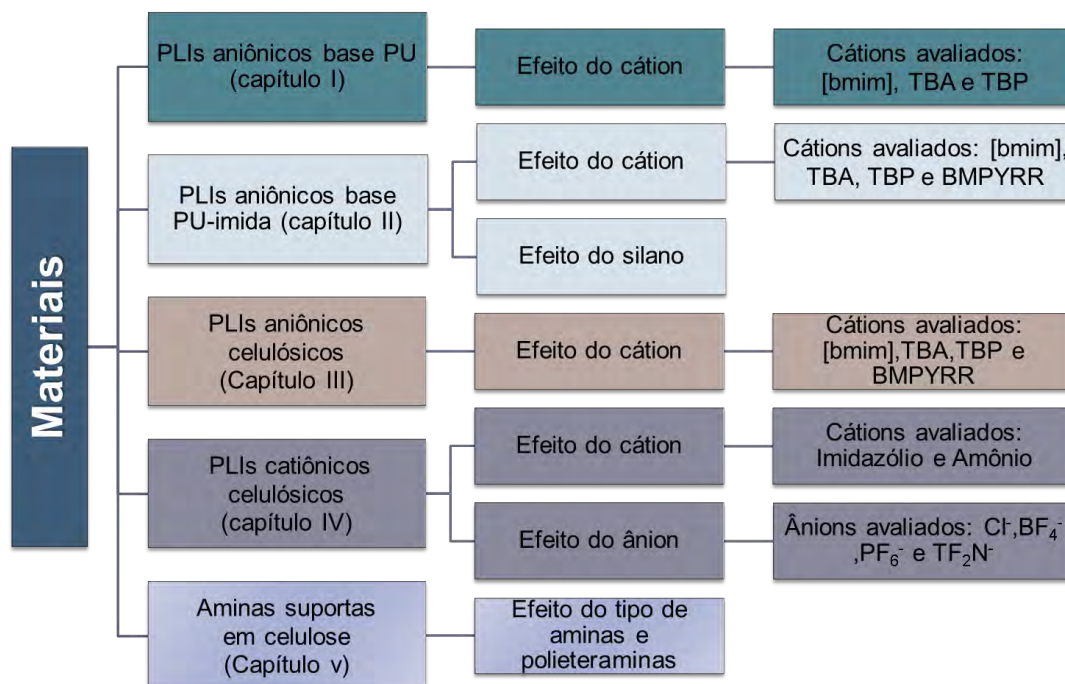


Figura 4.1. Representação esquemática dos capítulos e assuntos abordados.

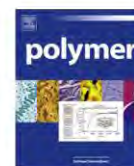
#### 4.1. Capítulo I: Poli (líquidos iônicos) aniônicos base poliuretano

Este capítulo descreve a síntese e caracterização de PLIs aniônicos base PU, por meio do artigo intitulado “*CO<sub>2</sub> capture: Tuning cation-anion interaction in urethane based poly(ionic liquids)*” publicado na Polymer. Três tipos diferentes de cátions (imidazólio, fosfônio e amônio) foram usados para formação de PLIs. Os materiais obtidos foram caracterizados por FTIR, RMN, DSC, TGA, DMA, MEV-FEG e AFM. A capacidade de sorção de CO<sub>2</sub> foi avaliada a 303,15 K e pressões de 0,8 até 50 bar, utilizando uma PTGA. Simulações ab initio foram empregadas com o objetivo de compreender o papel dos principais grupos funcionais presentes na estrutura dos PLIs sobre a afinidade ao CO<sub>2</sub>. Os resultados mostraram a influência do tipo de contra-cátion sobre as propriedades térmicas, mecânicas e capacidade de sorção de CO<sub>2</sub>. Ensaio experimentais e de simulação mostraram que cátions que coordenam fracamente o sítio COO<sup>-</sup> (TBA e TBP) favorecem a sorção de CO<sub>2</sub>. Os melhores desempenhos de sorção de CO<sub>2</sub> e propriedades mecânicas foram obtidos para os PLIs PU-TBP e PU-TBA (PU-TBP 15,7 mgCO<sub>2</sub>/g e PU-TBA 16,1 mgCO<sub>2</sub>/g a 30°C e 0,82 bar). Estes PLIs demonstraram capacidade de sorção de CO<sub>2</sub> superior quando comparada a outros PLIs descritos na literatura.



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## CO<sub>2</sub> capture: Tuning cation-anion interaction in urethane based poly(ionic liquids)



Franciele L. Bernard<sup>a, b</sup>, Barbara B. Polesso<sup>c</sup>, Fabiana W. Cobalchini<sup>c</sup>, Augusto J. Donato<sup>b</sup>, Marcus Seferin<sup>a, b</sup>, Rosane Ligabue<sup>a, b</sup>, Vitaly V. Chaban<sup>d</sup>, Jailton F. do Nascimento<sup>e</sup>, Felipe Dalla Vecchia<sup>c</sup>, Sandra Einloft<sup>a, b, \*</sup>

<sup>a</sup> Post-Graduation Program in Materials Engineering and Technology, Pontifical Catholic University of Rio Grande do Sul – PUCRS, Brazil

<sup>b</sup> School of Chemistry, Pontifical Catholic University of Rio Grande do Sul – PUCRS, Brazil

<sup>c</sup> School of Engineering, Pontifical Catholic University of Rio Grande do Sul – PUCRS, Av. Ipiranga, 6681, Partenon, Porto Alegre, CEP: 90619-900, Brazil

<sup>d</sup> Institute of Science and Technology (ICT), Federal University of São Paulo (UNIFESP), São José dos Campos, SP, Brazil

<sup>e</sup> Petrobras / CENPES, Ilha do Fundão Qd. 07, Rio de Janeiro, RJ, Brazil

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### ABSTRACT

The development of new urethane based poly(ionic liquids) (PILs) is a promising solution to address CO<sub>2</sub> capture. The obtention of low cost polyurethane based PILs using different ionic liquids cations fosters the emergence of new materials for CO<sub>2</sub> capture. The synthesized PILs were characterized by GPC, FTIR, NMR, DSC, TGA, DMTA and AFM. CO<sub>2</sub> sorption capacity was gravimetrically assessed in a magnetic suspension balance. In addition, *ab initio* simulations were performed. Experimental and simulation results allow understanding the varying performance of the polymeric anion with the bmim, TBA, and TBP cations in CO<sub>2</sub> sorption. A successful compound for gas capture must exhibit the weakest possible cation-anion coordination and smaller molecular masses. Better CO<sub>2</sub> sorption capacity was achieved for the cation TBA as compared to TBP and bmim (the amounts of CO<sub>2</sub> sorption at 50 bar and 303.15 K were 162 mgCO<sub>2</sub>/g for the PU-bmim, 168 mgCO<sub>2</sub>/g for PU-TBP and 226 mgCO<sub>2</sub>/g PU-TBA). At lower pressures, these compounds present a high CO<sub>2</sub> sorption capacity (at 0.82 bar of pressure and 303.15 K PU-bmim presented a CO<sub>2</sub> sorption of 9.4 mgCO<sub>2</sub>/g; PU-TBP 15.7 mgCO<sub>2</sub>/g and PU-TBA 16.1 mgCO<sub>2</sub>/g). The compounds PU-TBP and PU-TBA presented higher sorption values as compared to literature.

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### 1. Introduction

The growing atmospheric concentration of carbon dioxide (CO<sub>2</sub>) continues to be a global issue due to the common awareness about the connection of this gas with global warming problem [1,2]. Traditionally, aqueous solutions of alkanolamines are widely used for CO<sub>2</sub> removal in capture plants [3,4]. However, there are some disadvantages in commercial use of these amines solutions, including the high energy consumption during regeneration, formation of corrosive by-products, amines loss (requiring solvent replacement) and the need of large volume absorber [3–6].

The development of versatile sorbents with smart design for CO<sub>2</sub> capture has become one of the main research focus in this field

[2] and Poly(ionic liquid)s (PILs) appear as promising materials for CO<sub>2</sub> capture [2,7–9]. Poly(ionic liquids) or polymerized ionic liquids are a subclass of polyelectrolytes, in which each repeating unit is ionic. The units are connected through a polymeric backbone forming a macromolecular structure [8,10]. These compounds combine the benefits of ILs (high ionic conductivity, chemical and thermal stability) with the polymer properties (mechanical stability, processing and tunable macromolecular design) [8,11–13].

The PILs offer an extremely versatile and adjustable platform for designing sorbents, mainly for CO<sub>2</sub> separation from the flue gas (CO<sub>2</sub>/N<sub>2</sub>) and natural gas purification (CO<sub>2</sub>/CH<sub>4</sub>) [2,11].

It was described elsewhere [7] that polymers synthesized from the IL monomers present a higher sorption capacity as compared to room-temperature ILs (RTILs). It was also highlighted that the sorption/desorption of these polymers is higher than that of RTILs being completely reversible. A large number of studies have reported the solubility of CO<sub>2</sub> in different types of PILs [11,14–19]. Recently, we reported the synthesis of versatile and inexpensive

\* Corresponding author. Post-Graduation Program in Materials Engineering and Technology, Pontifical Catholic University of Rio Grande do Sul – PUCRS, Brazil.  
E-mail address: [einloft@pucrs.br](mailto:einloft@pucrs.br) (S. Einloft).

polyurethane based PILs designed for CO<sub>2</sub> capture [18]. Two different approaches were used for the synthesis of this material. The first one produces anionic PILs with the introduction of the imidazolium as a cation. The second one synthesizes cationic polyurethanes. The anionic PILs presented higher CO<sub>2</sub> sorption capacity as compared to the cationic polymers. It was also observed that PILs based in 1-butyl-3-methylimidazolium, [bmim]<sup>+</sup>, as a cation achieved higher CO<sub>2</sub> sorption than PIL with 1-dimethylbutyl-3-methylimidazolium, [dmbmim]<sup>+</sup> as a cation. This behavior can probably be due to the steric hindrance of the branched alkyl side chain of imidazolium structure compared with the linear structure. The anionic PILs also showed superior performance when compared with the IL [bmim] [Tf<sub>2</sub>N], MDEA 50% (w/w) as well as the PILs as described in literature [20]. The use of anionic PU chain to produce PILs allows an easy way to obtain different materials to be tested in CO<sub>2</sub> sorption. The CO<sub>2</sub> affinity by RTILs is influenced by the cation type. The same behavior must be expected for PILs.

The present work was undertaken on the same line, i.e., to develop new anionic polyurethanes introducing three different types of cations (imidazolium-, phosphonium- and ammonium-based cations). Quantum mechanical simulations were employed to understand the varying performance of the polymeric anion with the 1-butyl-3-methylimidazolium (bmim), tetrabutylammonium (TBA), and tetrabutylphosphonium (TBP) cations.

## 2. Experimental

### 2.1. Materials

Poly(tetramethylene ether) glycol (PTMG, Mn 2000 g/mol, Sigma-Aldrich), hexamethylene diisocyanate (HDI, 99%, Merck), dimethylol propionic acid (DMPA, 99%, Perstorp), dibutyl tin dilaurate (DBTDL, Miracemanuodex), N-methyl-2-pyrrolidone (NMP, 99.92%, Neon), methylethylketone (MEK, 99%, Mallinckrodt), tetrahydrofuran (THF, 99%, Vetec), potassium hydroxide (KOH, ≥85%, Sigma-Aldrich) were used as received. The tetrabutylammonium bromide (TBAB, 99%, Acros Organics), tetrabutylphosphonium bromide (TBPB, 98%, Sigma Aldrich), were dried under vacuum for 12 h at 60 °C. The ionic liquid 1-butyl-3-methylimidazolium chloride (bmim Cl) was synthesized following procedures well described in literature [21–23]. The ionic liquid (bmim Cl) was characterized by Fourier transform infrared spectroscopy (FTIR), using a Perkin-Elmer spectrophotometer model Spectrum 100 FT-IR with full attenuated reflectance model (ATR), as well as by proton nuclear magnetic resonance (<sup>1</sup>H NMR) on a Varian Spectrophotometer, model VNMRS 300 MHz, using DMSO-*d*<sub>6</sub> as solvent and glass tubes of 5 mm in diameter and are in accordance with the literature. <sup>1</sup>H NMR (300 MHz, DMSO-*d*<sub>6</sub>, 25 °C) δ (ppm): 1.01 (m, CH<sub>3</sub>), 1.29 (m, CH<sub>2</sub>CH<sub>3</sub>), 1.83 (m, CH<sub>2</sub>), 3.97 (s, CH<sub>3</sub>), 4.25 (t, CH<sub>2</sub>N), 7.79 (s, H5), 7.91 (s, H4), 9.48 (s, H2). FTIR ν (cm<sup>-1</sup>): 3141 (N–H of imidazolium group), 3058 (C–H of imidazolium group), 2959 (C–H of CH<sub>2</sub>), 2870 (C–H of CH<sub>3</sub>), 1640 (C=N of imidazolium group), 1558–1453 (C=C and C–N of imidazolium group), 749 (Cl<sup>-</sup>).

### 2.2. Poly(ionic liquids) syntheses

Polyurethane was prepared according to the following procedure (see Fig. 1): PTMG polyol (2000 g/mol) (0.04 mol), DMPA diol (0.11 mol) dissolved in NMP and DBTDL (0.1% wt) as catalyst, were dissolved in MEK (50 mL) in a five-necked flask. Then, HDI was added dropwise to the system and the reaction mixture was stirred at 60 °C for 2 h. The NCO/OH ratio of 1.05 (0.157 mol HDI/0.15 mol OH) was used. The second step was the addition into the reaction medium of the selected IL (see Fig. 1). In this step, the solid content

was quantified as well as the acidity of the polymer (by titration with KOH 0.5 M). The acid number reached a value of 25 mgKOH/g (0.45 mmol of IL/g polymer). After that, in order to neutralize the carboxylic acid group, the system was cooled to 40 °C and IL (bmim Cl; TBAB or TBPB) was charged into the reactor. The system was kept at this temperature (40 °C) for 4 h, under stirring to obtain a PIL with the desired cation. The neutralization step occurred at a molar ratio COOH/IL = 1:1. Finally, films with a thickness close to 0.15 mm were produced by casting and dried under vacuum at 63 °C for 24 h.

### 2.3. Characterization of poly (ionic liquid)s

The structural elucidation of the samples was carried out by Fourier Transform Infrared (FTIR) and Proton Nuclear Magnetic Resonance (<sup>1</sup>H NMR) spectroscopic techniques. FTIR spectra were recorded on a Perkin-Elmer Spectrum 100 spectrometer in ATR (Attenuated Total Reflection) mode. <sup>1</sup>H NMR spectra were obtained using a Bruker Avance DRX-400 spectrometer at 400 MHz in DMSO-*d*<sub>6</sub>. Molecular weights were acquired from a Gel Permeation Chromatograph (GPC), equipped with a Waters 1515 pump and a Waters 2412 refractive index detector, using THF as eluent at a flow rate of 1 mL/min; samples to be analyzed were dissolved in DMSO and then diluted with THF. Differential Scanning Calorimetry (DSC) thermograms were attained by using a TA Instrument Q20 differential scanning calorimeter in the range of –90 °C–170 °C, or 200 °C at a heating rate of 10 °C/min under nitrogen. Thermogravimetric Analysis (TGA) was performed using a TA Instrument SDT-Q600 between 25 °C and 600 °C at a heating rate of 10 °C/min in a nitrogen atmosphere. Tensile tests (stress x strain curves) were carried out at 25 °C with rectangular shape films (12 mm long; 7 mm wide) with a thickness close to 0.15 mm, on a DMTA equipment (model Q800, TA Instruments) with 1 N/min. The Young Modulus of the materials was determined according to ASTM D638. The analyses were carried out in triplicate. The field emission scanning electron microscopy (FESEM) analyses were performed in FEI Inspect F50 equipment in secondary electrons (SE) mode. The AFM analyses were performed in Peak force tapping mode using a Bruker Dimension Icon PT equipped with a TAP150A probe (Bruker, resonance frequency of 150 kHz and 5 N m<sup>-1</sup> spring constant). The equipment was calibrated prior to samples measurements. The scanned area of the images was 60 × 60 μm<sup>2</sup> with a resolution of 512 frames per area. The DMT Modulus map was derived from PeakForce.

### 2.4. Sorption measurements

The CO<sub>2</sub> sorption tests were gravimetrically assessed in a Magnetic Suspension Balance (MSB), (Rubotherm Präzisionsmesstechnik GmbH, 350 bar and 673.15 K). The materials sorption capability was isothermally assessed in the range of 0.8–50.0 bar pressure at 303.15 K. The apparatus details are described elsewhere [24,25]. The experimental procedure was detailed in our previous work [18].

### 2.5. Computer simulations

Molecular dynamics simulations were carried out by the PM7-MD method [26–31]. In PM7-MD, immediate forces acting on all atoms, including hydrogens, are found according to the PM7 semi-empirical Hamiltonian at every time-step. The motion of atomic nuclei is, therefore, directly correlated with the evolution of the system's electronic configuration. Quantum effects are reproduced in PM7-MD simulations, out of which inter- and intra-molecular electronic polarization is most important for the present work.



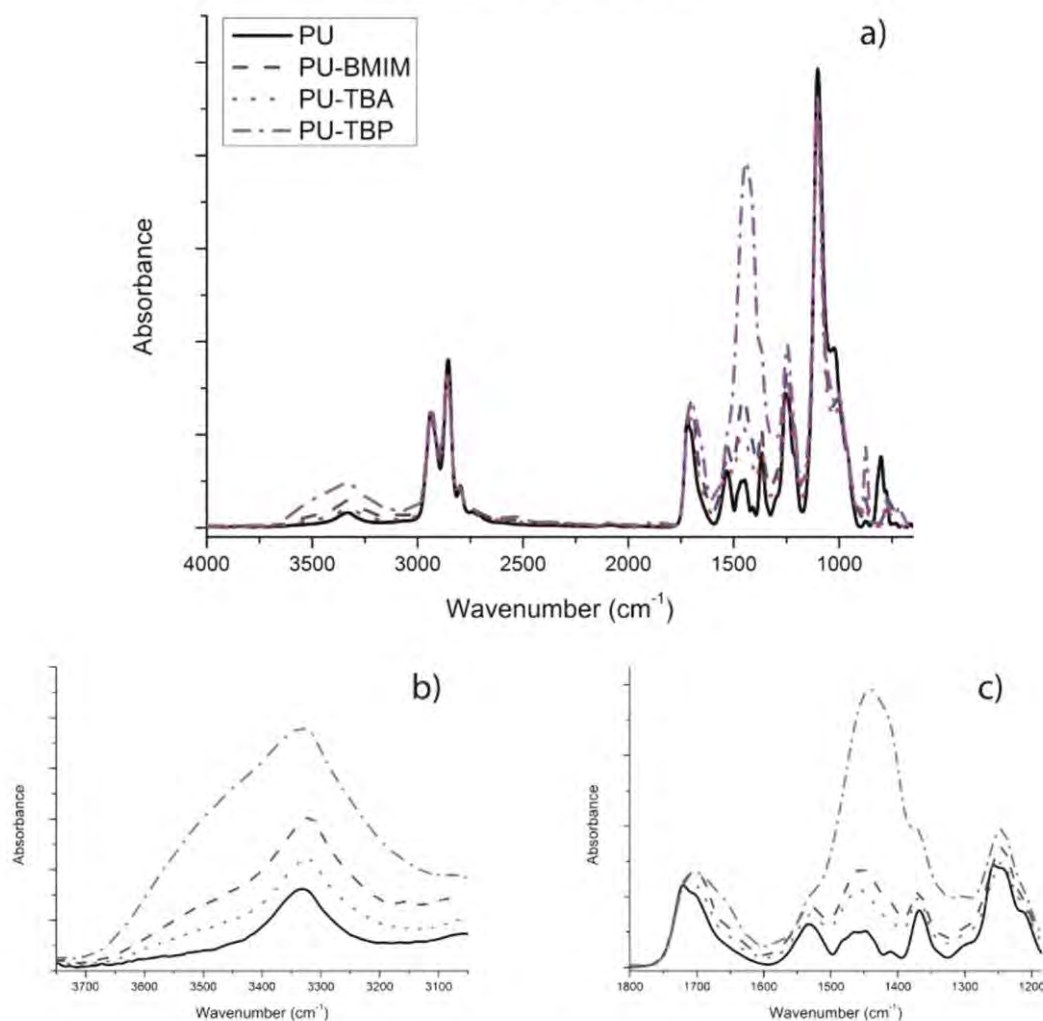


Fig. 2. FTIR spectra of polyurethane (PU) and PILs: (a) 660–4000  $\text{cm}^{-1}$ , (b) 3000–3500  $\text{cm}^{-1}$ , (c) 1200–2000  $\text{cm}^{-1}$ .

area evidences dependence of the absorptivity coefficient on the strength of hydrogen bonding, which in turn varies upon presence of the cation, especially tetrabutylphosphonium.

The carbonyl stretching region in the spectrum of polyurethane also indicates the presence of bands, at about 1721  $\text{cm}^{-1}$  and 1699  $\text{cm}^{-1}$  associated with “free” (non-bonded hydrogen) carbonyl groups and bonded hydrogen to carbonyl groups, respectively [39]. In this region, a broad band is observed being assigned to superposition of the hydrogen bonded and free carbonyl group absorption. Increase of the band area and frequency decrease from 1720  $\text{cm}^{-1}$  (PU) to 1702  $\text{cm}^{-1}$  upon the cation addition (Fig. 2c) indicate the hydrogen bond formation. The C=O bands from the carboxylic acid group (COOH) and the urethane group (NHCO) are in overlay [38,41] (Fig. 2c).

The synthesized PU presents COOH group from PTMG. It can create hydrogen bond both with hard segments and with C–O–C group of the soft segments [41]. After ionization, the carboxylic acid group (COOH) was transformed to the COO<sup>-</sup> group. Decrease in the band area of C–O–C group (1103  $\text{cm}^{-1}$ ) (Fig. 2a) was observed.

Thus, the number of hydrogen bonds between soft and hard segments was reduced. The absorption range of –COO<sup>-</sup> (1470  $\text{cm}^{-1}$ ) [18,42,43] overlaps with that of –CH<sub>2</sub> and –CH<sub>3</sub> [43] (Fig. 2c). It was verified that after ionization the band shifts to lower frequency (from 1462  $\text{cm}^{-1}$  to 1444  $\text{cm}^{-1}$ ) and significantly increases the area. The intensity increase after ionization indicates presence of the carboxylate ion (–COO<sup>-</sup>) in the PILs. The band intensity for TBP is more pronounced due to a larger diameter of the phosphorus atom and, consequently, a smaller electron density on this atom exhibiting the weakest possible cation-anion coordination, as corroborated by simulation results discussed later.

The IR spectra (Fig. 2a) also reveal others characteristic polyurethane adsorption bands: 2936  $\text{cm}^{-1}$  (C–H of CH<sub>2</sub>), 2858  $\text{cm}^{-1}$  (C–H of CH<sub>3</sub>), 1532  $\text{cm}^{-1}$  (H–N), 1245  $\text{cm}^{-1}$  (C–N and C–O of urethane), 1026  $\text{cm}^{-1}$  (C–O–C) and 865  $\text{cm}^{-1}$  (C–O–C) [18,44]. Accordingly, the structures of the synthesized polymers were analyzed by <sup>1</sup>H NMR. The different resonance peaks and their corresponding assignments are in accordance with literature [18,44] and were also compared with the starting materials spectra

from the Spectral Database for Organic Compounds (SDBS). A typical  $^1\text{H}$  NMR spectrum of the sample PU-TBA is presented, for reference, in Fig. 3.

The peaks of the all NMR spectra showed the signals of the polyurethane chain,  $\delta$  (ppm): 1.57 (2H,  $\text{NH}-\text{CO}-\text{O}-\text{CH}_2-\text{CH}_2$ , m, of PTMG), 3.35 (2H, t,  $\text{O}-\text{CH}_2-\text{CH}_2-$  of PTMG), 3.98 (2H, t,  $\text{NH}-\text{CO}-\text{O}-\text{CH}_2-$  of PTMG), 1.28 (2H, m,  $\text{N}-\text{CH}_2-\text{CH}_2-(\text{CH}_2)_2-$  of HDI), 1.42 (2H, m,  $\text{N}-\text{CH}_2-\text{CH}_2-$  of HDI), 3.00 (2H m,  $\text{N}-\text{CH}_2-\text{CH}_2-$ , m, of HDI), 1.10 (3H, s,  $\text{CH}_3$  of DMPA), 4.08 (2 H, s,  $\text{R}-\text{NH}-\text{CO}-\text{O}-\text{CH}_2-\text{C}$  of DMPA), 7.10 (1H, s, NH urethane).

The presence of the TBA cation are assigned at  $\delta$  (ppm): 3.25 (2H t,  $\text{CH}_3\text{CH}_2\text{CH}_2\text{CH}_2\text{N}$  of TBA), 1.63 (2H, m,  $\text{CH}_3\text{CH}_2\text{CH}_2\text{CH}_2\text{N}$  of TBA), 1.38 (2H, m,  $\text{CH}_3\text{CH}_2\text{CH}_2\text{CH}_2\text{CH}_2\text{N}$  of TBA), 1.02 (3H, t,  $\text{CH}_3\text{CH}_2\text{CH}_2\text{CH}_2\text{N}$  of TBA).

The presence of the bmim cation is manifested by  $\delta$  (ppm): 9.22 (1H, s,  $\text{NCHN}$  of bmim), 7.84 (1H, t,  $\text{CH}_3\text{NCHCHN}$  of bmim), 7.77 (1H, t,  $\text{CH}_3\text{NCHCHN}$  of bmim), 4.25 (2H, t,  $\text{NCH}_2(\text{CH}_2)_2\text{CH}_3$  of bmim), 3.91 (3H, s,  $\text{NCH}_3$  of bmim), 1.81 (2H, m,  $\text{NCH}_2\text{CH}_2\text{CH}_2\text{CH}_3$  of bmim), 1.30 (2H, m,  $\text{N}(\text{CH}_2)_2\text{CH}_2\text{CH}_3$  of bmim), 0.97 (3H, t,  $\text{N}(\text{CH}_2)_3\text{CH}_3$  of bmim).

The signals of TBP cation are assigned at  $\delta$  (ppm): 2.25 (2H, m,  $\text{CH}_3\text{CH}_2\text{CH}_2\text{CH}_2\text{P}$  of TBP), 1.55 (2H, m,  $\text{CH}_3(\text{CH}_2)_2\text{CH}_2\text{P}$  of TBP), 0.98 (3H, t,  $\text{CH}_3\text{CH}_2\text{CH}_2\text{CH}_2\text{P}$  of TBP).

The DSC measurements were carried out to PILs and PU. The DSC curves of all samples (see Fig. 4) show an endothermic peak (PURE PU = 21 °C; PU-bmim = 22.2 °C; PU-TBP = 23.4 °C and PU-TBA = 21.5 °C), which represents the melting of the crystalline microphase. This behavior is due to the molecular weight ( $M_n = 2000$ ) of soft segments (PTMG) which favors the crystallization. The same behavior was not evidenced for PTMG with lower molecular weight ( $M_n = 1000$ ) [45,46].

The melting temperature of the microphase is not influenced by the cation insertion into the polymeric chain. The same behavior was observed elsewhere [40] for cationic polyurethane containing

pendant trialkylphosphonium. The melting and crystallization enthalpy also presented a small variation (PU- $\Delta H_m = 26.0$  J/g and  $\Delta H_c = 22.5$  J/g; samples PU-bmim, PU-TBP, PU-TBA presented values of  $\Delta H_m = 22.2$  J/g and  $\Delta H_c = 21.2$  J/g,  $\Delta H_m = 21.9$  J/g and  $\Delta H_c = 19.5$  J/g,  $\Delta H_m = 22.4$  J/g and  $\Delta H_c = 19.7$  J/g, respectively). Nevertheless, the crystallization temperatures of PU (−26.9 °C) and the poly(ionic liquid) PU-bmim (−26.9 °C) are similar, but the chains with the cations TBP and TBA presented a lower crystallization temperature (−30.6 °C and −30.3 °C, respectively).

Thermal stability of PU and PILs was investigated by thermogravimetric analysis (TGA) (see Table 1 and Fig. 5). The cation insertion decreased the thermal stability of anionic structures. The same behavior was not observed by Zhang et al. [40] for cationic structures. The insertion of TBP in the polymeric structure resulted in higher thermal stabilities. The TGA thermograms prove that they are stable up to 190 °C and 210 °C to PILs depending on the cation and 236 °C for PU. The TGA thermograms indicate two stages of thermal degradation processes.

In the first step, the decomposition temperatures correspond mainly to the degradation of hard segment, which occurs basically through the dissociation or breaking of the urethane bonds [47,48]. The second stage of thermal degradation is attributed to the decomposition of polyol, i.e. PTMG soft segments [49]. Meanwhile, PILs in this stage may go through the decomposition of the part related to the cation. Such decomposition has been reported in a temperature interval of 140–350 °C [50]. The final step of weight loss occurs in the temperature range 250–507 °C and could be due to the disintegration of the PU moiety into simpler molecules like  $\text{CO}_2$  and ethers.

Fig. 6 shows SEM images of PU and the PILs. The PU microstructure exhibited a smooth surface without pores. The SEM images of the PILs are very similar to those obtained for PU.

We observed by AFM images (see Fig. 7) that the hard domains of PU (lighter regions) and the soft domains (dark regions) present

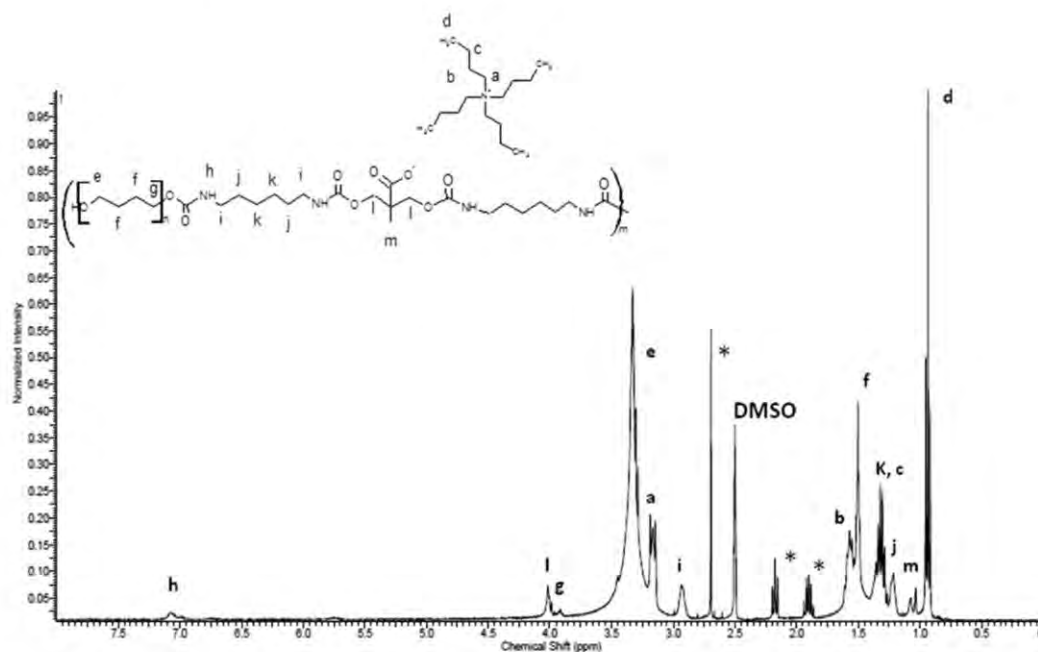


Fig. 3.  $^1\text{H}$  NMR spectrum for the PU-TBA. \*: Residual N-methyl-2-pyrrolidone (NMP) solvent.

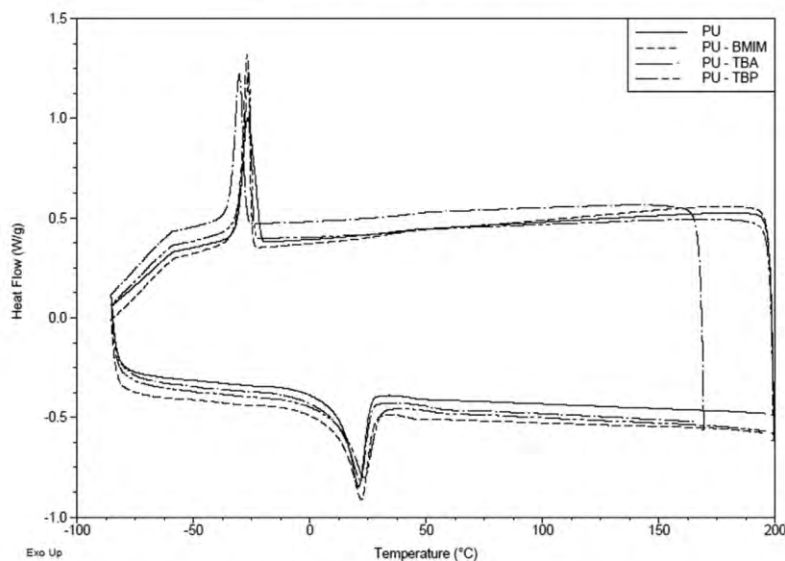


Fig. 4. DSC thermograms for PU and PILs.

**Table 1**  
Temperature range obtained from derivative TGA curves of PU and PILs.

Sample	Degradation stage	T <sub>onset</sub> (°C)	T <sub>endset</sub> (°C)
PU	1	236	354
	2	354	493
PU-bmim	1	202	250
	2	250	493
PU-TBP	1	210	256
	2	256	488
PU-TBA	1	190	356
	2	356	507

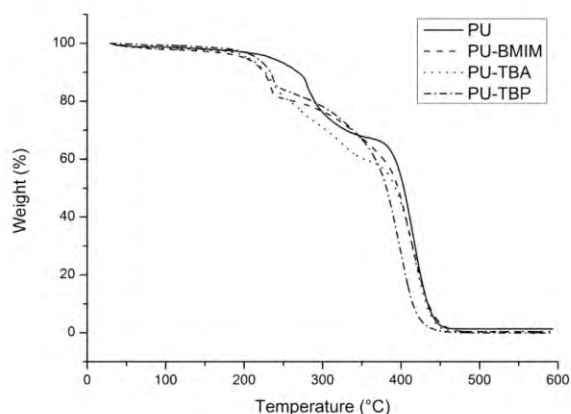


Fig. 5. TGA thermograms for PU and PILs.

some connectivity or continuity associated with them [40,51]. This behavior is probably due to the hydrogen bonding between hard/hard segments and hard/soft segments.

Upon cation addition to the polymeric chain, the morphology changes. In this case, the hard and soft segments reorganize

changing their mutual miscibility resulting in an increased degree of microphase segregation. This behavior can influence mechanical properties of the materials [51].

The tensile properties of PU and the poly(ionic liquids) are presented in Fig. 8. The Young moduli are given in Fig. 9. The stress values increase with the cation addition (TBA and TBP).

The values of the Young modulus (See Fig. 9) increased in the following order: PU-bmim < PU < PU-TBA < PU-TBP. The poly(ionic liquid) with the imidazole cation presented the lower elastic modulus, probably thanks to a higher interaction of the imidazole ring with the carboxyl group. PILs PU-TBA and PU-TBP presented higher modulus of elasticity comparing with PU and PU-bmim. The changes evidenced by AFM and FTIR when the ionic compound is formed probably influence mechanical behavior of the poly(ionic liquids).

Zhang et al. [40] reported the synthesis of cationic polyurethanes containing pendant trialkylphosphonium. They evidenced that the addition of phosphonium into the polyurethanes resulted in several desirable advantages including improved microphase separation and enhanced mechanical performance compared to the non-charged polymer.

### 3.2. CO<sub>2</sub> sorption - experimental results

The results of CO<sub>2</sub> sorption obtained for the poly(ionic liquids) and PU are shown in Fig. 10. Also in Table 2 is presented a comparative study of CO<sub>2</sub> sorption of this work with different poly(ionic liquids) described in literature.

The CO<sub>2</sub> sorption results presented in Fig. 10 show typical physical adsorption shape curves (no chemical reaction takes place). It increases upon increasing CO<sub>2</sub> partial pressure [52,53]. PU presents a CO<sub>2</sub> sorption value of 7.4 mgCO<sub>2</sub>/g (0.82 bar) and 140 mgCO<sub>2</sub>/g (50 bar) due to the interaction between CO<sub>2</sub> and the polar groups (usually O, N) of the polymeric chains [18,54,55]. It can be seen in Fig. 10 that the cation plays a role in CO<sub>2</sub> sorption values. The CO<sub>2</sub> solubility was increased by the inclusion of the cation. For a pressure of 0.82 bar PU-bmim presented a CO<sub>2</sub> sorption of 9.4 mgCO<sub>2</sub>/g, PU-TBP exhibited a value of 15.7 mgCO<sub>2</sub>/g and PU-TBA



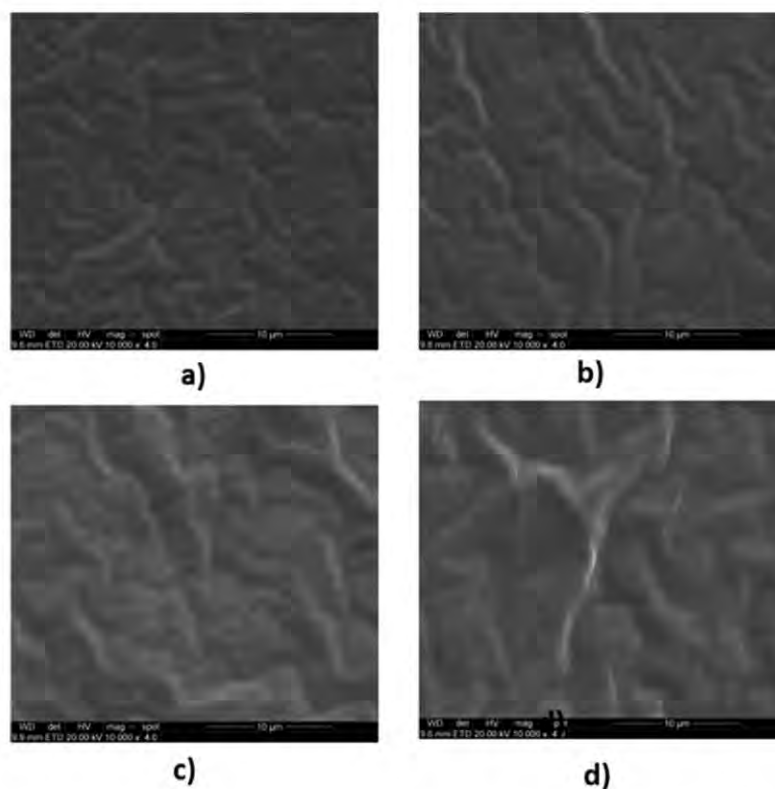


Fig. 6. Micrographs of the materials at magnification of 10000 $\times$ : (A) PU, (B) PU-BMIM, (C) PU-TBP, (D) PUTBA.

a sorption capacity of 16.1 mgCO<sub>2</sub>/g. The amounts of CO<sub>2</sub> sorption at 50 bar were 162 mgCO<sub>2</sub>/g for the PU-bmim, 168 mgCO<sub>2</sub>/g for PU-TBP and 226 mgCO<sub>2</sub>/g for PU-TBA.

In order to examine the reusability of PILs, PU-TBA was subjected to five additional runs. The sorption was carried out at 303.15 K under 1 MPa CO<sub>2</sub> pressure. After each step, the CO<sub>2</sub> desorption was carried out under vacuum at 298.15 K for 6 h. The results for the six consecutive tests are shown in Fig. 11.

The CO<sub>2</sub> sorption capacity was slightly increased in the first three recycles, probably due to impurities withdrawal remaining constant for the next cycles. These results indicate that the PU-TBA has high stability and capacity of being reused. In order to obtain performance data for our PU-TBA and PU-TBP, the CO<sub>2</sub> sorption capacity was compared with other PILs described in literature (Table 2).

The CO<sub>2</sub> sorption values showed that the CO<sub>2</sub> uptake capacity of the PILs reported in literature are lower, as compared to PU-TAB and PU-TBP. This behavior is probably related with the weakest possible cation-anion coordination achieved by TBA and TBP, as described in Section 3.3. The results suggest that the novel anionic polyurethane using tetraalkylammonium and tetrabutylphosphonium as cations presents significant potential for carbon dioxide capture.

### 3.3. CO<sub>2</sub> sorption – simulation results

Simulations were conducted to understand the varying performance of the polymeric anion with the bmim, TBA, and TBP cations. Two reasons can be responsible for this. First, certain cations, e.g. the imidazolium-based ones, capture CO<sub>2</sub> with a reasonable

efficacy. The electron deficient hydrogen atom located at the C-2 position is known to form a relatively weak hydrogen bond with the oxygen atom of CO<sub>2</sub>. Second, the cation may play an adverse role provided that it blocks an active site in the anion and does not introduce alternative CO<sub>2</sub> binding sites. PCFs derived for selected atom pairs (Fig. 12) indicate a clear hydrogen bond, with a length  $L_{PM7-MD} = 2.1$  Å, between the hydrogen atom of the imidazole ring and the oxygen atom of the carboxyl group. This predicted hydrogen bond from molecular dynamics appears in excellent agreement with the result of the structure optimization employing hybrid DFT (B3LYP),  $L_{DFT} = 2.0$  Å. A small discrepancy,  $\Delta L = L_{PM7-MD} - L_{DFT} = +0.1$  Å, can be fully explained by thermal expansion of the system upon heating up to 300 K. No distance correlations were found for the ester and imide groups (Fig. 12a). Therefore, the bmim cation blocks an important CO<sub>2</sub> binding site (the deprotonated carboxyl group, COO<sup>-</sup>) and decreases an overall performance of the compound, as follows from the experiment. In turn, neither TBA nor TBP exhibit such a behavior. The corresponding peaks, N-carboxyl and P-carboxyl, is fuzzy significantly in the range 3–4 Å. Importantly, all cations coordinate the carboxyl group of the polymeric anion, whereas other polar sites of the anion are largely ignored by all investigated cations. The heights of the PCFs corresponding to phosphorus of TBP are somewhat smaller as compared to the same heights in the case of TBA. This is due to a larger diameter of the phosphorus atom and, consequently, a smaller electron density on this atom.

The CO<sub>2</sub> capturing performance of [TBA][anion], mg/g, is better than that of [TBP][anion], since the mass of nitrogen (14 a.m.u) is over two times smaller than the mass of phosphorus (31 a.m.u). Our

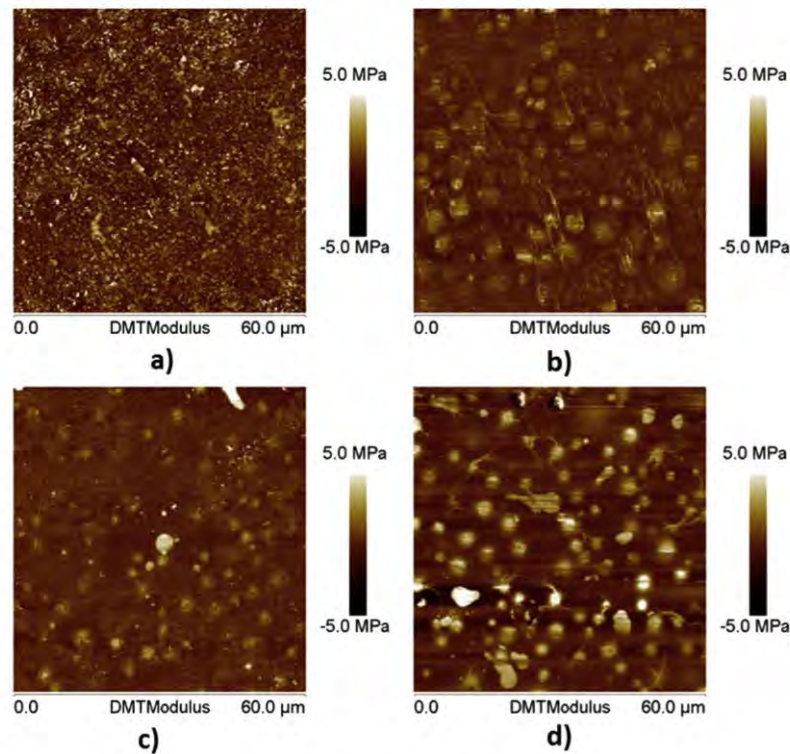


Fig. 7. DMT Modulus maps obtained by AFM images: (A) PU, (B) PU-bmim, (C) PU-TBP, (D) PU-TBA.

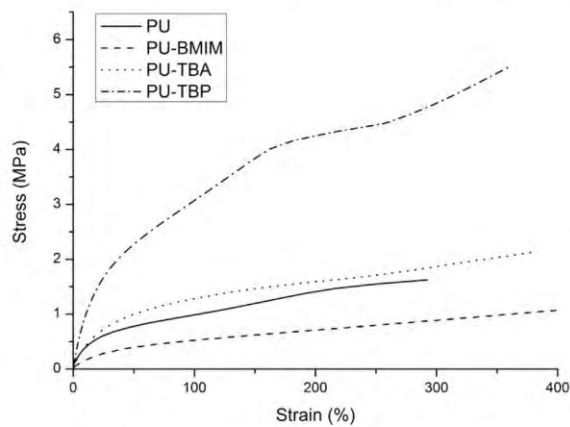


Fig. 8. Stress/Strain curves of pure PU and poly(ionic liquids).

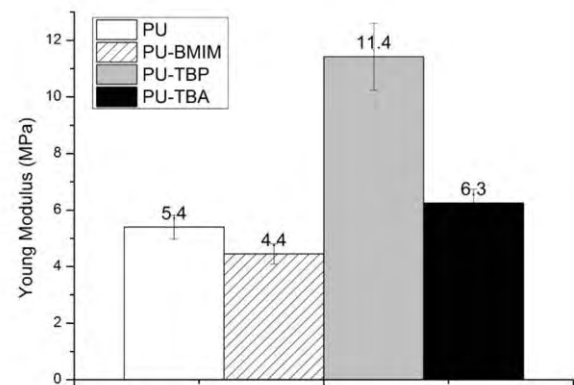


Fig. 9. Young modulus of pure PU and poly(ionic liquids).

investigation reveals that a successful compound for gas capture must exhibit the weakest possible cation-anion coordination and smaller molecular masses [56].

#### 4. Conclusions

This study reveals experimentally and by simulation a new and simple concept for low-cost sorbents design. A successful

compound for gas capture must exhibit the weakest possible cation-anion coordination and smaller molecular masses. The mechanical properties are influenced by cation insertion. The values of the Young modulus increased in the order PU-bmim < PU < PU-TBA < PU-TBP. The highest CO<sub>2</sub> sorption values were obtained for the anionic polyurethanes using tetrabutylammonium and tetrabutylphosphonium as cations. These new compounds show improved CO<sub>2</sub> sorption capacity, as compared to other PIL reported in the literature presenting significant potential for carbon dioxide capture.

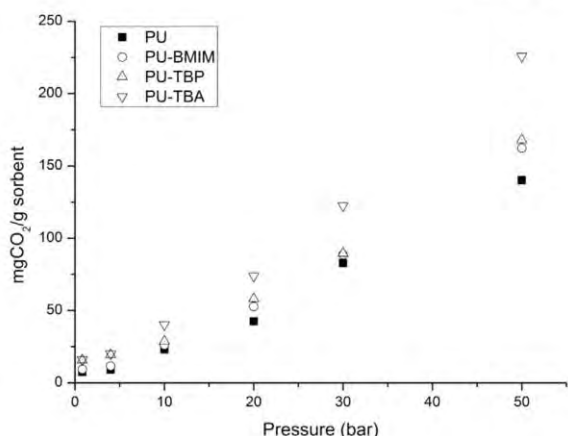


Fig. 10. Pure PU and poly(ionic liquids) sorption values at 303.15 °C.

Table 2

Comparison of PU-TBA and PU-TBP with different poly(ionic liquids) in terms of CO<sub>2</sub> sorption.

PIL	CO <sub>2</sub> sorption (mg/g)	Conditions (P, T)	Ref.
[PVBIT] <sup>a</sup>	3.05	0.79 bar, 295.15 K	[7]
[PVBH] <sup>b</sup>	3.22	0.79 bar, 295.15 K	[7]
P[AMIM][BF <sub>4</sub> -AN] <sup>c</sup>	14.30	1 bar, 273.15 K	[17]
PG [BIEMA][Br] <sup>d</sup>	3.34	1 bar, 278.15 K	[11]
PG [BIEMA][acetate] <sup>e</sup>	12.46	1 bar, 278.15 K	[11]
P[VBTEA][PF <sub>6</sub> ] <sup>f</sup>	14.04	1 bar, 278.15 K	[5]
PU-TBP	15.70	0.82 bar, 303.15 K	This study
PU-TAB	16.10	0.82 bar, 303.15 K	This study

<sup>a</sup> Poly[1-(4-vinylbenzyl)-3-butylimidazolium tetrafluoroborate].

<sup>b</sup> Poly[1-(4-vinylbenzyl)-3-butylimidazolium hexafluorophosphate].

<sup>c</sup> Poly(ionic liquid)s based on the copolymer of 1-allyl-3-methylimidazolium tetrafluoroborate and acrylonitrile.

<sup>d</sup> Poly[2-(1-butylimidazolium-3-yl)ethyl methacrylate bromide].

<sup>e</sup> poly[2-(1-butylimidazolium-3-yl)ethyl methacrylate acetate].

<sup>f</sup> Poly(4-vinylbenzyltriethylammonium hexafluorophosphate).

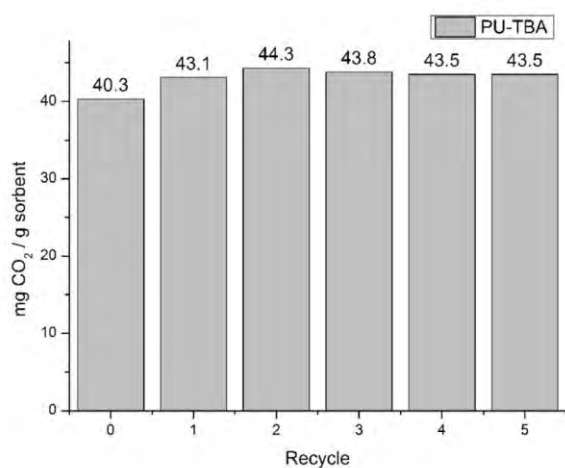


Fig. 11. CO<sub>2</sub> sorption/desorption tests for the PIL PU-TBA.

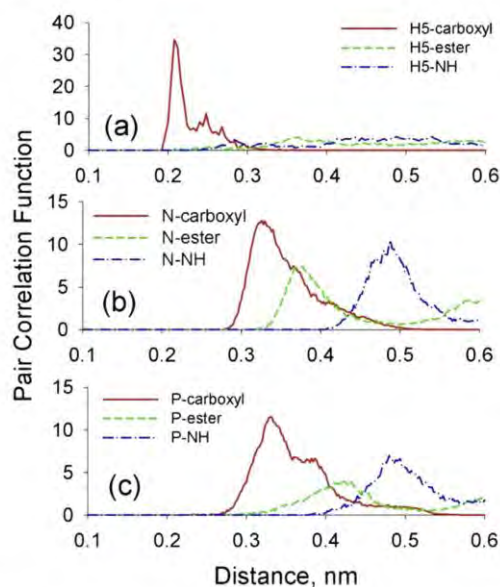


Fig. 12. Pair correlation functions for selected inter-molecular distances (see legends) in (a) [bmim][anion]; (b) [TBA][anion]; (c) [TBP][anion]. "H5" is an intrinsically acidic hydrogen atom of the imidazole ring; "P" and "N" are the central atoms of the TBA and TBP cations, respectively; "carboxyl" corresponds to the oxygen atoms of the negatively charged moiety of the polymeric anion (one per polymer unit); "ester" is the oxygen atoms in the –COO– group (two per polymer unit); "NH" is the nitrogen atom of the imide group (two per polymer unit).

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## 4.2. Capítulo II: Poli(líquidos iônicos) aniônicos base poliuretano-imida

A síntese e caracterização de PLIs aniônicos base PU-imida são descritas neste capítulo, por meio do artigo intitulado *“Hybrid alkoxysilane-functionalized urethane-imide-based poly(ionic liquids) as a new platform for carbon dioxide capture”*, submetido ao *“Energy & Fuels”* (comprovante de submissão Anexo A). Neste trabalho, foi investigado o efeito da concentração do grupo silano e do contra-cátion (imidazólio, fosfônio, amônio e pirrolidínio) sobre as propriedades térmicas, mecânicas morfológicas e capacidade de sorção de CO<sub>2</sub> dos PLIs base PU-imida. Os materiais obtidos foram caracterizados por FTIR, RMN, DSC, TGA, DMA e AFM. A capacidade de sorção e reuso no processo de captura foi determinada usando uma célula de decaimento de pressão na temperatura de 303,15°C e pressões de 0,8 e 10 bar. Também foram utilizados estudos de simulação para avaliar a interação entre o CO<sub>2</sub> os grupos funcionais presentes na estrutura dos PLIs. Os valores de sorção de CO<sub>2</sub> e estudos de simulação demonstraram que o cátion tem um efeito importante sobre a afinidade pelo CO<sub>2</sub>, enquanto que a concentração de silano não exerce impacto significativo. Estudos de simulação mostraram que o principal sítio de interação com o CO<sub>2</sub> é o COO<sup>-</sup>, assim o uso de contra-cations que coordenem fracamente este sítio favorecem a sorção de CO<sub>2</sub>. O PLI que demonstrou boas propriedades térmicas, mecânicas e melhor desempenho de sorção foi o HPLI-02-TBA (33,1mg/g a 303,15 K e 0,82 bar). Este PLI exibiu elevada propriedade mecânica e alta capacidade de sorção de CO<sub>2</sub>, quando comparado com PLIs base PU.

## Hybrid alkoxysilane-functionalized urethane-imide-based poly(ionic liquids) as a new platform for carbon dioxide capture

Franciele L. Bernard<sup>a,b</sup>, Barbara B. Polesso<sup>c</sup>, Fabiana W. Cobalchini<sup>c</sup>, Vitaly V. Chaban<sup>d</sup>, Jailton F. do Nascimento<sup>e</sup>, Felipe Dalla Vecchia<sup>c</sup> and Sandra Einloft<sup>a,b\*</sup>

a) Post-Graduation Program in Materials Engineering and Technology. Pontifical Catholic University of Rio Grande do Sul – PUCRS.

b) School of Chemistry. Pontifical Catholic University of Rio Grande do Sul – PUCRS.

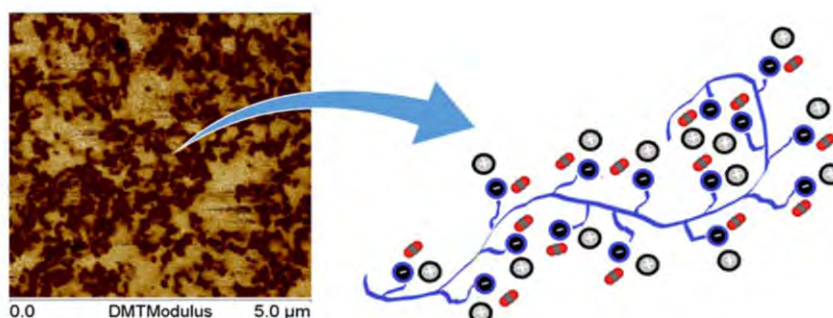
c) School of Engineering, Pontifical Catholic University of Rio Grande do Sul – PUCRS. Av. Ipiranga, 6681, Partenon, Porto Alegre Brazil, CEP: 90619-900.

d) Institute of Science and Technology (ICT), Federal University of São Paulo (UNIFESP); São José dos Campos, SP – Brazil.

e) Petrobras / CENPES. Ilha do Fundão Qd. 07; Rio de Janeiro – RJ, Brasil.

\* Corresponding author: [einloft@pucrs.br](mailto:einloft@pucrs.br)

### Graphical Abstract



**Abstract.** Development of new and improvement of the existing materials for carbon dioxide (CO<sub>2</sub>) capture is an urgent and significant goal for emission reduction. We hereby report synthesis of hybrid urethane-imide-based poly-ILs (HPILs) and their CO<sub>2</sub> capture capacities. The synthesized HPILs were characterized by FTIR, NMR, DSC, TGA, DMTA, AFM. CO<sub>2</sub> physisorption and reusability were assessed by the pressure-decay technique at a few conditions. Density functional theory calculations were used to identify binding energies between CO<sub>2</sub> and each center of HPILs. Cations of HPILs play an important role in CO<sub>2</sub> physisorption. The impact of the silane content was found to be relatively insignificant. Weakly coordinating cations foster better CO<sub>2</sub> sorption. The best performances were obtained for the tetrabutylammonium-based HPILs (33.1mg/g in HPIL-02-TBA and 31.7 mg/g in HPIL-06-TBA at 303.15 K and 0.82 bar). HPIL-02-TBA possesses the highest CO<sub>2</sub> sorption capacity out of all reported poly(ionic liquids) thus far and exhibits interesting thermal stability and competitive mechanical properties.

**Keywords:** poly ionic liquids, poly(urethane-imide)-based, hybrid, anionic, carbon dioxide; capture

## 1. Introduction

Fossil fuel burning is, allegedly, the largest human contribution to climate change due to heavy emissions of carbon dioxide (CO<sub>2</sub>) [1–3]. Reduction of CO<sub>2</sub> emissions is necessary to mitigate global warming and unpredictable climate changes [2,4–6]. Carbon capture and storage (CCS) is an effective short-term solution to control CO<sub>2</sub> emissions from the industrial energy sources [2,5]. In this context, development of new, more efficient sorbents for CO<sub>2</sub> capture has become a major research focus in this field [4].

Room temperature ionic liquids (RTILs) are formally created by combining an organic cation and inorganic/organic anion. RTILs possess melting points below 100°C [5,7–9]. RTILs have been described as solvents to remove CO<sub>2</sub> from the flue gas [10–15]. Many RTILs exhibit unique physicochemical properties, including negligible vapor pressure, low flammability, high thermal stability, etc [16]. Physicochemical properties of RTILs can be finely tuned by combining hundreds of cations with hundreds of anions. Most RTILs are miscible with one another thanks to the prevailing electrostatic attraction between any cation and any anion. High viscosity and relatively low CO<sub>2</sub> sorption/desorption rates [15–18] are seen as drawbacks of RTILs. Recent success in polymerized RTILs (PILs) [16,17,19] motivates development of improved CO<sub>2</sub> sorbents.

PILs represent an emerging class of the polyelectrolyte species consisting of the RTIL monomers [4,5,19–22]. These polymers constitute a new, more versatile platform giving rise to a variety of high-performance sorbents [4,5,14–17,19,20,23]. PILs combine useful features of RTILs, e.g. tenability [5,19,20], with mechanical stability, processability, and adjustable macromolecular design of polymers [5,14,19,20,22,24]. CO<sub>2</sub> sorption and desorption in PILs are completely reversible and occur more quickly than in conventional RTILs [25,26]. PILs synthesis is conducted via two general procedures. The first one is a direct polymerization of the RTIL monomers [5]. In turn, the second one involves condensation reactions and/or polymer modification [5,22]. Polycondensation and modification of the existing polymers is more attractive because it utilizes commercially available

polymers of high molecular weight [22,27]. A few successful PILs for CO<sub>2</sub> capture were synthesized by this procedure: polyurethane (PU) [28–31], polybenzimidazoles [27,32], Cellulosic [33] and polyimides [22]. Multiple factors affect CO<sub>2</sub> solubility in PILs, e.g. ion structure, alkyl chain length, molecular weight, electron density distribution, moisture and polymeric macrostructure [4,5].

We have recently reported synthesis of cationic and anionic PU-based PILs introducing different cations imidazolium, phosphonium, and ammonium families [28–30]. Highest CO<sub>2</sub> sorption capacity was attained for anionic polyurethane using tetrabutylammonium (16.1 mgCO<sub>2</sub>/g at 0.82 bar and 303.15K) as cation. This compound exhibited weak cation-anion coordination. These new compounds exhibit superior CO<sub>2</sub> sorption, as compared to other PILs. For example, PIL based on the copolymer of 1-allyl-3-methylimidazolium tetrafluoroborate and acrylonitrile presented a sorption value of 14.30 mgCO<sub>2</sub>/g at 1 bar and 278.15K [16].

Development of organic-inorganic PILs was exemplified by Cheng et al. by immobilization of (p-vinylbenzyl)-triethylammonium tetrafluoroborate and (p-vinylbenzyl)-triethylammonium hexafluorophosphate onto a mesoporous silica by surface-initiated atom-transfer radical polymerization[6].

In this work, several new hybrid anionic poly(urethane-imide)-based poly(ionic liquids) were obtained by condensation polymerization. Imide and silane groups were added to polyurethane chain in order to obtain HPILs with improved mechanical properties and sorption capacity. The following organic cations were used: imidazolium, ammonium, phosphonium, pyrrolidinium. We, furthermore, evaluate the impact of silane content on the polymeric properties and CO<sub>2</sub> sorption performance.

## **2. Experimental**

### **2.1 Materials**



Poly(tetramethylene ether) glycol (PTMG,  $M_n=2000$  g/mol, Sigma-Aldrich), hexamethylene diisocyanate (HDI, 99%, Merck), dimethylol propionic acid (DMPA, 99%, Perstorp), 4,4'-oxydiphthalic anhydride (ODPA, 97%, Sigma Aldrich), 3-aminopropyltrimethoxysilane (APS, 97%, Sigma Aldrich) dibutyl tin dilaurate (DBTDL, Miracemanuodex), N-methyl-2-pyrrolidone (NMP, 99.92%, Neon), methylethylketone (MEK, 99%, Mallinckrodt), tetrahydrofuran (THF, 99%, Vetec), potassium hydroxide (KOH,  $\geq 85\%$ , Sigma-Aldrich) were used as purchased. Tetrabutylammonium bromide (TBAB, 99%, Acros Organics), tetrabutylphosphonium bromide (TBPB, 98%, Sigma Aldrich), 1-butyl-1-methylpyrrolidinium chloride (BMPYRR, 99%, Sigma Aldrich) were dried under vacuum during 12 hours at 60°C. 1-butyl-3-methylimidazolium chloride (bmimCl) was synthesized as described elsewhere [34,35]. The synthesized bmimCl was characterized by Fourier-transform infrared spectroscopy (FTIR) using Perkin-Elmer spectrophotometer model Spectrum 100 FT-IR with full attenuated reflectance (ATR), and by proton nuclear magnetic resonance ( $^1\text{H-NMR}$ ), in Varian spectrophotometer, VNMRs 300 MHz, using DMSO- $d_6$  as solvent and glass tubes of diameter 5 mm.  $^1\text{H-NMR}$  (300 MHz, DMSO $d_6$ , 25°C),  $\delta$  (ppm): 1.01 (m,  $\text{CH}_3$ ), 1.29 (m,  $\text{CH}_2\text{CH}_3$ ), 1.83 (m,  $\text{CH}_2$ ), 3.97 (s,  $\text{CH}_3$ ), 4.25 (t,  $\text{CH}_2\text{N}$ ), 7.79 (s,  $\text{H}_5$ ), 7.91 (s,  $\text{H}_4$ ), 9.48 (s,  $\text{H}_2$ ). FTIR  $\nu$  ( $\text{cm}^{-1}$ ): 3141 (N-H of imidazole), 3058 (C-H of imidazole), 2959 (C-H of  $\text{CH}_2$ ), 2870 (C-H of  $\text{CH}_3$ ), 1640 (C=N of imidazole), 1558-1453 (C=C and C-N of imidazole), 749 ( $\text{Cl}^-$ ).

## 2.2. Synthesis of hybrid poly(ionic liquid)s

Synthesis of HPILs was performed in three steps (Fig. 1), as described elsewhere [30,36–38]. Initially, the NCO-terminated PU pre-polymer was synthesized in five-necked flask at 65°C during 70 min using excess of diisocyanate (HDI), PTMG polyol, DMPA diol and DBTDL (0.1 wt%) as catalyst in the mixture of solvents (MEK and MNP, 1:1). The NCO/OH ratio of 2 (2 HDI/ PTMG 0.3 + DMPA 0.7) was used. At the next step, diisocyanate excess was neutralized with aromatic dianhydride and alkoxy silane (ODPA + APS) according to the following procedure: ODPA

temperature during 120 min providing a hybrid PU-imide solution. The synthesized hybrid PU-imide solution was labelled as PUIS-x, where x is APS fraction. For example, PUIS-02 means (PTMG + DMPA)/HDI/(ODPA + APS) = 1/2/(0.8 + 0.2), whereas PUIS-06 means (PTMG + DMPA)/HDI/(ODPA + APS) = 1/2/(0.4 + 0.6). The third step is addition of the selected RTIL (Fig. 1). The solid content and the polymer acidity were identified by titration with KOH 0.5 M. The acid number is 114 mgKOH/g (2 mmol of RTIL/g polymer). To neutralize carboxylic acid, the system was cooled down to 40°C and RTIL (bmimCl, TBAB, TBPB, BMPYRR) was loaded into the reactor. The system was kept at 40 °C during 240 min under stirring to obtain PILs. The neutralization step occurs at molar ratio COOH/IL = 1:1. Films around 0.15 mm thick were produced by casting. The films were post-cured at 150°C during 120 min under vacuum[36].

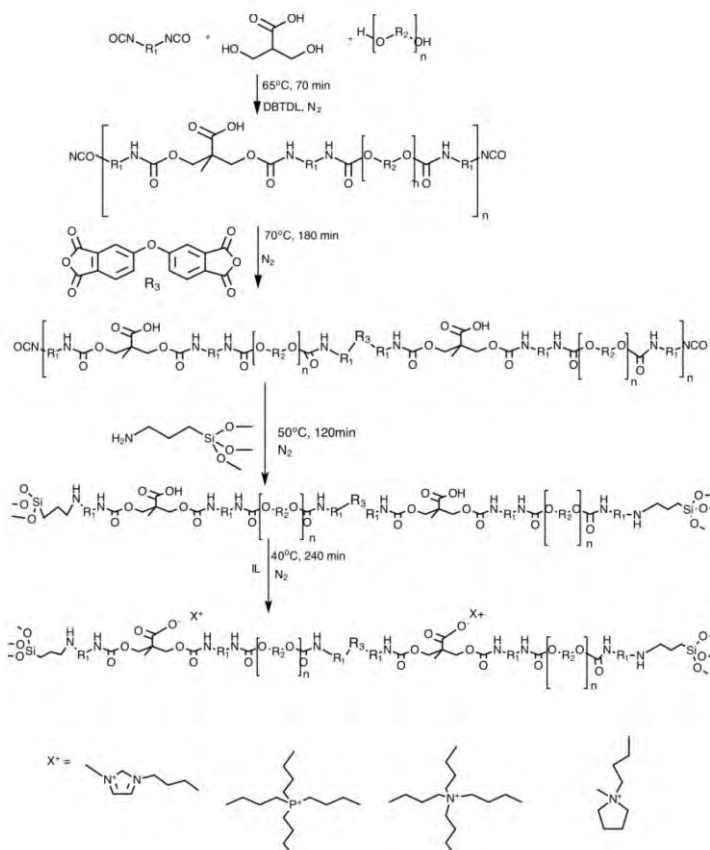


Figure 1. Synthesis steps to obtain the hybrid poly(urethane-imide)-based poly(ionic liquids).

### 2.3. Characterization of the hybrid poly(ionic liquids)

Structures of the products were identified by Fourier-transform infrared (FTIR) and  $^{13}\text{C}$ -NMR spectroscopies. FTIR spectra were recorded on Perkin-Elmer Spectrum 100 spectrometer in ATR (attenuated total reflection) mode. The solid-state  $^{13}\text{C}$ -NMR spectra were obtained using Bruker Avance DRX-400 spectrometer at 400 MHz. Differential Scanning Calorimetry (DSC) thermograms were obtained by TA Instrument Q20 differential scanning calorimeter in  $-90$ - $170^\circ\text{C}$  range with heating rate  $10^\circ\text{C}/\text{min}$  under nitrogen. Thermogravimetric analysis (TGA) was performed by TA Instrument SDT-Q600 between  $25^\circ\text{C}$  and  $600^\circ\text{C}$  at  $10^\circ\text{C}/\text{min}$  heating rate in nitrogen atmosphere. Tensile tests (stress  $\times$  strain curves) were obtained at  $25^\circ\text{C}$  with rectangular-shaped films ( $12 \times 7 \times 0.15$  mm) in DMTA equipment (model Q800, TA Instruments), 1 N/min. The Young modulus was determined according to ASTM D638. The analysis was conducted in triplicate. The field-emission scanning electron microscopy (FESEM) analysis was performed in FEI Inspect F50 equipment in the secondary electrons mode. The AFM analysis was performed in the peak force tapping mode by Bruker Dimension Icon PT equipped with TAP150A probe (Bruker, resonance frequency of 150 kHz and  $5 \text{ N m}^{-1}$  spring constant). The scanned area of the images was  $5 \times 5 \mu\text{m}^2$  with resolution 512 frames per area. The DMT modulus map was derived from PeakForce. Average roughness ( $R_a$ ), root mean square roughness ( $R_q$ ), and maximum height roughness ( $R_{\text{max}}$ ) were obtained from topography images (height).

### 2.4. Sorption measurements

The pressure-decay technique was used to determine  $\text{CO}_2$  sorption capacity at  $T = 303.15 \text{ K}$  and  $p_1 = 0.82$  and  $p_2 = 10$  bar. The dual-chamber gas sorption cell was similar to that of Koros et al [39]. Operation of such device was reported elsewhere [40,41]. Pressure in the sorption chamber and reservoir was measured with pressure transducer ( $\pm 0.01$  bar).

Samples ( $W_s = 1.0\text{-}1.5\text{ g}$ ) were weighted and transferred to the sorption chamber. The system was subjected to  $10^{-3}$  mbar vacuum at  $25^\circ\text{C}$  during 6 h. Afterwards,  $\text{CO}_2$  (air liquid/99.998%) was introduced into the reservoir at specified pressure and was allowed to enter the sorption chamber, it was partially dissolved in the sample.

Gas solubility at equilibrium was determined from the difference between the initial and final number of moles of gas using Eqs. 1-2, respectively. Mass of gas dissolved in the sample is denoted as  $W_{\text{CO}_2/g}$

$$n_{\text{CO}_2} = \frac{p_i V_{gc}}{Z(p_i, T_i) R T_i} - \frac{p_{eq} (V_t - V_p)}{Z(p_{eq}, T_{eq}) R T_{eq}}, \quad (1)$$

$$W_{\text{CO}_2/g} = \frac{n_{\text{CO}_2} M}{W_s}, \quad (2)$$

where  $V_{gc}$  is volume of the gas chamber,  $p_i$  and  $T_i$  are pressure and temperature in the gas chamber,  $p_{eq}$  and  $T_{eq}$  are pressure and temperature at equilibrium.  $V_c$  is the sorption cell total volume and  $Z$  is the compressibility factor for the pure gas calculated by the Span-Wagner equations-of-state for  $\text{CO}_2$  [42].

## 2.5. Computational Methodology

All binding energies and closest-approach distances were computed for the relaxed geometries. The geometry was considered relaxed when the energy difference between the two subsequent iterations in the LBGS procedure was inferior to  $10^{-5}$  Hartree [43]. The electronic structure was obtained from hybrid density functional theory (DFT), B3LYP [44,45], using the split-valence triple-zeta polarized basis set with diffuse functions, 6-311+G\*. The same basis set was employed to both relax the nuclear geometry and compute binding energy. The basis set superposition error was estimated using ghost atoms (counterpoise technique) and deducted from the reported results. Determination of each group's impact is useful for a thoughtful polymer design. Such analysis is currently available by means of computational chemistry only.

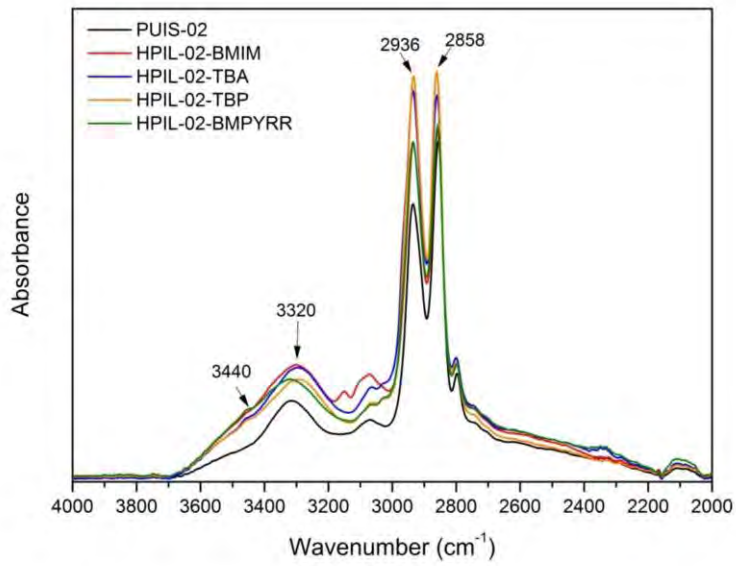
### 3. Results and discussion

#### 3.2. Polymer characterization

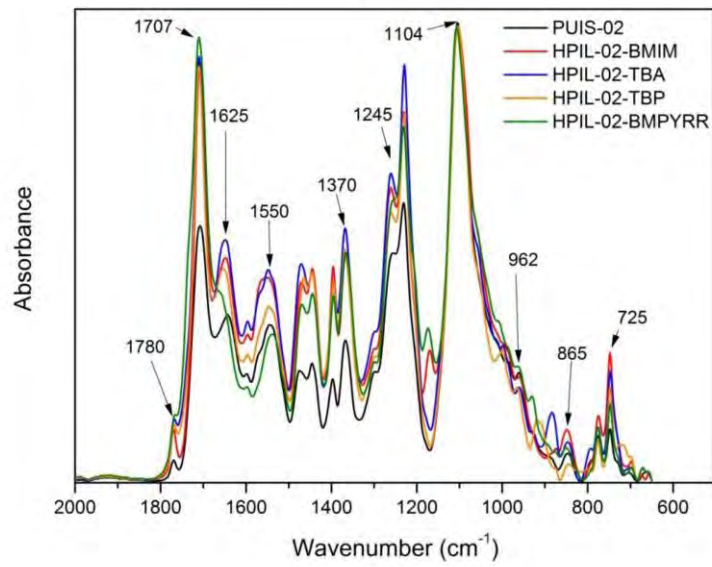
Fig. 2 - 3 shows FTIR spectra of the hybrid PU-imide and hybrid poly(ionic liquids). No band at around  $2270\text{ cm}^{-1}$  (N=C=O stretching vibration) indicates no free NCO group in the polymer structure [36,46].

The bands at  $1780\text{ cm}^{-1}$  and  $1707\text{ cm}^{-1}$  are attributed to the asymmetric and symmetric stretching vibration of C=O (imide groups), respectively. Therefore, the imide groups were successfully introduced into the PU chain [36,47,48]. The absorption band at  $1625\text{ cm}^{-1}$  corresponds to C=O of the urethane group [47,48]. After conversion from molecular to ionic material, the bands got wider, indicating that more carbonyl groups participate in H-bonding [30]. New band at  $962\text{ cm}^{-1}$  (O-Si-O) indicates reaction between -NCO of PU-imide and -NH<sub>2</sub> of APS [36]. The absorption bands at  $1370\text{ cm}^{-1}$  are attributed to C-N stretching, while the band at  $725\text{ cm}^{-1}$  corresponds to the imide ring deformation [36,46,47]. H-bonding was detected by FTIR (Fig. 2b - 3b). The N-H stretching band of the urethane group presents two main N-H absorption bands, one associated with non-bonded at  $3440\text{ cm}^{-1}$  and another with bonded hydrogen at  $3320\text{ cm}^{-1}$  [49,50]. The FTIR spectra contain a small shoulder at  $3440\text{ cm}^{-1}$  and a dominant absorption band at  $3320\text{ cm}^{-1}$ . This proves that the N-H groups participate in H-bonding [49,51]. The band area augmented at  $3320\text{ cm}^{-1}$  upon coordination of the polymer chain by the cations. This observation indicates formation of additional H-bonds between the cation and the anion (Fig. 2a - 3a).

The FTIR spectra (Fig. 2 - 3) also reveal urethane-specific bands:  $2936\text{ cm}^{-1}$  (C-H of CH<sub>2</sub>),  $2858\text{ cm}^{-1}$  (C-H of CH<sub>3</sub>),  $1550\text{ cm}^{-1}$  (H-N),  $1245\text{ cm}^{-1}$  (C-N and C-O of urethane),  $1104\text{ cm}^{-1}$  (C-O-C), and  $865\text{ cm}^{-1}$  (C-O-C) [28,30,52]. The C=O bands from the carboxylic acid group (COOH) and the urethane group (NHCO) are in overlay [51,53]. The asymmetric,  $1550\text{-}1610\text{ cm}^{-1}$ , and symmetric,  $1335\text{-}1420\text{ cm}^{-1}$ , stretching vibrations of COO are also present [54-56].



a)



b)

Figure 2. FTIR spectra of PUIS-02 and the corresponding HPILs; a) wavenumber from 4000 to 2000  $\text{cm}^{-1}$  and (b) wavenumber from 2000 to 650  $\text{cm}^{-1}$ .

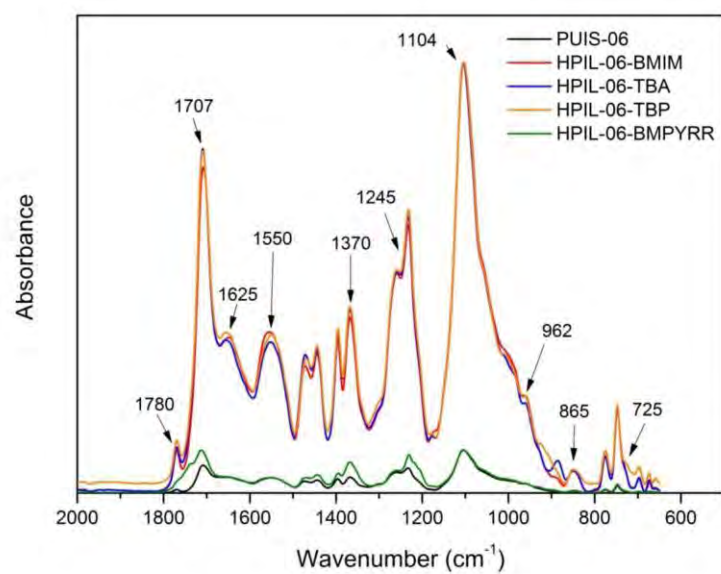
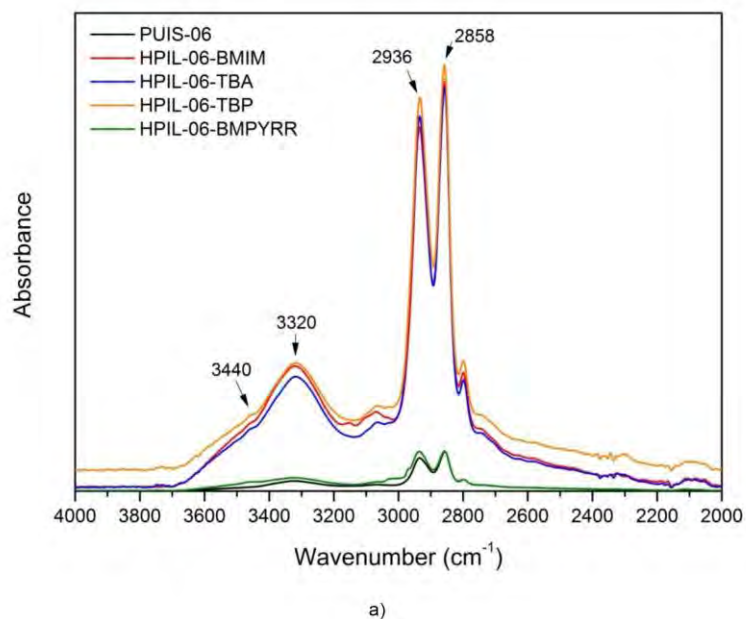


Figure 3. FTIR spectra of PUIS-06 and the corresponding HPILs; a) wavenumber from 4000 to 2000  $\text{cm}^{-1}$  and (b) wavenumber from 2000 to 650  $\text{cm}^{-1}$ .

Typical  $^{13}\text{C}$ -NMR spectrum of HPIL-06-TBP is shown in Fig. 4. Peaks at 167, 157, 156 ppm correspond to the carbonyl group and, thus, prove PU-imide polymer structure formation. The 167 ppm peak was assigned to

C=O of the imide groups. The 156 and 157 ppm peaks correspond to the carbonyl groups of urethane and urea, respectively [57,58]. Carbonyl groups of carboxylate of DMPA are responsible for the 175 ppm peak [58]. The aromatic carbon atoms of OPDA are revealed by a range of peaks, 120-173 ppm [59]. The 18 and 27 ppm peaks were attributed to CH<sub>3</sub> of DMPA and CH<sub>2</sub>-C of PTMG, respectively [58]. The 71 ppm peak reveals CH<sub>2</sub> of PTMG next to the urethane bonds [58]. The 64 and 50 ppm peaks correspond to CH<sub>2</sub>-O and C-CH<sub>2</sub>-O of DMPA, respectively [58]. The carbon atoms of HDI linked to urethane bond, N-CH<sub>2</sub>-CH<sub>2</sub>-CH<sub>2</sub>, were detected at 26, 31, and 41 ppm. Characteristic peaks of APS, -Si-CH<sub>2</sub>-CH<sub>2</sub>-CH<sub>2</sub>-N, are located at 14, 24, and 41 ppm [60,61]. The 18 ppm peak corresponds to the methoxy groups of APS [61]. The TBP cation, P-CH<sub>2</sub>-CH<sub>2</sub>-CH<sub>2</sub>-CH<sub>3</sub>, provides peaks at 14, 18, 23, 24 ppm. The TBA cation, N-CH<sub>2</sub>-CH<sub>2</sub>-CH<sub>2</sub>-CH<sub>3</sub> at 14, 18, 24, 59 ppm. Bmpyrr was detected by the following peaks: 14 (butyl-CH<sub>3</sub>), 18 (butyl-CH<sub>2</sub>), 21 (4-3 CH<sub>2</sub>), 26 (butyl-CH<sub>2</sub>), 48 (N-CH<sub>3</sub>), 63 (N-CH<sub>2</sub>); 64 ppm (5-2 N-CH<sub>2</sub>) [62]. Bmim was detected by the following peaks: 14 (butyl-CH<sub>3</sub>), 19 (butyl-CH<sub>2</sub>), 31 (butyl-CH<sub>2</sub>), 36 (N-CH<sub>3</sub>), 51 (N-CH<sub>2</sub>); 121 (C-4), 123 (C-5), 137 ppm (C-2) [63]. The peaks were assigned according to the Spectral Database for Organic Compounds[59].

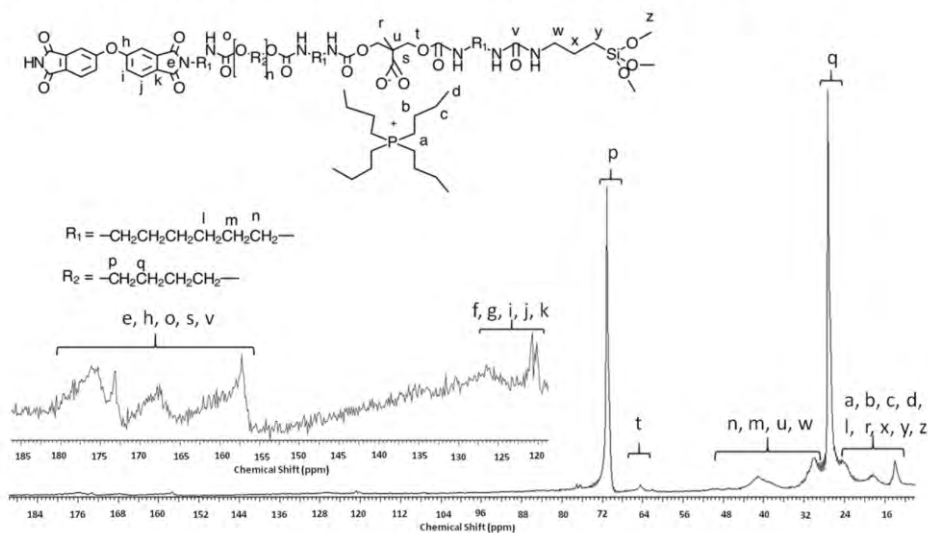


Figure 4. The solid-state <sup>13</sup>C-NMR spectra of HPUIS-06-TBP.



Surface morphologies were obtained by SEM (Fig. 5). A non-porous and relatively smooth morphology was observed in all films besides HPIL-06-TBA (Fig. 5h) and HPIL-06-BMPYRR (Fig. 5j) which present an irregular surface. This is due to cation (TBA and BMPYRR) on the polymeric chain of PUIS-06. Irregular surfaces may affect mechanical properties and, possibly, CO<sub>2</sub> sorption.

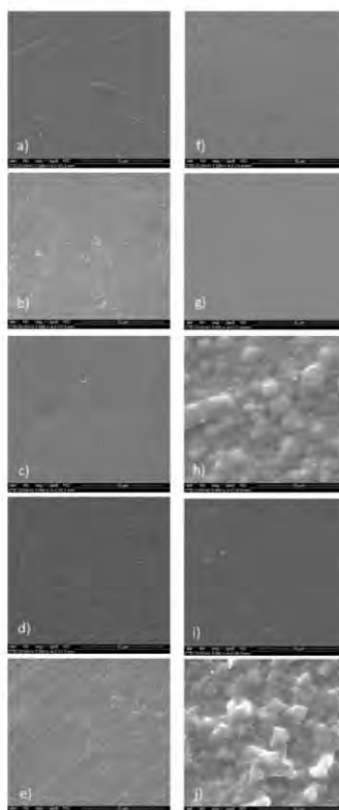


Figure 5. SEM Micrographs of the materials obtained with the 5000x magnification: (a) PUIS-02, (b) HPIL-02-BMIM, (c) HPIL-02-TBA, (d) HPIL-02-TBP, (e) HPIL-02-BMPYRR, (f) PUIS-06, (g) HPIL-06-BMIM, (h) HPIL-06-TBA, (i) HPIL-06-TBP, (j) HPIL-06-BMPYRR.

DSC thermograms for PU-imide and HPILs are shown in Fig. 6. PUIS-02, HPIL-02-BMIM, HPIL-02-BMPYRR, PUIS-06, HPIL-06-BMIM, and HPIL-06-BMPYRR provide endothermic peaks at +13.8, +7.8, -3.9, +11.3, +11.1, +8.5°C, respectively. These temperatures correspond to crystalline microphase melting ( $T_m$ ) [30,64–66]. HPIL-02-TBA, HPIL-02-TBP, HPIL-

06-TBA, HPIL-06-TBP exhibit a single glass transition temperature ( $T_g$ ): 11.0, 11.4, 10.4, 10.3°C, respectively. Thus, both silane and cation influence polymer chain mobility substantially. The increase of cross-linking, caused by an elevated silane content, reduces mobility of chains and limits crystallization of soft segments [67]. Reduction of  $T_m$  in HPIL-02-BMIM, HPIL-06-BMIM, HPIL-02-BMPYRR, HPIL-06-BMPYRR together with glass transitions in HPIL-02-TBA, HPIL-02-TBP, HPIL-06-TBA, HPIL-06-TBP suggest a strong interaction between the carboxyl group of the polymeric chain and the cations. This strong interaction results in the chain backbone organization. The cations containing a ring (BMIM and BMPYRR) foster crystallization of soft segments and chain backbone organization.

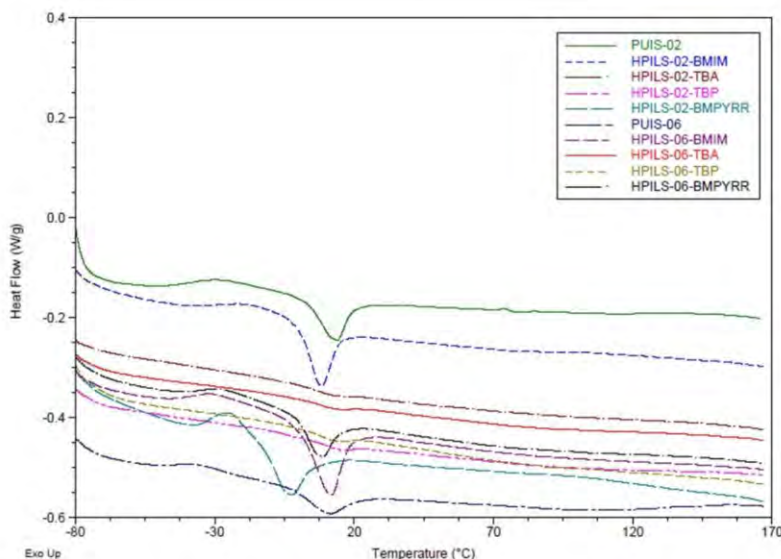


Figure 6. DSC thermograms for PUIS and hybrid poly(ionic liquids).

Thermogravimetric analysis (TGA) was performed to assess thermal stability of the synthesized materials (Table 1). All samples showed three thermal events (Fig. 7). The first event corresponds to PUIS and HPILs decomposition and occurs at about 175°C leading to around 5% mass loss. The mass loss is attributed to water evaporation, DMF, and residual low-molecular-weight monomers [57]. The second event is attributed to

urethane linkages decomposition [68]. The third event is degradation of the imide rings and the organic moiety of APS [67,68].

Degradation temperatures increase somewhat at the higher silane content. Compare 208 and 358 °C in PUIS-02 to 228 and 385°C in PUIS-06. This is due to the increase of cross-linking at higher silane content which decreases chain mobility and increases thermal stability [57,67]. Ionic PUIS-based materials are less thermally stable than their neutral precursors. The same behavior was reported in other PU-based PILs [32]. The residue consists of an inorganic content of the material [67,69]. The residue content increases as the silane content increases. Compare 14.8 wt% in PUIS-02 to 18.7 wt% in PUIS-06. The residue content decrease in the ionic materials (Table 1). To recapitulate, PUIS is stable above 200°C, whereas HPILs exhibit systematically lower thermal stabilities.

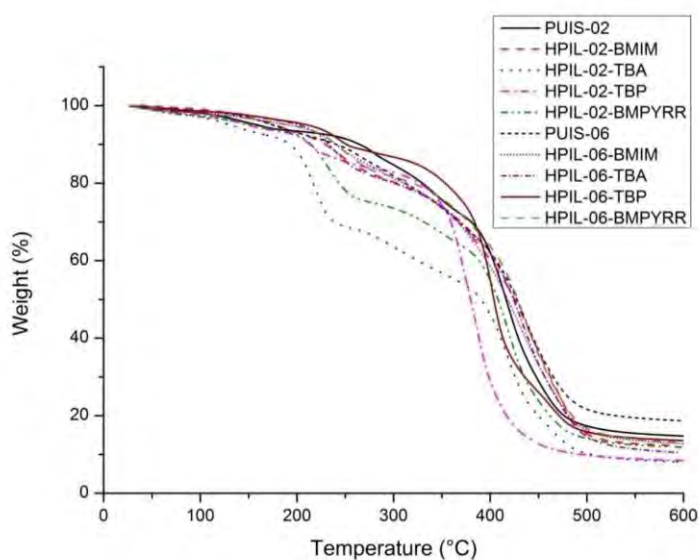


Figure 7. TGA thermograms for PUIS-02, PUIS-06 and the corresponding hybrid poly(ionic liquids).

Table 1. Results of thermal analyses of PUIS-2, PUIS-06 and the corresponding hybrid poly(ionic liquids).

Sample	Degradation stage	T <sub>onset</sub> (°C)	T <sub>endset</sub> (°C)	wt% remaining at 600°C
PUIS-02	2	208	358	
	3	358	524	14.8
HPIL-02-BMIM	2	194	300	
	3	300	572	12.1
HPIL-02-TBA	2	175	253	
	3	253	535	8.1
HPIL-02-TBP	2	184	268	
	3	268	522	8.5
HPIL-02-BMPYRR	2	178	276	
	3	276	555	11.8
PUIS-06	2	228	385	
	3	385	541	18.7
HPIL-06-BMIM	2	192	317	
	3	317	570	12.8
HPIL-06-TBA	2	185	289	
	3	289	580	10.4
HPIL-06-TBP	2	201	287	
	3	287	548	13.5
HPIL-06-BMPYRR	2	192	298	
	3	298	573	13.1

Surface morphology of the films was identified by atomic force microscopy (Figs. 8-9, Table 2). Surface structure changes occur with the increase of silane content and upon ionization of the material. One observes discontinuous lighter and darker regions in the images of DMT modulus. These patterns relate to microphase separation in the polymeric structure [70]. The microphase separation was observed upon silane content increase and upon the organic ionic introduction. Ionization of the materials fosters miscibility of hard and soft segments and, therefore,

improves microphase segregation [30,70]. Ionization is more important for microphase separation in the HPILs derived from PUIS-02. In turn, higher silane content influences microphase reorganization in HPILs. Height sensor images and roughness data of the surface suggest that the increase of the silane content reduces roughness of the films. Compare  $R_a = 37.1$  in PUIS-02 to  $R_a = 9.71$  in PUIS-06. Cross-linking improves film surface smoothness [70]. The roughness is smaller in the HPIL-02 series, as compared to their precursor, PUIS-02. The impacts of the cations show different strength:  $\text{BMPYRR} < \text{BMIM} < \text{TBP} < \text{TBA}$ . Film roughness in the HPIL-06 series is, however, larger than in PUIS-06 (Table 2).

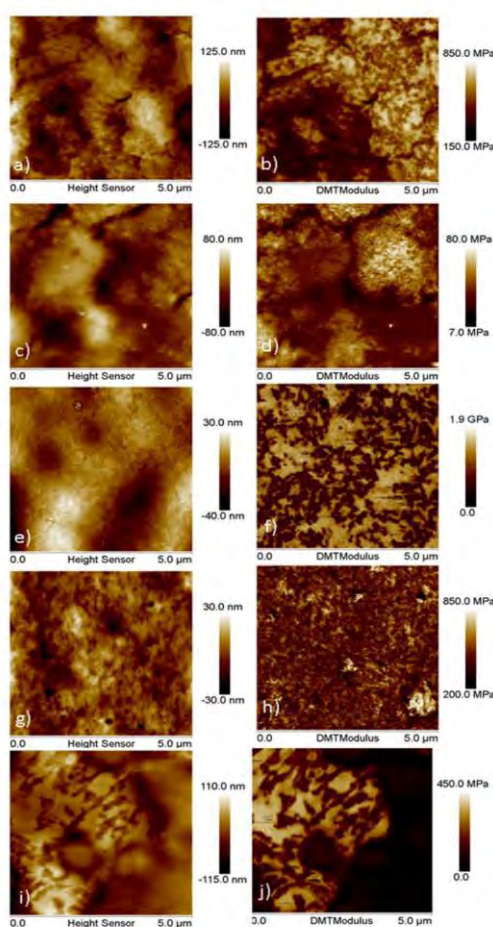


Figure 8. AFM height sensor images (left) and DMT modulus images (right): (a-b) PUIS-02, (c-d) HPIL-02-BMIM, (e-f) HPIL-02-TBA, (g-h) HPIL-02-TBP, (i-j) HPIL-02-BMPYRR.

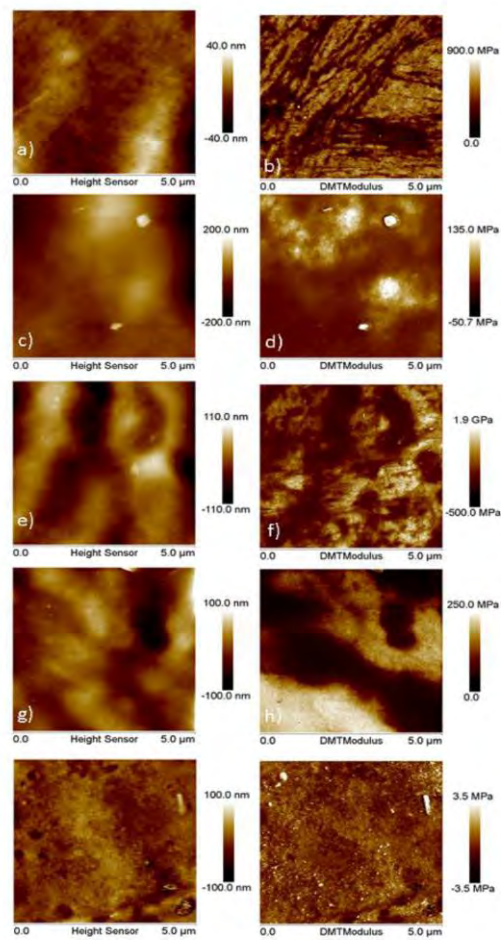


Figure 9. AFM height sensor images (left) and DMT modulus images (right): (a-b) PUIS-06, (c-d) HPIL-06-BMIM, (e-f) HPIL-06-TBA, (g-h) HPIL-06-TBP, (i-j) HPIL-06-BMPYRR.

Table 2. Average roughness ( $R_a$ ); root-mean-square roughness ( $R_q$ ); maximum height roughness ( $R_{max}$ ) for the prepared films

Sample	$R_a$ (nm)	$R_q$ (nm)	$R_{max}$ (nm)
PUIS-02	37.1	29.6	264.0
HPIL-02-BMIM	22.5	17.9	240.0
HPIL-02-TBA	10.6	8.2	92.2
HPIL-02-TBP	15.3	12.0	100.0
HPIL-02-BMPYRR	30.4	23.4	225.0
PUIS-06	9.71	7.32	74.2
HPIL-06-BMIM	34.9	19.3	498.0
HPIL-06-TBA	33.9	27.3	210.0
HPIL-06-TBP	27.2	21.2	271.0
HPIL-06-BMPYRR	55.9	44.8	521

Tensile properties and Young moduli (Figs. 10-11) indicate that higher silane content leads to larger Young moduli: 4.9 MPa in PUIS-02 and 16.8 MPa in PUIS-06. Therefore, a significant silane content is desirable to obtain competitive mechanical strengths and improve films rigidity [67,68,70]. Ionic materials exhibit better mechanical properties than their neutral precursors [30]. Furthermore, ionization significantly increases mechanical properties of HPIL-02 (56.4 MPa in HPIL-02-TBA). Ionization of PUIS-06 results in insignificant Young modulus increase, e.g. 21.8 MPa in HPIL-06-TBA.

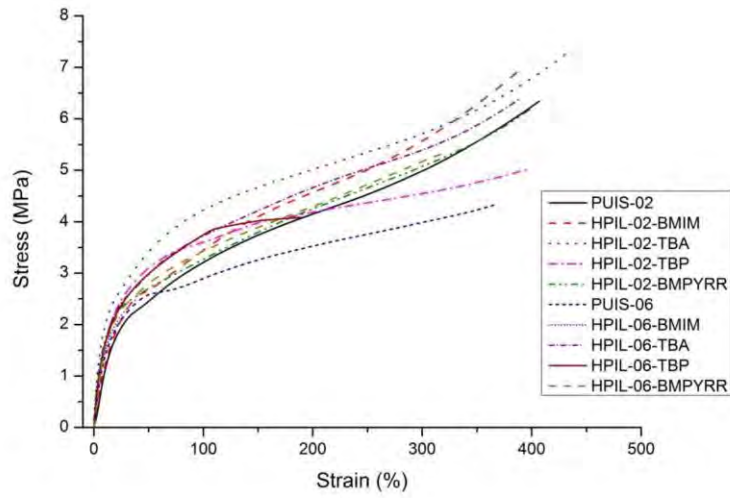


Figure 10. Stress/strain curves for the developed HPIL-based films.

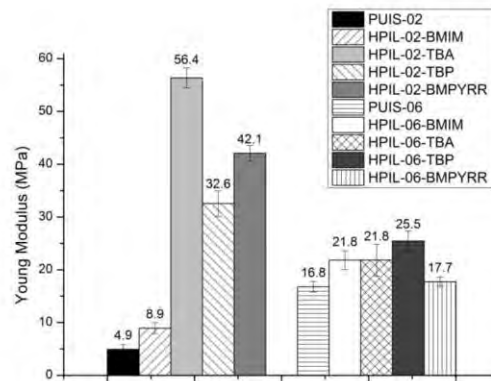


Figure 11. Young moduli for the developed HPIL-based films.

Figs. 12 describe CO<sub>2</sub> sorption by PUIS-02/PUIS-06 and the corresponding HPILs, whereas Table 3 compares them to the recently published results. The silane content increase in PU-imide leads to higher CO<sub>2</sub> sorption capacities. Compare 21.8 mg/g in PUIS-02 to 25.0 mg/g in



PUIS-06. Affinities to CO<sub>2</sub> of the ionized materials are even better. Ionization impact in PUIS-02 are more substantial than in PUIS-06. If cross-linking increases microvoid volume, due to a rigidity increase, this may complicate HPIL-CO<sub>2</sub> interactions [71].

The best CO<sub>2</sub> sorption values were obtained when TBA was used as a cation: 33.1 mg/g in HPIL-02-TBA and 31.7 mg/g in HPIL-06-TBA. Other cations (TBP, BMIM and BMPYRR) provide somewhat less efficient CO<sub>2</sub> sorption. This trend is in concordance with that observed in other PU-based PILs [30]. The electronic-structure simulations reveal that the cations exhibiting weaker coordination of the carboxyl group promote electrostatic binding of the CO<sub>2</sub> molecule and the carboxyl group [30].

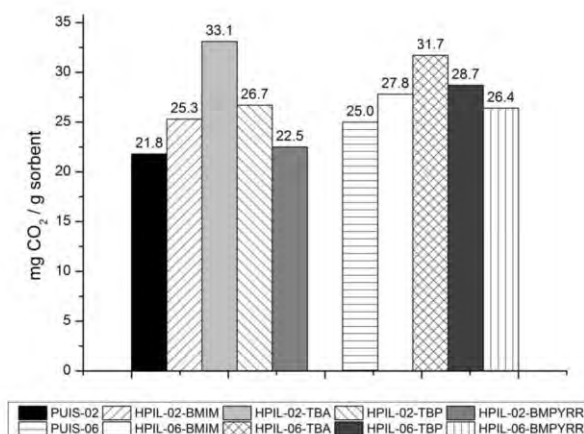


Figure 12. CO<sub>2</sub> sorption capacities of PUIS-2/PUIS-06 and the corresponding HPILs at 303.15 K and 0.82 bar.

Fig. 13 summarizes CO<sub>2</sub> sorption by PUIS-02 and the corresponding HPILs obtained at 10 bar and 303.15 K. The CO<sub>2</sub> sorption values are proportional to those at lower pressure. Again, the best result was recorded for the TBA containing HPILs, 61.0 mgCO<sub>2</sub>/g.

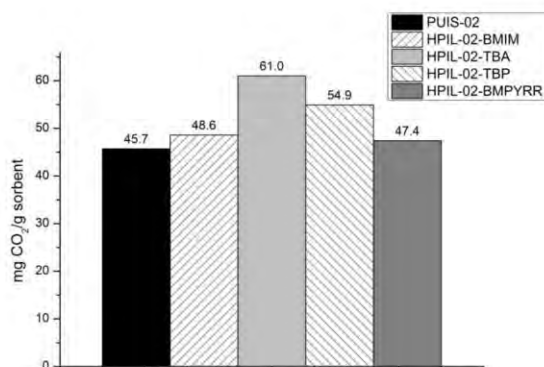


Figure 13. CO<sub>2</sub> sorption by the synthesized HPILs at 30°C and 10 bar.

At comparable temperatures and pressures, CO<sub>2</sub> sorption of developed HPILs are systematically higher (Table 3).

Table 3. Comparison of CO<sub>2</sub> sorption by HPIL-02-TBA with competitive poly(ionic liquids)

PIL	CO <sub>2</sub> sorption (mg/g)	Conditions (P, T)	Ref.
P[VBIT][BF <sub>4</sub> ] <sup>1</sup>	3.05	0.79 bar, 295.15K	[23]
P[VBIH][PF <sub>6</sub> ] <sup>2</sup>	3.22	0.79 bar, 295.15K	[23]
P([AMIM]BF <sub>4</sub> -AN) <sup>3</sup>	14.30	1 bar, 273.15K	[16]
P6 [BIEMA][Br] <sup>4</sup>	3.34	1 bar, 278.15K	[24]
P6 [BIEMA][acetate] <sup>5</sup>	12.46	1 bar, 278.15K	[24]
P[VBTEA][PF <sub>6</sub> ] <sup>6</sup>	14.04	1 bar, 278.15K	[17]
mpPIL	20.24	1 bar, 273.15 K	[21]
SiO <sub>2</sub> -meso-([VBTMA]-[BF <sub>4</sub> ])	17.7	1 bar, 303.15K	[6]
PIL-8.1.BF <sub>4</sub>	24.76	1bar, 273.15K	[31]
PU-TBP	15.70	0.82 bar, 303.15K	[30]
PU-TAB	16.10	0.82 bar, 303.15K	[30]
PUIS-02-TBA	33.1	0.82 bar, 303.15K	This study

<sup>1</sup>poly[1-(4-vinylbenzyl)-3-butylimidazolium tetrafluoroborate]; <sup>2</sup>poly[(1-(4-vinylbenzyl)-3-butylimidazolium hexafluorophosphate]; <sup>3</sup>poly(ionic liquid)s based on the copolymer of 1-allyl-3-methylimidazolium tetrafluoroborate and acrylonitrile; <sup>4</sup>Poly[2-(1-butylimidazolium-3-yl)ethyl methacrylate bromide]; <sup>5</sup>poly[2-(1-butylimidazolium-3-yl)ethyl methacrylate acetate]; <sup>6</sup>poly(4-vinylbenzyltriethylammonium hexafluorophosphate).

Reusability of the developed class of HPILs was characterized by subjecting HPIL-02-TBA to six sorption/desorption cycles. CO<sub>2</sub> sorption was carried out at 303.15 K and 0.82 bar. CO<sub>2</sub> desorption was carried out under vacuum at 298.15 K during 6 hours. Sorption capacity of HPIL-02-TBA increased insignificantly (61.0 mgCO<sub>2</sub>/g to 62.5 mgCO<sub>2</sub>/g) over the first cycles (probably due to the removal of moisture) and remained constant over the last cycles. These results indicate that HPIL-02-TBA sorbent offers necessary stability.

### 3.3. Binding energies

The PU-based anion and the cations constituting the new HPILs are expected to contain a number of groups for binding CO<sub>2</sub>. DFT calculations were used to identify the impact of those centers according to binding energies (Table 4). The most successful CO<sub>2</sub> capturing group is deprotonated carboxyl. Coordination of CO<sub>2</sub> by carboxyl brings -19 kJ mol<sup>-1</sup> to the system. Since there are two carboxyl groups per polymer unit, their total impact amounts to -38 kJ mol<sup>-1</sup> being noticeably higher than that of any other group. The imidazole group appeared also rather useful for CO<sub>2</sub> capture, -14 kJ mol<sup>-1</sup>, thanks to the acidic hydrogen atom of the imidazole ring. Two bmim cations are required to compensate for a negative charge of each polymer unit. Since the CO<sub>2</sub> capturing experiments reveal a better performance of the TBA and TBP containing HPILs, 33.1 and 26.7 mgCO<sub>2</sub>/g (Fig. 12), one concludes that imidazole decreases activity of carboxyl substantially, and vice versa. TBA and TBP belong to the non-coordinating cations, which do not block the carboxyl groups of the polyurethane anion. Interestingly, a somewhat poorer CO<sub>2</sub> affinity of TBP detected by the binding energy calculations is confirmed by the actual CO<sub>2</sub> capacities of the respective HPILs. In turn, the bmim and BMPYRR containing HPILs exhibit lower CO<sub>2</sub> capacities, 25.3 and 22.5 mg CO<sub>2</sub>/g. Hence, these cation-anion combinations should be considered less efficient. Although the ether and amide groups perform rather modestly, the

polymer units contains a number of such groups, therefore, their integral impact per mole of polymer units appears substantial.

The TBA and TBP cations exhibit a marginal affinity to CO<sub>2</sub>, -6.7 and -3.7 kJ mol<sup>-1</sup>, respectively. In BMPYRR, the primary electron-deficient center, nitrogen, is sterically blocked by two alkyl groups, while maintaining four covalent bonds with carbon atoms in total. This structure results in a relatively modest value, -9.3 kJ mol<sup>-1</sup>. Table 4 also summarizes binding energies of each polymer unit for the ease of further comparison, which is left as an exercise for an interested reader.

Table 4. Binding energies  $E_b$  and closest-approach distances  $r_{\min}$  for a variety of neutral and ionic moieties participating in the CO<sub>2</sub> capture. Decomposition into the interacting moieties was performed according to conventional chemical logics.

Chemical Moiety	$-E_b$ (10 <sup>-1</sup> eV)	$r_{\min}$ (Å) (atom pair)
-CH <sub>2</sub> -COO <sup>-</sup>	1.97	2.54 (C <sub>CO2</sub> -O)
-CH <sub>2</sub> -NH-CO-CH <sub>2</sub> -	0.93	2.89 (C <sub>CO2</sub> -O)
-CH <sub>2</sub> -N(CO) <sub>2</sub> -C <sub>6</sub> H <sub>3</sub> -	0.83	2.95 (C <sub>CO2</sub> -O)
-CH <sub>2</sub> -O-CH <sub>2</sub> -	1.04	2.75 (C <sub>CO2</sub> -O)
-C <sub>6</sub> H <sub>4</sub> -O-C <sub>6</sub> H <sub>4</sub> -	0.46	3.18 (C <sub>CO2</sub> -O)
-CH <sub>2</sub> -Si-(OCH <sub>3</sub> ) <sub>3</sub>	0.76	2.93 (C <sub>CO2</sub> -O)
bmim	1.45	2.31 (O <sub>CO2</sub> -H)
TBA	0.69	2.67 (O <sub>CO2</sub> -H)
TBP	0.38	3.24 (O <sub>CO2</sub> -H)
BMPYRR	0.96	2.62 (O <sub>CO2</sub> -H)

## Conclusions

We report new hybrid alkoxysilane-functionalized urethane-imide-based poly(ionic liquids) as a new efficient platform for CO<sub>2</sub> capture. The HPILs are expected to contain a number of functional groups to bind CO<sub>2</sub>. DFT calculations were used to identify the impacts of those centers. TBA and TBP are non-coordinating cations living the polyurethane anion carboxyl groups more available. HPILs containing these cations exhibit the

highest affinities to CO<sub>2</sub>. The best results were obtained for the TBA-based HPILs (33.1mg/g in HPIL-02-TBA and 31.7 mg/g in HPIL-06-TBA at 303.15 K and 0.82 bar). CO<sub>2</sub> sorption capacity of HPIL-02-TBA is the highest one from all reported poly(ionic liquids). Furthermore, the synthesized and characterized HPILs exhibit superior mechanical properties than anionic and cationic PU based PIL.

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### 4.3. Capítulo III: Poli(líquidos iônicos) aniônicos base celulose

O artigo intitulado “*New cellulose based ionic compounds as low-cost sorbents for CO<sub>2</sub> capture*” publicado na Fuel Processing Technology apresentado neste capítulo, descreve a obtenção de PLIs aniônicos celulósicos e sua avaliação para sorção de CO<sub>2</sub>. Para síntese dos PLIs utilizou-se celulose extraída da casca de arroz. A celulose foi modificada com ácido cítrico e funcionalizada com quatro diferentes cátions de líquidos iônicos (Cloreto de 1-butil-3-metilimidazólio; Brometo de tetrabutílfosfínio, Brometo de tetrabutílamônio e Cloreto de 1-butil-1-metilpirrolidínio). Os compostos obtidos foram caracterizados por FTIR, DRX, TGA, BET e MEV-FEG. A capacidade de sorção foi determinada utilizando uma microbalança termogravimétrica por Suspensão Magnética de alta precisão (PTGA), na temperatura de 25°C e pressões de 0,1 – 3 MPa. Estudos de simulações ab initio foram empregados para compreender o efeito do cátion e dos grupos funcionais presentes na estrutura da celulose modificada sobre a sorção de CO<sub>2</sub>. Estes compostos exibiram valores de sorção elevados, boa estabilidade térmica e capacidade de reutilização no processo de captura. O uso de resíduos para obtenção da celulose e a pequena quantidade de LIs utilizados na modificação permitiu a obtenção de sorventes de baixo custo para captura de CO<sub>2</sub>. Resultados experimentais e de simulação revelaram que o CO<sub>2</sub> interage de forma mais efetiva com o grupo carboxilato, e, em menor grau com os grupos polares da estrutura da celulose. O melhor desempenho foi obtido para o PLI CL-TBA (44 mgCO<sub>2</sub>/g em 0,1 MPa).



## Research article

New cellulose based ionic compounds as low-cost sorbents for CO<sub>2</sub> capture

Franciele L. Bernard<sup>a,b</sup>, Daniela. M. Rodrigues<sup>c</sup>, Barbara B. Polesso<sup>c</sup>, Augusto J. Donato<sup>b</sup>, Marcus Seferin<sup>a,b</sup>, Vitaly V. Chaban<sup>d</sup>, Felipe Dalla Vecchia<sup>c</sup>, Sandra Einloft<sup>a,b,\*</sup>

<sup>a</sup> Post-Graduation Program in Materials Engineering and Technology, Pontifical Catholic University of Rio Grande do Sul – PUCRS, Brazil

<sup>b</sup> School of Chemistry, Pontifical Catholic University of Rio Grande do Sul – PUCRS, Brazil

<sup>c</sup> School of Engineering, Pontifical Catholic University of Rio Grande do Sul – PUCRS, Av. Ipiranga, 6681, Partenon, Porto Alegre CEP: 90619-900, Brazil

<sup>d</sup> Institute of Science and Technology (ICT), Federal University of São Paulo (UNIFESP), São José dos Campos, SP, Brazil

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Quantum mechanical simulations

## ABSTRACT

Supporting ionic liquids in solid materials is a promising solution to address CO<sub>2</sub> capture. Cellulose is a versatile and low-cost material. The support of different ionic liquids cations on cellulose extracted from rice husk is an excellent strategy for CO<sub>2</sub> capture. This study reports chemical modification of cellulose fibers extracted from rice husk with different ionic liquids cations and its potential for CO<sub>2</sub> capture. The obtained ionic compounds were characterized by FTIR, XRD, FESEM, TGA and BET. CO<sub>2</sub> sorption capacity was gravimetrically assessed in a Magnetic Suspension Balance (MSB). In addition, ab Initio calculations were performed. First, the carboxylate groups are responsible for the CO<sub>2</sub> capture process. Second, the cation maintains the CO<sub>2</sub> binding sites of the anion partially or fully non-occupied, improving the CO<sub>2</sub>-ionic compound affinity. The best result for CO<sub>2</sub> sorption was obtained for the ionic compound CL-TBA of 71 mgCO<sub>2</sub>/g at 3 MPa and 25 °C. At lower pressures, this compound presents a high CO<sub>2</sub> sorption value (44 mgCO<sub>2</sub>/g at 0.1 MPa). We conclude that these ionic compounds deserve further attention in the post-combustion process.

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## 1. Introduction

An increasing level of carbon dioxide concentration in the atmosphere related to fossil fuels processing and utilization contributes to global warming considerably. Carbon capture and storage technologies appear among the key technologies to reduce CO<sub>2</sub> emissions and mitigate the effects of climate change [1]. Chemical absorption processes using aqueous amine solutions have been extensively studied and used in industry for decades. However, they have some operational drawbacks, such as large amount of energy required for the amine regeneration process [2,3], amines degradation/evaporation, and equipment corrosion [2–4]. Ionic liquids have been proposed as the next generation of solvents for a selective CO<sub>2</sub> separation [5]. These compounds are versatile and less harmful to the environment than conventional organic solvents [6–10]. They present unique properties, such as negligible vapor pressure, non-flammability, high thermal stability, and tunability (myriad of possible combinations of cations and anions [2,3,5]). Nevertheless, these solvents suffer from high viscosity and high production costs, when compared to aqueous amines solutions [5,11,12]. Implementing ionic moieties of ionic liquids into solid materials can provide a solution of these problems. This pathway has attracted the attention of researchers in the last decade. Solid sorbents,

typically studied for CO<sub>2</sub> capture are metal organic frameworks (MOFs) [1], mesoporous materials [13,14], activated carbon [15,16], polymeric and inorganic membranes [17], poly(ionic liquids) [18–20], covalent organic frameworks (COFs) [21], silica nanoparticles encapsulating amine-functionalized ionic liquids [22], etc. Functionalized cellulose containing amino groups has also been recently considered for CO<sub>2</sub> capture [23].

Cellulose is a versatile, low-cost material that undergoes chemical changes primarily by the hydroxyl groups [24]. Cellulose constitutes ca. 33% of rice husk [25], Brazil belonging to the 10 largest world rice producers [26]. The present utilization of the rice husk is far from being optimal [25,27]. Strategies to use these low-cost materials for CO<sub>2</sub> capture will foster emergence of green technologies and sustainable development.

The main aim of this study was to investigate chemical modification of cellulose fibers, extracted from rice husk, with different cations of ionic liquids and subsequent CO<sub>2</sub> capture of the developed materials. Quantum mechanical simulations were employed to reveal a role of each moiety of the ionic compounds in CO<sub>2</sub> capture.

## 2. Materials and methods

## 2.1. Materials

The rice husks were donated by Cooperativa Arrozadeira Extremo Sul Ltda. The obtained material was washed with distilled water and dried

\* Corresponding author at: Post-Graduation Program in Materials Engineering and Technology, Pontifical Catholic University of Rio Grande do Sul – PUCRS, Brazil.  
E-mail address: [einloft@pucrs.br](mailto:einloft@pucrs.br) (S. Einloft).

in an oven at 100 °C for 8 h. The dried husk was grounded in a knife mill. The fraction that passed through the 20 mesh sieve (0.841 mm) was collected. Anhydrous citric acid (CA, 99.5% Synth), sodium hydroxide (NaOH, 97% Vetec), hydrogen peroxide (H<sub>2</sub>O<sub>2</sub>, 35%, Neon), sulphuric acid (H<sub>2</sub>SO<sub>4</sub>, F. Maia, P. A), N,N-dimethylformamide (DMF, 99.5%, Merck) and ethanol (CH<sub>3</sub>CH<sub>2</sub>OH, 99%, Vetec) were used as purchased. Tetrabutylammonium bromide (TBAB, 99%, Acros Organics), tetrabutylphosphonium bromide (TBPB, 98%, Sigma Aldrich), and 1-butyl-1-methylpyrrolidinium chloride (BMPYRR, 99%, Fluka) were dried under vacuum at 60 °C for 12 h. 1-butyl-3-methylimidazolium chloride (BMIM Cl) was synthesized as described in literature [28–30]. BMIM Cl was characterized by Fourier transform infrared spectroscopy (FTIR), using a Perkin-Elmer spectrophotometer model Spectrum 100 FT-IR with full attenuated reflectance model (ATR), as well as by proton nuclear magnetic resonance (<sup>1</sup>H NMR) on a Varian Spectrophotometer, model VNMRs 300 MHz, using DMSO-*d*<sub>6</sub> as solvent, in accordance with the literature. <sup>1</sup>H NMR (300 MHz, DMSO-*d*<sub>6</sub>, 25 °C) δ (ppm): 1.01 (m, CH<sub>3</sub>), 1.29 (m, CH<sub>2</sub>CH<sub>3</sub>), 1.83 (m, CH<sub>2</sub>), 3.97 (s, CH<sub>3</sub>), 4.25 (t, CH<sub>2</sub>N), 7.79 (s, H5), 7.91 (s, H4), 9.48 (s, H2). FTIR  $\nu$  (cm<sup>-1</sup>): 3141 (N–H of imidazole group), 3058 (C–H of imidazole group), 2959 (C–H of methylene), 2870 (C–H of methyl), 1640 (C=N of imidazole group), 1558–1453 (C=C and C–N of imidazole group), 749 (Cl<sup>-</sup>).

## 2.2. Cellulose extraction

The cellulose extraction procedure was adapted from literature procedures [25,31]. The material was treated with 4 wt% NaOH at 90 °C for 2 h to remove lignin and hemicellulose of the rice husk fibers. The acid hydrolysis treatment was conducted on the fibers after alkali treatment using 10.0 mol/L H<sub>2</sub>SO<sub>4</sub> at 50 °C for 40 min under continuous stirring. Afterwards, bleaching was carried out by addition of 16 v/v% H<sub>2</sub>O<sub>2</sub> and 5 wt% NaOH solution in 1:1 proportion (v/v) at 55 °C for 2 h to remove the remaining lignin. The rice husk:solution ratio was 0.05 g/mL. Each step was repeated five times. The solid was filtered and washed with distilled water until neutral pH was achieved and dried in oven at 50 °C after each treatment. The detailed procedure used for determining cellulose content is available in Technical Association of Pulp and Paper Industry (TAPPI standard T203 OS-74). The extracted cellulose content from rice husk was 75 wt%.

## 2.3. Cellulose chemical modification

Cellulose chemical modification was performed in two stages. The first stage was modification with citric acid (Fig. 1). The second step was attachment of the cationic moiety (Fig. 2). The modification with citric acid was performed by a similar procedure described elsewhere [32,33]. The cellulose was poured into a flask containing 1.2 mol/L aqueous solution of citric acid to obtain 100 g L<sup>-1</sup> cellulose solution. This solution was stirred for 0.5 h at room temperature. The cellulose was separated by filtration and placed into an oven at 120 °C and kept constant for 1.5 h. After cooling, the cellulose was washed with distilled

water in proportion of 200 mL/g several times to remove any excess of citric acid and dried at 55 °C for 24 h. The concentration of carboxylic groups (mol·mg<sup>-1</sup>) introduced into the modified cellulose was determined by back titration [34,35], by the treatment of the material (0.1 g) with 100 mL of an aqueous solution of NaOH (0.02 mol/L) for 1 h under constant stirring. The materials were separated by filtration, and the solution was titrated with HCl (0.02 mol/L).

The functionalization reaction is described elsewhere [18]. In a Schlenk flask, under N<sub>2</sub> atmosphere, 1 g of citric-acid-modified cellulose was added to the ionic liquid (molar ratio COOH/IL = 1:1) dissolved in 15 mL of dimethylformamide (DMF). The mixture was stirred at 40 °C for 2 h. The materials were separated by filtration, washed repeatedly with distilled water and ethanol and dried at 55 °C for 24 h. The concentration of ionic liquid introduced into the modified cellulose was determined by back titration by treating the material (0.1 g) with 100 mL of an aqueous HCl solution (0.02 mol/L) for 1 h under constant stirring. Then, the materials were separated by filtration and the solution was titrated with NaOH (0.02 mol/L) (Fig. 2).

## 2.4. Characterization

The synthesized materials were characterized by Universal Attenuated Total Reflectance sensor (UATR-FTIR) using a Perkin-Elmer Model 100 FTIR Spectrum, in the range 4000–650 cm<sup>-1</sup>. The field emission scanning electron microscopy (FESEM) analyses were performed in FEI Inspect F50 equipment in secondary electrons (SE) mode. Thermogravimetric Analysis (TGA) was performed using a TA Instruments SDT-Q600 between 25 and 600 °C with a heating rate of 10 °C/min in nitrogen atmosphere. The specific surface area (SBET) and pore size were calculated from nitrogen sorption data using Brunauer-Emmett-Teller (BET) at -196 °C (Micromeritics, ASAP 2420). Powder X-ray diffraction (XRD) pattern was recorded on a Bruker-AXS D8 ADVANCE diffractometer operated at 40 kV and 20 mA using Cu K $\alpha$  radiation ( $\lambda = 1.5406 \text{ \AA}$ ) in the range (3–40°) with step of 0.02 and scanning time of 1.0 min. Eq. (1) was used to calculate the crystallinity index of the materials (CrI) by the Segal method [36].

$$\text{CrI} = \frac{I_{002} - I_{\text{AM}}}{I_{002}} \times 100 \quad (1)$$

where  $I_{002}$  is maximum intensity of the 002 peak and  $I_{\text{AM}}$  is intensity scattered by the amorphous part of the sample. The diffraction peak for plane (002) is located at diffraction angle around  $2\theta = 22^\circ$  and intensity scattered by the amorphous part is measured as the lowest intensity at diffraction angle around  $2\theta = 18^\circ$  [25].

## 2.5. Ab initio calculations

The binding energies and equilibrium geometries of ions and molecules were computed from hybrid density functional theory (HDFT) using Becke-3-Lee-Yang-Parr (B3LYP) approach [37,38]. The atom-

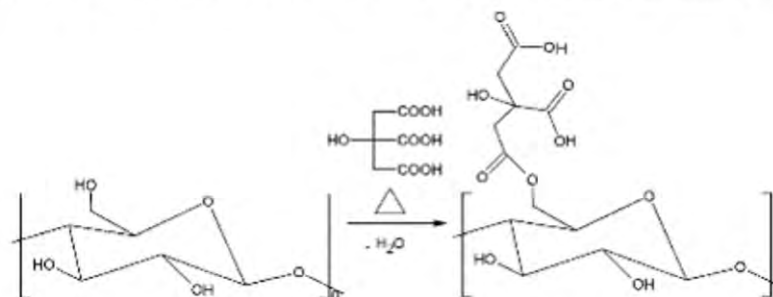


Fig. 1. Schematic representation of the carboxylic groups introduced onto the cellulose surface.

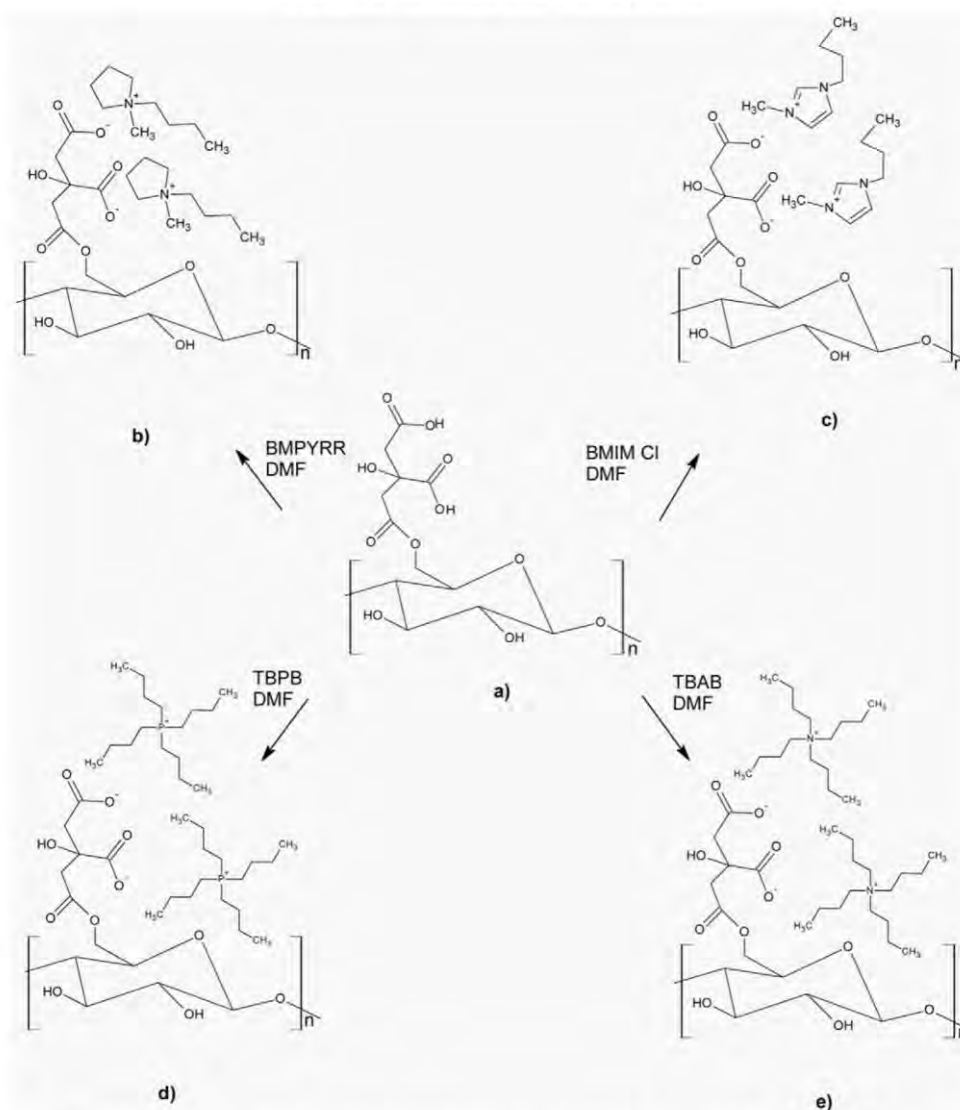


Fig. 2. Ionic compounds obtained from cellulose modification: (a) CL-CA; (b) CL-BMPYRR; (c) CL-BMIM; (d) CL-TBPB; (e) CL-TBA.

centered split-valence double-zeta polarized 6-31G(d) basis set was used for all atoms (carbon, hydrogen, oxygen) at all stages of the calculations. The basis set superposition errors (BSSE) were estimated by the counterpoise method and deducted from the reported binding energies. The self-consistent field convergence criterion was set to  $10^{-8}$  Eh. The geometry convergence criterion was set to  $0.12 \text{ kJ mol}^{-1} \text{ nm}^{-1}$  for the largest force. The successful location of the stationary point (local minimum) was ensured by analyzing vibrational frequencies.

## 2.6. Gas adsorption measurements

Pure gas sorption ( $\text{CO}_2$ ) in the samples were gravimetrically assessed in a Magnetic Suspension Balance (MSB) (Rubotherm Prazisionsmesstechnik GmbH, 35 MPa and  $400^\circ\text{C}$ ) equipped with a single sinker device for absorbate density determination and

thermostated with an oil bath (Julabo F25/ $\pm 1^\circ\text{C}$ ). The apparatus details are described elsewhere [39–41]. When compared to other gravimetric sorption methods, the MSB device allows high-pressure sorption measurements since the sample can be potted into a closed chamber coupled to an external accurate balance (accuracy of  $\pm 10 \mu\text{g}$ ). The samples (0.06 to 0.09 g) were weighted and transferred to the MSB sample container. The system was subjected to a  $10^{-7}$  MPa vacuum at the temperature of the sorption measurement,  $25^\circ\text{C}$ , for 24 h (constant weight was achieved within this time). The gas ( $\text{CO}_2$ , Air Liquide/99.998%) was admitted into the MSB pressure chamber up to the desired pressure, 0.1–3 MPa. In this study, pressure gauge with an accuracy of  $10^{-3}$  MPa was used to control system pressure. The gas solubility in the samples for each isotherm and pressure considered was measured for 3–4 h until no more weight increasing for gas sorption was observed. At this step of gas solubility in the samples, the weight reading from



**Table 1**  
Specific surface area, ionic liquid loading and crystallinity.

Samples	SBET (m <sup>2</sup> /g)	Ionic liquid loading (mol)/mg	Crystallinity (%)
Cellulose	0.2398	–	87
CL-CA	0.4575	–	84
CL-BMIM	0.6984	$1.57 \times 10^{-6}$	89
CL-TBP	0.6430	$1.16 \times 10^{-6}$	88
CL-TBA	1.7370	$0.68 \times 10^{-6}$	87
CL-BMPYRR	0.5901	$0.95 \times 10^{-6}$	89

the microbalance at pressure  $P$  and temperature  $T$  was recorded as  $W_t(P, T)$ . The mass of absorbed gas in the sample ( $W$ ) was calculated using the following (Eq. (2)):

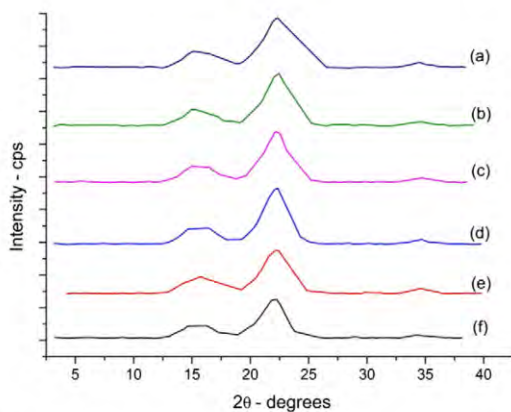
$$W = [W_t(P, T) - W_{sc}(P, T) + \rho_g(P, T) \cdot (V_{sc}(T) + V_s(T))] - W_s(\text{vac}, T) \quad (2)$$

where  $W_{sc}(P, T)$  is the weight of sample container,  $\rho_g(P, T)$  stands for gas density, directly measured with the MSB coupled single-sinker device,  $V_{sc}(T)$  is the volume of the sample container, determined from a buoyancy experiment when no sample is charged into the sample container,  $V_s(T)$  the specific solid sample volume,  $W_s(\text{vac}, T)$  is the weight of samples under vacuum and the term  $\rho_g(P, T) \cdot (V_{sc}(T) + V_s(T))$  represents the buoyancy force. In this work, the solubility in the sample was treated as excess solubility.

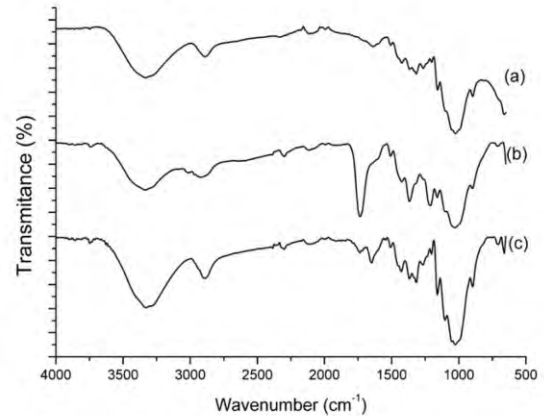
### 3. Results

#### 3.1. Specific surface area measurements

The determined concentration of carboxylic functions of CL-CA sample was  $2.15 \times 10^{-6}$  (mol/mg); this result is close to the previously known investigations [42]. Table 1 shows surface area data, ionic liquid loading and crystallinity for each sample. Even if the presence of the counteranion increases surface area, the obtained values for all ionic materials are small. We highlight that the amount of ionic liquid loaded is low, indicating the possibility of the capture process cost reduction. Crystallinity value obtained in the samples is discussed in Section 3.2.



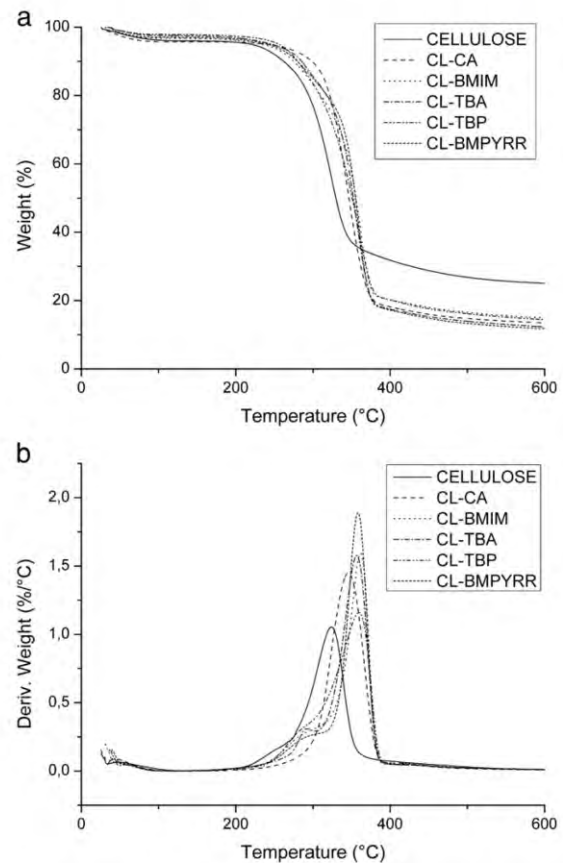
**Fig. 3.** XRD pattern of cellulose and modified cellulose samples: (a) CL-TBP; (b) CL-TBA; (c) CL-BMIM; (d) CL-BMPYRR; (e) cellulose; (f) CL-CA.



**Fig. 4.** FTIR spectra of: (a) cellulose, (b) cellulose modified with citric acid and (c) ionic compound CL-BMPYRR.

#### 3.2. X-ray diffraction (XRD)

Fig. 3 shows X-ray diffraction profile of the synthesized cellulose based ionic compounds and that of the extracted cellulose. All samples exhibit typical cellulose peaks around  $2\theta = 16^\circ, 22.6^\circ$  and  $35^\circ$  [25].



**Fig. 5.** (a) TG and (a) DTG curves for cellulose and modified cellulose samples.

The crystallinity index (CrI) of the cellulose was ca. 87%, whereas for the sample CL-CA CrI was ca. 84%. The presence of the counter cation does not affect considerably the crystallinity of the samples when compared with cellulose and CL-CA samples (Table 1).

### 3.3. Spectroscopic characterization

The FTIR spectra of all samples (Fig. 4) showed a broadband located at 3330–3340  $\text{cm}^{-1}$ , attributed to the stretching of —OH groups, band near 2920  $\text{cm}^{-1}$  related to the C—H stretching, band at 1160  $\text{cm}^{-1}$  corresponding to C—O—C asymmetrical bridge stretching, broadbands around 1050  $\text{cm}^{-1}$  and 890  $\text{cm}^{-1}$  related to the C—H and C—O stretching vibration of the cellulose structure [25,27]. Having compared spectra for the citric acid modified cellulose (Fig. 4b) and cellulose (Fig. 4a), we identified a strong band at 1740  $\text{cm}^{-1}$  attributed to the carboxylic acid (—COOH). This evidences introduction of citric acid into the cellulose chain [24].

Fig. 4c presents FTIR for the ionic compound CL-BMPYRR. A new band at 1648  $\text{cm}^{-1}$ , which is present in all ionic compounds, suggests

formation of carboxylates. In turn, band at 1740  $\text{cm}^{-1}$  corresponds to the unreacted carboxylic acid [24,43].

### 3.4. Thermogravimetric Analysis (TGA)

The Thermogravimetric Analysis (TGA) was performed in order to characterize thermal stability of the synthesized material. Fig. 5 presents the TG and DTG curves obtained for all materials. All samples showed two thermal events. The earlier weight loss was ca. 4% and  $T_{\text{onset}}$  below 100 °C attributed to water vaporization [25]. The later weight loss is attributed to cellulose degradation ( $T_{\text{onset}}$  of 289 °C and  $T_{\text{max}}$  of 324 °C), being 71%. The modified samples presented a higher thermal stability than cellulose (citric acid modified samples  $T_{\text{onset}} = 318$  °C and  $T_{\text{max}} = 346$  °C; samples CL-BMIM, CL-TBP, CL-TBA and CL-BMPYRR presented values of  $T_{\text{onset}} = 330$  °C and  $T_{\text{max}} = 362$  °C;  $T_{\text{onset}} = 314$  °C and  $T_{\text{max}} = 359$  °C,  $T_{\text{onset}} = 324$  °C and  $T_{\text{max}} = 357$  °C;  $T_{\text{onset}} = 333$  °C and  $T_{\text{max}} = 358$  °C, respectively) and weight loss around 84%. This behavior is associated with the reduction of the cellulose weight as a result of the chemical modification with citric acid and the ionic

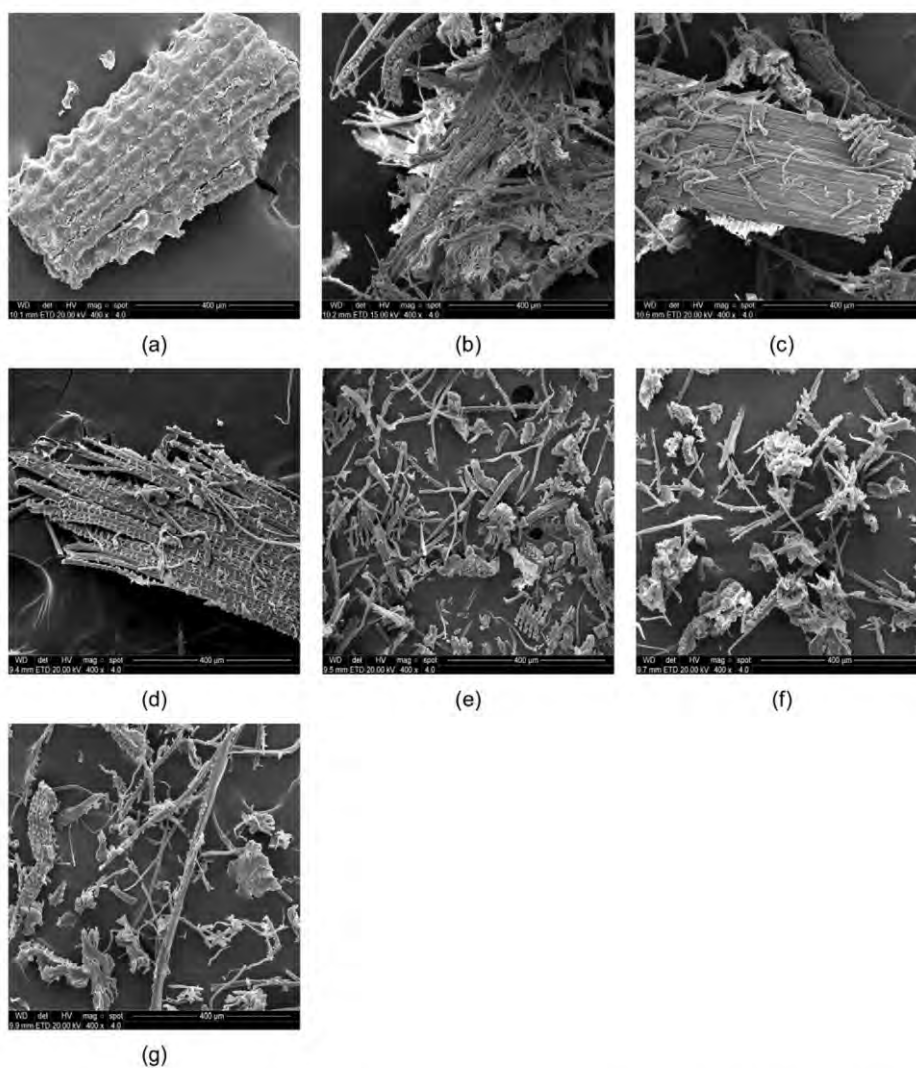


Fig. 6. Micrographs of the materials: (a) rice husk fibers, (b) cellulose, (c) CL-CA, (d) CL-BMIM, (e) CL-TBA, (f) CL-TBP, (g) CL-BMPYRR.

liquids cations. When comparing the DTG curves (Fig. 5b) of cellulose and the acid citric modified sample with the ionic compounds (CL-BMIM; CL-TBP; CL-BMPYRR and CL-TBA), a new shoulder is observed, possibly due to the anion–cation interaction.

### 3.5. Field emission scanning electron microscopy (FESEM)

Using FESEM, Fig. 6 exemplifies different effects on the rice husk surface after cellulose extraction as well as chemical modification. Fig. 6a shows the smooth compacted surface of the untreated rice husk fiber. Fig. 6b shows the same after bleaching. Cellulose exists in the form of bundles of fibers. The FESEM micrograph of chemically modified cellulose fibers with citric acid (Fig. 6c) showed similar morphology to that shown by the sample after bleaching treatment (Fig. 6b). The chemical modification of cellulose with the ionic liquid led to separation of the fiber bundles into individual fibers and provided an apparent reduction of the average size of cellulose particles (see Fig. 6 e, f and g) with the exception for BMIM cation (see Fig. 6d). This behavior might contribute to the increase of CO<sub>2</sub> adsorption, due to the increased contact surface.

### 3.6. Quantum mechanical simulations

The BMPYRRR, TBP and TBA cations do not coordinate CO<sub>2</sub>. They are used as weakly coordinating counter-ions for the modified cellulose. The cations of this sort are expected not to form strong cation–anion bonds and, therefore, to maintain the CO<sub>2</sub> binding sites of the anion partially or fully non-occupied. In turn, the BMIM cation coordinates CO<sub>2</sub> thanks to a weak hydrogen bond originating from the intrinsically acidic hydrogen atom of the imidazole ring and the oxygen atom of CO<sub>2</sub>. The CO<sub>2</sub> capture ability of BMIM was thoroughly previously investigated [11,44,45]. An elementary unit of the polymeric anion possesses 11 oxygen atoms constituting eight prospective binding sites (Fig. 7): two carboxylate groups, three hydroxyl groups, two ether groups, and one ester group. Such an abundance of coordinating sites (polar and relatively groups) attracts an interest to the cellulose derived anions. Somewhat different chemical environment permits to consider the mentioned groups as unique sites in relation to CO<sub>2</sub>. For systematicity, we used the tetramethylammonium cation to represent a weakly coordinating organic cation in all calculations.

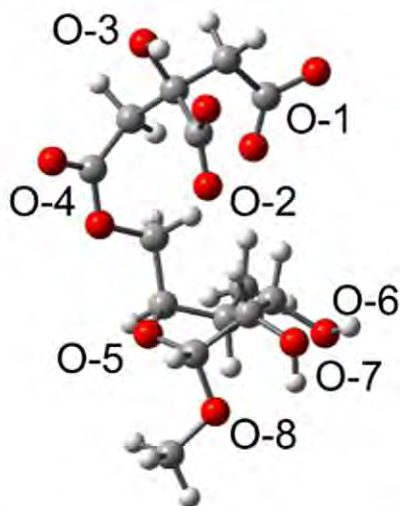


Fig. 7. The CO<sub>2</sub> binding sites of the modified cellulose moiety: carbon atoms are grey, hydrogen atoms are white, oxygen atoms are red. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

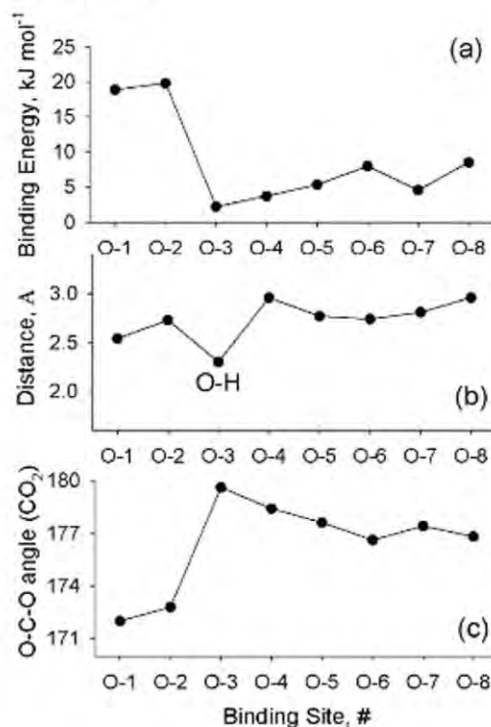


Fig. 8. Primary characteristics of the CO<sub>2</sub> physisorption process: (a) binding energy; (b) closest-approach distance; (c) CO<sub>2</sub> deviation from its genuine linearity due to binding. In most cases, the closest-approach distance is carbon–oxygen, except at O-3, in which oxygen of CO<sub>2</sub> coordinates hydrogen of hydroxyl.

Specific contributions of the O-x, where x = 1...8, sites cannot be clearly derived from the experiment, but can be obtained from the HDFT calculations in terms of binding energies, closest-approach distances, and selected characteristic angles (Fig. 8, Table 2). We investigated 8 local minima by initially putting CO<sub>2</sub> in the vicinity of the desired binding site and applying a geometry optimization procedure to locate a stationary point for every configuration. The strongest CO<sub>2</sub> binding is observed at the O-2 site (see Fig. 9). This is due to the simultaneous attraction to the carboxyl and hydroxyl groups of the citric acid moiety and despite steric hindrance preventing a tight coordination of the carbon atom of CO<sub>2</sub> by the carboxyl group. Compare  $r(\text{C}-\text{O}) = 2.54 \text{ \AA}$  (O-1) to  $r(\text{C}-\text{O}) = 2.73 \text{ \AA}$  (O-2). The ester, ether and hydroxyl groups, per se, perform poorly and the respective binding energy in no case exceeds  $10 \text{ kJ mol}^{-1}$ . In the meantime, the ether groups, O-5 and O-8, appear somewhat more successful among others. This conclusion is in concordance with the recent experimental study [46], in which

Table 2

The oxygen (CO<sub>2</sub>)–carbon (cellulose) distances, carbon (CO<sub>2</sub>)–oxygen (cellulose)–carbon (cellulose) non-valence angles, BSSE, BSSE percentage with respect to the computed binding energy.

Site, #	$r(\text{O}-\text{C}), \text{ \AA}$	angle (C–O–C), °	BSSE, kJ mol <sup>-1</sup>	BSSE, %
O-1	4.0	140.7	14.8	44.0
O-2	3.7	120.9	17.0	46.2
O-3	3.9	102.6	4.60	67.1
O-4	3.9	131.4	7.52	66.9
O-5	3.4	120.2	8.57	61.6
O-6	3.7	125.9	8.52	51.6
O-7	3.3	108.9	9.57	67.8
O-8	3.9	135.3	15.3	64.1

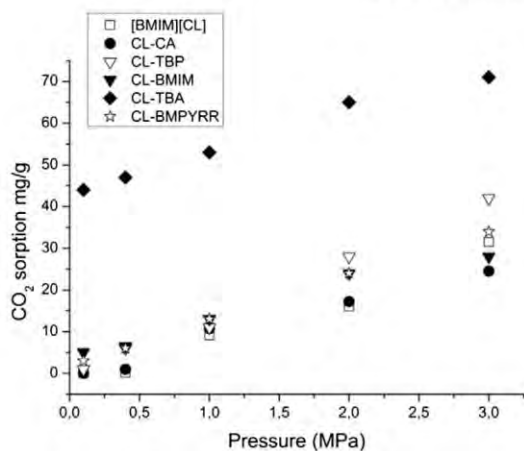


Fig. 9. CO<sub>2</sub> sorption for the ionic compounds, cellulose and the ionic liquid [BMIM] [Cl].

higher density of carboxylate and ether groups per adsorbent particle was linked to better CO<sub>2</sub> capture. Note that chemical environment matters significantly in the process of gas molecule binding and the impact of neighboring groups cannot be fully isolated. For instance, oxygen of CO<sub>2</sub> coordinates hydrogen of hydroxyl at O-3; nevertheless, this pattern is not implemented either in O-6 or O-7 (also hydroxyl group containing sites of the modified cellulose). The reason for that is formation of intramolecular hydrogen bond between the hydroxyl groups at the O-6 and O-7 positions. Consequently, both groups are located above the ring (*cis*-conformation) and are insufficiently available to oxygen of CO<sub>2</sub>. Compare the depicted closest-approach distances in Fig. 8b.

The CO<sub>2</sub> molecule in vacuum is linear, with the O-C-O angle equaling to 180°. Binding perturbs an original shape of CO<sub>2</sub>, the magnitude of perturbation being proportional to the binding energy (Fig. 8). Whereas the O-C-O angle is 172° at O-2 and 173° at O-1, this same angle exceeds 177° in all other cases. The weakest binding, at O-3, coincides ideally with the most linear CO<sub>2</sub> shape.

The BSSEs constitute a significant portion of the binding energies (Table 2). Their contribution increases when the binding energy decreases and becomes less electrostatic in nature. See the cases of hydroxyl and ether groups. Using a larger basis set generally favors smaller BSSE, which nevertheless needs to be estimated and deducted.

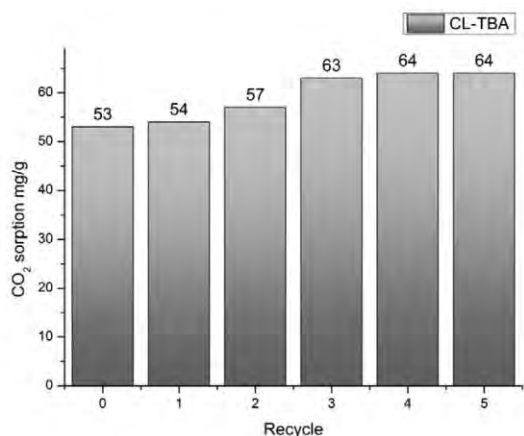


Fig. 10. CO<sub>2</sub> sorption/desorption tests for the ionic compound CL-TBA.

### 3.7. Gas sorption measurements

The experimental results of CO<sub>2</sub> sorption by the synthesized materials are shown in Fig. 9. The sorption behavior for the ionic liquid [BMIM] [Cl] at the same experimental conditions is also presented. Note that in Fig. 9 the CO<sub>2</sub> sorption values for modified cellulose with citric acid (CL-CA) were similar to those found for the ionic liquid [BMIM] [Cl]. However, the sorption of CO<sub>2</sub> was slightly higher in the CL-BMIM sample, suggesting that the system [BMIM] cation-cellulose improves the affinity with CO<sub>2</sub>. As suggested by the quantum mechanical simulations, the CO<sub>2</sub> interacts quite strongly with the cellulose chain. The effect of the counteraction in increasing CO<sub>2</sub> sorption is evident when comparing the results for CO<sub>2</sub> sorption of the ionic compounds (CL-BMIM; CL-TBP; CL-BMPYRR and CL-TBA) with the sample without counteraction (CL-CA). These results suggest that the solubility of CO<sub>2</sub> in these materials is mainly influenced by the interaction of the counteraction with the cellulose chain and by the surface area (CL-TBP 0.6430 m<sup>2</sup>/g and CL-TBA 1.7370 m<sup>2</sup>/g). An amount of ionic liquid loading (Table 1) is weaker. Quantum mechanical simulations evidenced that the CO<sub>2</sub> interacts with the cellulose chain mainly by O-1 and O-2. Hence, the cation plays a stabilizing role in the ionic compound system. The weakly coordinating counteractions for the modified cellulose stabilize the system maintaining the CO<sub>2</sub> binding sites of the anion partially or fully non-occupied, increasing CO<sub>2</sub>-philicity of the ionic compound. The best result for CO<sub>2</sub> sorption was obtained for the ionic compound CL-TBA of 71 mgCO<sub>2</sub>/g at 3 MPa and 25 °C, but as depicted in Fig. 9, even at lower pressures, this compound presents a high CO<sub>2</sub> sorption value (44 mgCO<sub>2</sub>/g at 0.1 MPa). The low surface area values explain this behavior. The sorption capacity at low pressures is dependent on the CO<sub>2</sub> interaction with the ionic compound, while at higher pressures the low specific surface area do not facilitate the CO<sub>2</sub> sorption. The usage of industrial wastes, such as fly ash derived materials for CO<sub>2</sub> sorption, are described elsewhere [47]. Maroto-Valer et al. [48] reported different materials obtained by modification of fly ashes (AC) aiming to introduce chemical adsorption sites as well as evaluate their sorption capacity. The specific surface areas of the products are considerably higher when compared to our results (AC untreated 818 m<sup>2</sup>/g; AC treated with MEA 241 m<sup>2</sup>/g and cellulose 0.2398 m<sup>2</sup>/g; CL-TBA 1.7370 m<sup>2</sup>/g). Even with low specific surface areas, the results obtained for CO<sub>2</sub> sorption of AC treated with MEA varied from 68.6 mg CO<sub>2</sub>/g sorbent to 5.5 mgCO<sub>2</sub>/g depending on the MEA content. When AC was modified with MDEA and DEA, lower values were found when compared to MEA (30 °C, at ca. 0.240 MPa with flow rate of 100 mL/min). These values are lower, as compared with those found in the present work.

### 3.8. Recycling experiments for CL-TBA

In order to examine the reusability of PILs, CL-TBA was subjected to five more runs. The sorption was carried out at 25 °C under a 1 MPa CO<sub>2</sub> pressure. After each step, the CO<sub>2</sub> desorption was carried out under vacuum at 25 °C for 8 h. The results for the six consecutive tests are shown in Fig. 10. The CO<sub>2</sub> sorption capacity was slightly increased in the first three recycles, probably due to impurities withdrawal remaining constant for the next cycles. These results indicate that the CL-TBA has high stability and capacity of being reused.

## 4. Conclusions

The chemical modification of cellulose extracted from rice husk with different ionic liquid cations was performed and the potential for CO<sub>2</sub> capture of these materials was evaluated. The quantum mechanical simulations and the experimental results revealed the role of each moiety of the ionic compounds in the gas capture process. The abundance of coordinating sites attracts an interest to the cellulose-derived anions, since CO<sub>2</sub> interacts with the cellulose chain. The presence of the counteraction in increasing the CO<sub>2</sub> solubility was evidenced and the best result

for CO<sub>2</sub> sorption was obtained for the ionic compound CL-TBA of 71 mgCO<sub>2</sub>/g at 3 MPa and 25 °C. Weakly coordinating counterions for the modified cellulose stabilize the system maintaining the CO<sub>2</sub> binding sites of the anion partially or fully non-occupied, improving the CO<sub>2</sub>-philicity of the ionic compound. Our results present a new sustainable CO<sub>2</sub> capture option that can add value to a waste being also affordable to be used in post-combustion process. However, the sorbents should be further tested under more representative post-combustion conditions.

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#### 4.4. Capítulo IV: Poli (líquidos iônicos) catiônicos base celulose

O artigo intitulado “*Cellulose based poly(ionic liquids): Tuning cation-anion interaction to improve carbon dioxide sorption*” submetido a Fuel (comprovante de submissão Anexo B), descrito neste capítulo apresenta a produção de PLIs catiônicos a partir da celulose extraída da casca de arroz. Foi investigado o efeito da combinação de dois cátions (imidazólio e o amónio) com quatro diferentes contra-ânions ( $[Cl^-]$ ,  $[BF_4^-]$ ,  $[PF_6^-]$  ou  $[Tf_2N^-]$ ). Os PLIs sintetizadas foram caracterizados por RMN, XPS, FTIR, TGA e MEV-FEG. A capacidade de sorção de  $CO_2$  avaliada pela técnica de decaimento da pressão a 298,15 K e pressões de 0,1 - 3,0 Mpa. A capacidade de sorção/dessorção também foi investigada. A coordenação competitiva átomo-átomo foi investigada por simulação PM7-MD. Os PLIs sintetizados mostraram boa estabilidade térmica e uma elevada capacidade de sorção de  $CO_2$  influenciada diretamente pelo contra-ânion. O maior valor de sorção foi encontrado para PIL  $[CelEt_3N][PF_6]$  (38 mg / g a 0,1 MPa e 168 mg / g a 3 MPa a 298,15 K). Os ciclos de sorção/dessorção de  $CO_2$  evidenciaram a estabilidade destes materiais durante o processo de sorção de  $CO_2$ .

## Cellulose based poly(ionic liquids): Tuning cation-anion interaction to improve carbon dioxide sorption

Franciele L. Bernard<sup>a,b</sup>, Rafael B. Duczinski<sup>a</sup>, Marisol F. Rojas<sup>d</sup>, Maria Carolina C. Fialho<sup>c</sup>, Luz Ángela Carreño<sup>d</sup>, Vitaly V. Chaban<sup>e</sup> Felipe Dalla Vecchia<sup>a,c</sup> and Sandra Einloft<sup>a,b\*</sup>

*a ) Post-Graduation Program in Materials Engineering and Technology. Pontifical Catholic University of Rio Grande do Sul – PUCRS.*

*b ) School of Chemistry. Pontifical Catholic University of Rio Grande do Sul – PUCRS.*

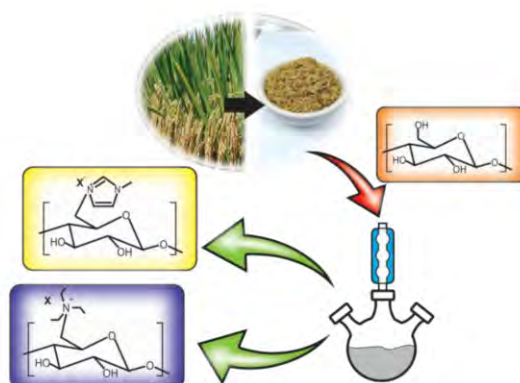
*c ) School of Engineering, Pontifical Catholic University of Rio Grande do Sul – PUCRS. Av. Ipiranga, 6681, Partenon, Porto Alegre Brazil, CEP: 90619-900.*

*d) School of Chemistry, Universidad Industrial de Santander, Street 9, 27 Bucaramanga, Colombia*

*e ) Institute of Science and Technology (ICT), Federal University of São Paulo (UNIFESP); São José dos Campos, SP – Brazil.*

\* Corresponding author: [einloft@pucrs.br](mailto:einloft@pucrs.br)

### Graphic Abstract



### Abstract

Mitigation of climate change is an urgent agenda. Development of materials from agroindustrial residue to be used as sorbents for CO<sub>2</sub> capture is a significant goal to be achieved both for environmental protection and resource utilization. In this work, cationic cellulosic PILs using cellulose extracted from rice husk were synthesized. Imidazolium and ammonium were evaluated as cation combined with

different counter- anions ( $[\text{Cl}]^-$ ,  $[\text{BF}_4]^-$ ,  $[\text{PF}_6]^-$  or  $[\text{TF}_2\text{N}]^-$ ). Synthesized PILs were characterized by NMR, XPS, DRX, FTIR, TGA and FESEM. Reusability and  $\text{CO}_2$  sorption capacity were performed by pressure decay-technique. Competitive atom-atom coordination was investigated by PM7-MD simulations. PILs showed a high  $\text{CO}_2$  sorption capacity directly influenced by anion. At 3 MPa of  $\text{CO}_2$  pressure the sorption capacity increases in the following order for both cations  $[\text{Cl}]^- < [\text{BF}_4]^- < [\text{TF}_2\text{N}]^- < [\text{PF}_6]^-$ . Simulation results suggest that  $\text{TF}_2\text{N}$  blocks important  $\text{CO}_2$  binding sites. The best sorption value was found for PIL  $[\text{Ce}(\text{Et}_3\text{N})][\text{PF}_6]$  of 38 mg/g at 0.1MPa and 168 mg/g at 3MPa at 298.15K. The recycling experiments of  $\text{CO}_2$  sorption demonstrated a good reusability of synthesized PILs for  $\text{CO}_2$  sorption.

## 1. Introduction

Carbon capture technologies emerge as one of the main strategies to reduce  $\text{CO}_2$  emissions and mitigate the climate change effect[1–11]. Development of versatile materials for  $\text{CO}_2$  separation from exhausted gases has become one of the main research niches in this area.[6,8,12,13]

Amines aqueous solutions stand out as the benchmark solvent for  $\text{CO}_2$  separation from exhaust gases[14–19] and natural gas[14,15,17,20]. However the use of this solvents presents operational drawbacks such as high energy consumption in regeneration stage, corrosive products formation as well as additional operational costs due to solvents substitution as a consequence of degradation[14,16,18,19,21].

Poly (ionic liquids) or polymeric ionic liquids (PILs) have shown potential for  $\text{CO}_2$  capture<sup>11,20–24</sup>. These functional polymers present an ionic liquid (IL) moiety on each repeated unit [8,12,22]. ILs polymerization resulted in increased  $\text{CO}_2$  capture efficiency. PILs present higher  $\text{CO}_2$  sorption capacity, faster sorption/desorption kinetics once compared to ILs [27]. PILs synthesis is carried out following these main routes: IL monomers direct radical polymerization,



polycondensation and/or polymers modification [8,28,29]. Among these routes the latter is extremely attractive allowing the use of commercially available polymers [8,28,29].

Different PILs obtained via polycondensation and/or polymers modification has been reported as well as their use in CO<sub>2</sub> capture. Literature describe PILs based on polybenzimidazoles [29–31], polyimides [28], poly(ether ketone)s [32] and polyurethane [33–35].

Cellulose is a versatile low cost natural polymer.[36,37] The possibility of its structure undergo chemical modification by reaction with available hydroxyl groups allows its application in several areas [36–38]. World rice crop is estimated at 500 million tons per year. The rice husk is the main agroindustrial residue generated in this cereal processing [39,40]. The use of this residue as raw material source for extracting cellulose can contribute significantly to the environment protection. In addition, if used in PILs synthesis it may reduce waste generation and produce low cost sorbents [41].

We have recently reported the synthesis of cellulose based anionic PILs and their ability to capture CO<sub>2</sub>[41]. PILs were obtained by chemical modification of cellulose fibers extracted from rice husk. Four different ionic liquids (IL) counter-cations were introduced (imidazolium, phosphonium, ammonium and pyrrolidinium) [41]. Cellulose based PILs were presented as a promising approach for CO<sub>2</sub> capture. Yet, counter-cation introduction into cellulose structure promoted increased CO<sub>2</sub> solubility and a completely reversible CO<sub>2</sub> sorption/desorption process. The best sorption result was obtained with ammonium as counter-cation (CL-TBA) [41].

A different approach was used to obtain cellulosic PILs in this work. Unlike our first work cationic cellulosic PILs were synthesized. PILs were obtained with imidazolium or ammonium cation, using different counter- anions ([Cl<sup>-</sup>], [BF<sub>4</sub><sup>-</sup>], [PF<sub>6</sub><sup>-</sup>] or [Tf<sub>2</sub>N<sup>-</sup>]). In addition, PM7-MD simulations were used to understand the competitive atom-atom coordination.

## 2. Experimental Section

### 2.1 Materials

Rice husk was donated by Cooperativa Arrozeira Extremo Sul Ltda. Sodium hydroxide (NaOH, 97% Vetec), hydrogen peroxide (H<sub>2</sub>O<sub>2</sub>, 35%, Neon), sulphuric acid (H<sub>2</sub>SO<sub>4</sub>, F.Maia, P.A), Thionyl chloride (SOCl<sub>2</sub>, 99%, Aldrich), ammonium hydroxide solution (30-32%, Vetec) 1-Methylimidazole (Aldrich, 99.0%), Triethylamine (99%, Vetec), Lithium(I) bis(trifluoromethanesulfonyl)imide (LiTf<sub>2</sub>N, 99%, Aldrich), Sodium tetrafluoroborate (NaBF<sub>4</sub>, 98 %, Aldrich), Sodium hexafluorophosphate (NaPF<sub>6</sub>, 98%, Aldrich) and N,N –dimethylformamide (DMF, 99.5%, Merck) were used as received

### 2.2 Cellulose extraction and modification

Cellulose fibers extraction procedure from rice husk was adapted from literature [42,43]. Fibers were obtained by submitting rice husk to alkali (NaOH) and acid (H<sub>2</sub>SO<sub>4</sub>) treatments followed by bleaching (H<sub>2</sub>O<sub>2</sub>/ NaOH). Cellulose extraction details are well reported elsewhere.[41]

Cellulosic based Poly (ionic liquid)s (PILs) syntheses were performed in a three step process as shown in Figure 1. The experimental procedure was based on literature.[44–47] Initially, cellulose was chemically modified with SOCl<sub>2</sub> to obtain chlorinated cellulose (6-chloro-6-deoxycellulose, CDC) (Figure 1a). A sample of 10 g of cellulose fiber was suspended in DMF (200 mL) followed by the slow addition of 35 mL of thionyl chloride (SOCl<sub>2</sub>) at 80°C under mechanical stirring. After the addition was complete, the reaction was kept for 2.5 h at 90 °C. The viscous solution was cooled to room temperature and poured into iced water (500 mL) under stirring. The precipitated (CDC) was treated with several aliquots of ammonium hydroxide solution and washed with water to bring the pH to neutral. The CDC was then separated by filtration and dried under vacuum at 60°C. In the next step, a sample of CDC (4g) was reacted with 40 mL of 1-Methylimidazole (at 100°C) or Triethylamine (at 80°C) and kept under stirring for 24 h. The cellulose

based PILs ([Celmim][Cl] or [CelEt<sub>3</sub>N][Cl]) obtained from this reaction were precipitated using methanol and then dried under vacuum at 60 °C. Finally, to exchange chloride from ([Celmim][Cl] or [CelEt<sub>3</sub>N][Cl]) by another anion, lithium or sodium salt of the corresponding anion was used. The ion exchange step was performed at room temperature with stirring for 24 h using saturated solution of LiTf<sub>2</sub>N, NaBF<sub>4</sub> or NaPF<sub>6</sub> (Figure 1b).

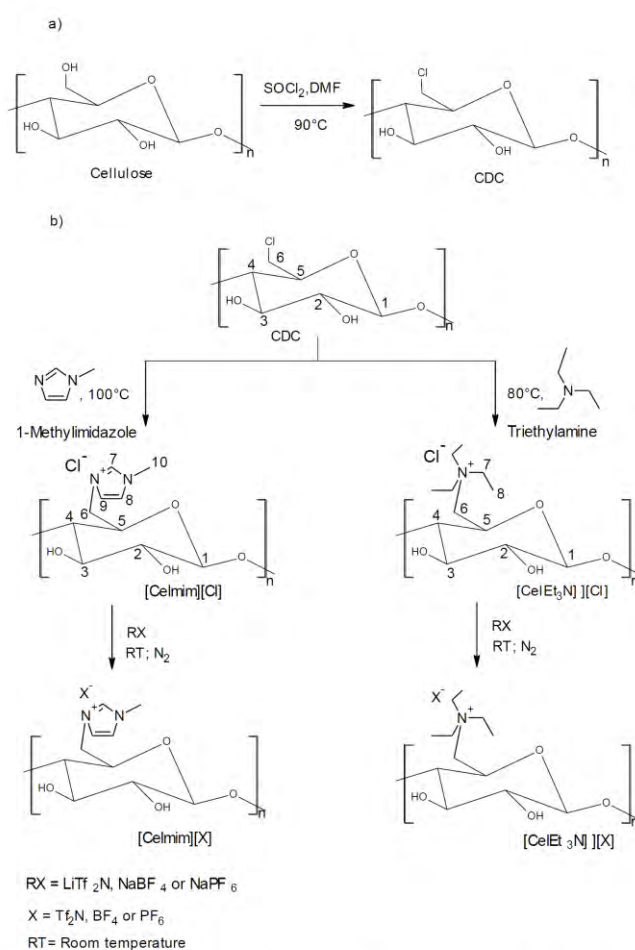


Figure 1. Synthesis steps and proposed structures of obtained materials: (a) 6-chloro-6-deoxycellulose (CDC); (b) Cellulosic poly (ionic liquids).

### 2.3 Characterization

The synthesized materials were characterized by Universal Attenuated Total Reflectance sensor (UATR-FTIR) using Perkin-Elmer 100 FTIR Spectrum, 4000-650  $\text{cm}^{-1}$  range. The solid-state  $^{13}\text{C}$  NMR spectra were obtained by Bruker Avance DRX-400 spectrometer at 400 MHz. Field emission scanning electron microscopy (FESEM) analyses were performed by FEI Inspect F50 equipment in secondary electrons (SE) mode. Thermogravimetric Analysis (TGA) was performed using TA Instruments SDT-Q600 between 25 and 500°C with a heating rate of 10°C/min in a nitrogen atmosphere. Powder X-ray diffraction (XRD) pattern was recorded by Bruker-AXS D8 ADVANCE diffractometer operated at 40 kV voltage and 20 mA current using Cu K $\alpha$  radiation ( $\lambda = 1.5406 \text{ \AA}$ ), 3 ~ 40° range with 0.02 step and 1.0 min scanning time. To evaluate the surface composition of the cellulosic poly (ionic liquids), survey scans were carried out by XPS/ISS/UPS-ACenteno surface characterization platform built by SPECS (Germany). The platform is equipped with PHOIBOS 150 2D-DLD energy analyzer. A monochromatic Al K $\alpha$  X-ray source (FOCUS 500) operated at 200 W was used for measurements. The pass-through energy of the hemispherical analyzer was set at 100 eV for the general spectra and 60 eV for high resolution spectra. The surface charge compensation was monitored with a Flood Gun device (FG 15/40-PS FG500) operated at 58  $\mu\text{A}$  and 2 eV. High-resolution spectra of the elements found were recorded for quantification, CasaXPS software was used for spectral analysis and curve fitting using C-C signal position at 285 eV as reference [47–49].

### 2.4 Sorption measurements

$\text{CO}_2$  sorption capacity was assessed by pressure-decay technique at 298,15 K and 0.1-3 MPa using a dual-chamber gas sorption cell [50]. Operation of such device was reported elsewhere [51,52]. Samples ( $W_s = 1.0\text{-}1.5 \text{ g}$ ) were weighted, moved to sorption chamber and subjected to  $10^{-3}$  mbar vacuum at 298.15°C during 1 h.  $\text{CO}_2$  (air liquid/99.998%) was

carried out into reservoir at desired pressure and allowed to enter the sorption chamber.

Gas sorption at equilibrium was determined from the difference between the initial and final number of moles of gas using Eqs. 1-2, respectively. The gas mass adsorbed in the sample is denoted as  $W_{CO_2/g}$

$$n_{CO_2} = \frac{p_i V_{gc}}{Z(p_i, T_i) R T_i} - \frac{p_{eq} (V_t - V_p)}{Z(p_{eq}, T_{eq}) R T_{eq}}, \quad (1)$$

$$W_{CO_2/g} = \frac{n_{CO_2} M}{W_s}, \quad (2)$$

where  $V_{gc}$  is volume of the gas chamber,  $p_i$  and  $T_i$  are pressure and temperature in the gas chamber,  $p_{eq}$  and  $T_{eq}$  are pressure and temperature at equilibrium.  $V_c$  is total volume of the sorption cell,  $Z$  is compressibility factor for the pure gas calculated by the Span-Wagner equations-of-state for  $CO_2$  (Equation: converging the fluid region at the triple-point temperature up to 1100K at pressures up to 800 MPa).[53]

## 2.5 Computer simulations

Semi-empirical molecular dynamics simulations, PM7-MD,[54–62] were employed to investigate atom-atom coordination. The wave function convergence criterion was set to  $10^{-7}$  Hartree with the Slater basis set. The equations-of-motion were propagated with a time-step of 0.5 fs during 4.0 ns for each system, of which first 0.5 ns were disregarded as equilibration. Equilibration was controlled by simultaneous monitoring of thermodynamic and structure properties. All simulated systems were maintained at 300 K by means of Andersen thermostat.[63] The simulated systems consisted of 2 to 8 monomers (Figure 1) and no qualitative dependence of the structure distributions on the system size was identified. Empirical corrections for van der Waals interactions and hydrogen bonding interactions were applied on-the-fly.[54,64,65] PM7-MD simulations were extensively used before, including for the  $CO_2$  capture problems,[55–57] and provided satisfactory to excellent correspondence with the experimental results.

Gabedit 2.8,[66] Packmol,[67] and VMD 191 [68] were used as supplementary utilities. All other operations were performed in the software developed by V.V.C.

### 3. Results and discussion

Structures of 6-chloro-6-deoxycellulose CDC, [Celmim] [Cl] and [CeEt<sub>3</sub>N] [Cl] synthesized materials were confirmed by <sup>13</sup>CNMR (Figure 2), XPS analysis (Figure 3) and UATR-FTIR (Figure 4).

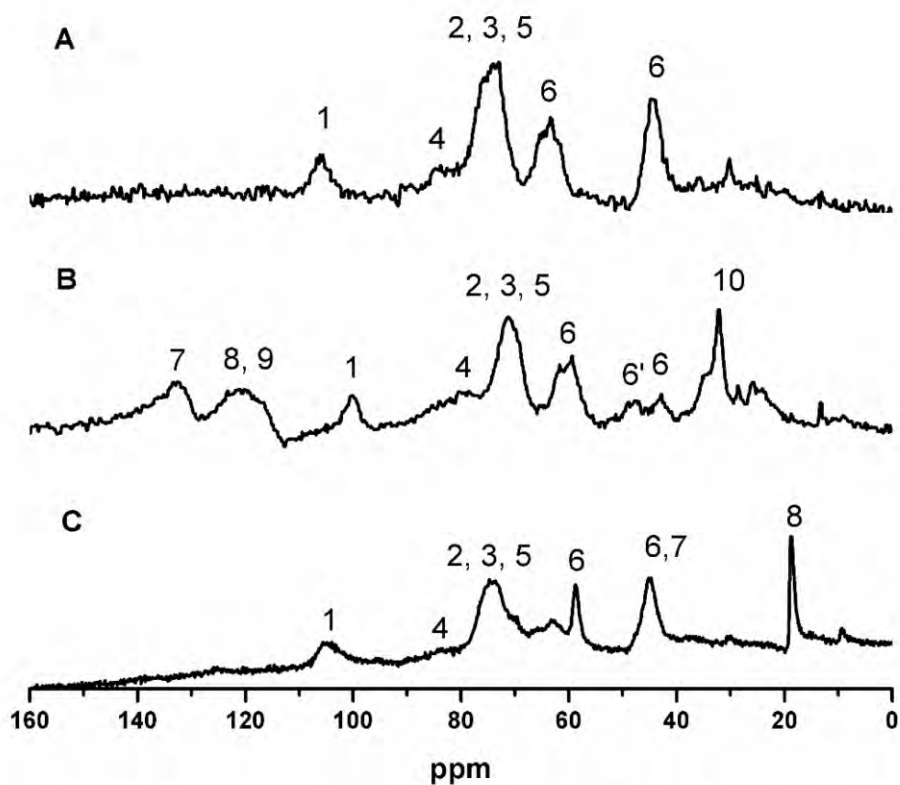


Figure 2. <sup>13</sup>C NMR spectra of CDC (A), [Celmim][Cl](B) and [CeEt<sub>3</sub>N][Cl](C) .

Resonance signals shown in Figure 2 and their assignments were compared to literature[44] and to starting materials spectra using the Spectral Database for Organic Compounds (SDBS)[69]. The six carbon atoms of CDC presented peaks

showed in spectrum: carbon 1 (O-C-O) at 105 ppm, carbon 4 (C-O) at 83 ppm, carbons 2, 3 e 5 (secondary carbons bonded to -CH or -OH) at 73 ppm are in overlay. For carbon 6 two peaks were revealed at 63 ppm, attributed to primary carbon bonded to -OH and at 44 ppm attributed to partial exchange of -OH group by Cl. Formation of [Celmim][Cl] was confirmed at 50 ppm peak attributed to the partial replacement of Cl group by the imidazole cation at carbon 6. Imidazolium was detected by the following peaks: carbon 7 was revealed at 142 ppm (NCN); carbon 8 and 9 at 127 ppm (CH-CH-N) and carbon 10 at 34 ppm (CH<sub>3</sub>N). For [CeIN222][Cl] sample a new peak at 18 ppm (CH<sub>3</sub>-CH<sub>2</sub>-N) and also the increase in peak intensity at 44 ppm indicating the overlap of carbon 6 (CH<sub>2</sub>-N or CH<sub>2</sub>-Cl) with carbon 7 (CH<sub>2</sub>-N of triethylamine) was observed.

Surface composition analysis confirms that cellulosic poly (ionic liquids) were obtained. Survey spectra of cellulose and cellulosic PILs are shown in Figure 3. Their elemental composition and elements ratio are shown in Table 1. The calculated elements ratio obtained from the proposed structure are also shown. The cellulose extracted is not completely pure, since the C/O ratio is higher than the theoretical value [47], probably due to residual lignin, hemicellulose, wax and other impurities [42,48]. In chlorinated cellulose (CDC), the C/O ratio decreased showing that impurities were removed from surface in the treatment with SOCl<sub>2</sub>/DMF. A small amount of sulfur impurities and a chlorine excess from SOCl<sub>2</sub> is observed both in CDC and in cellulosic PILs. In some of these, Si impurities come from de SiO<sub>2</sub> of rice husk[70] and the Na from NaBF<sub>4</sub> salt. In cellulosic PILs the ratio C/N is higher than the calculated, showing that probably the reaction with 1-methylimidazole and triethylamine was not given on all halogenated carbons. In cellulosic PILs the presence of the atoms from the ILs anions is observed. However, the C/F, C/P, C/B and C/S ratio were higher than expected because, as discussed previously, there is a smaller amount of available cations than projected. From deconvoluted high resolution carbon C 1s and Cl 2p spectra the relative amounts of carbon bond types were calculated and the organically bound chlorine and inorganic chloride were differentiate [71]; the results are shown in Table 2. The cellulose C 1s spectrum showed three signals at 285.0, 286.7, 288.0 and 289.9 eV,

assigned as C1 (C-C, C-H mainly from lignin), C2 (C-O, C-OH from cellulose), C3 (O-C-O, C=O from either cellulose or lignin) and C4 (O=C-O), respectively [48,49,71,72]. A high percentage of C1 was found, which confirms lignin presence [48,71,73]. For the CDC, the amount of C1 decreases, confirming impurities removal. In CDC and cellulosic PILs it was not possible to differentiate other carbon types as they appear at binding energies similar to those of cellulose, which is in greater proportion [74]. It is possible to observe for [Celmim][Tf<sub>2</sub>N] and [CeEt<sub>3</sub>N][Tf<sub>2</sub>N] a signal C5 at 293 eV corresponding to C-F<sub>3</sub> of the anion [75]. However, for [CeEt<sub>3</sub>N][Tf<sub>2</sub>N] this signal has a low intensity related to the lower amount of fluoride found on this one. In the CDC Cl 2p spectrum we found a high amount of organically bound chlorine at 200 eV and a low amount of inorganic chloride, whereas in all cellulosic PILs spectra the organically bound chlorine decreases considerably, even to zero for [CeEt<sub>3</sub>N][BF<sub>4</sub>], and an inorganic chloride at 198 eV was predominant [71]. This confirms that ion exchange did not take place in all halogenated carbons and that excess chlorine is due to inorganic chlorine. N 1s and S 2p High-resolution spectra for [Celmim][Tf<sub>2</sub>N] and [CeEt<sub>3</sub>N][Tf<sub>2</sub>N] allowed to identify clearly the nitrogen from the cation (at 400.1 eV) and from the anion (at 397.6 eV), as well as the sulfur of the anion (at 167.2 eV) and from the impurity (at 162.3 eV) [76,77].



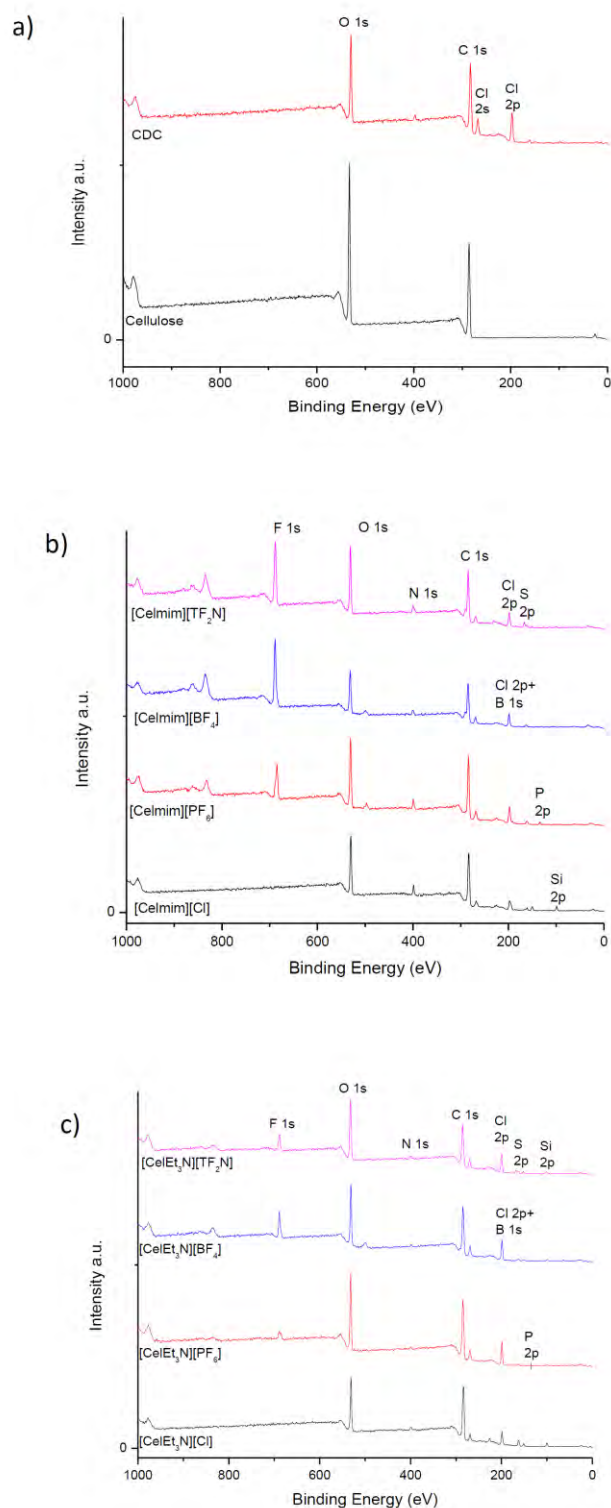


Figure 3. Survey spectra of cellulose and CDC(a), and cellulosic PILs with cation [Celmim] (b) and [CelEt<sub>3</sub>N] (c).

Table 1. Surface composition and element ratio of cellulose and cellulosic PILs from survey XPS spectra.

Sample	Surface composition (at.%)											Element ratio <sup>a</sup>										
	C	O	N	Cl	F	S	P	B	Na	Si	C/O	C/N	C/Cl	C/F	C/P	C/B	C/S					
	exp	cal	exp	cal	exp	cal	exp	cal	exp	cal	exp	cal	exp	cal	exp	cal	exp	cal				
Cellulose	64.5	35.5									1.8	1.2										
CDC	40.4	39.7	2.4	15.8		2.1					1.0	1.5		2.6	6.0							
[Celmim][Cl]	41.4	38.4	6.3	8.8		2.2			3.0		1.1	2.5	6.6	5.0	4.7	10						
[Celmim][PF <sub>6</sub> ]	51.4	22.2	4.9	5.4	11.8	1.6	1.5		1.1		2.3	2.5	10.5	5.0	9.5	--	4.4	1.7	34.1	10.0		
[Celmim][TF <sub>2</sub> N]	26.4	28.1	3.5	5.9	32.7	3.4					0.9	1.5	7.5	4.0	4.5	--	0.8	2.0		7.9	6.0	
[Celmim][BF <sub>4</sub> ]	44.6	16.4	2.7	10.9	22.7	0.9		0.5	1.3		2.7	2.5	16.3	5.0	4.1	--	2.0	2.5	87.1	10.0		
[CelEt <sub>3</sub> N][Cl]	65.3	20.2	2.2	5.6		3.2				3.4	3.2	3.0	29.3	12.0	11.6	12.0						
[CelEt <sub>3</sub> N][PF <sub>6</sub> ]	62.0	25.2	1.3	7.1	3.5	0.7	0.3				2.5	3.0	49.2	12.0	8.7	--	17.6	2.0	213.6	12.0		
[CelEt <sub>3</sub> N][TF <sub>2</sub> N]	55.4	27.7	2.3	5.1	5.3	2.5				1.7	2.0	1.8	23.8	7.0	10.9	--	10.4	2.3		10.4	7.0	
[CelEt <sub>3</sub> N][BF <sub>4</sub> ]	50.4	21.4	1.2	16.4	7.9	1.1		0.6	1.0		2.3	3.0	40.9	12.0	3.1	--	6.4	3.0	83.9	12.0		

<sup>a</sup> Calculated elements ratio obtained from the proposed structure

Table 2. Carbon type and chlorine percentages in cellulose and cellulosic PILs from high resolution XPS spectra.

Sample	Carbon type composition (at. %)					Chlorine (at. %)	
	C1	C2	C3	C4	C5	Cl 1	Cl 2
Cellulose	28.5	57.3	11.3	2.9		--	--
CDC	17.9	40.7	32.5	8.9		28.4	71.6
[Celmim][Cl]	27.1	48.0	22.9	2.0		74.7	25.3
[Celmim][PF <sub>6</sub> ]	14.4	59.6	26.0	ND		64.5	35.6
[Celmim][TF <sub>2</sub> N]	12.9	65.4	10.0	ND	11.7	64.5	35.5
[Celmim][BF <sub>4</sub> ]	21.7	53.9	12.4	ND	11.9	74.6	25.4
[CelEt <sub>3</sub> N][Cl]	45.0	36.6	15.0	3.4		66.3	33.7
[CelEt <sub>3</sub> N][PF <sub>6</sub> ]	21.9	43.8	28.7	5.5		66.6	33.4
[CelEt <sub>3</sub> N][TF <sub>2</sub> N]	28.3	28.2	39.7	3.8		69.1	30.9
[CelEt <sub>3</sub> N][BF <sub>4</sub> ]	14.8	48.8	31.7	3.3	1.4	100.0	0.0

C1 (C-C, C-H), C2 (C-O, C-OH), C3 (O-C-O, C=O), C4 (O=C-O), C5 (C-F<sub>3</sub>), Cl 1 (inorganic chloride), Cl 2 (organically bound chlorine).

Typical infrared spectra of cellulose, chlorinated cellulose (CDC), [Celmim][Cl] and [CelEt<sub>3</sub>N][Cl] are shown in Figure 4. IR spectra of all samples exhibited characteristic bands of cellulose structure: 3330 - 3340 cm<sup>-1</sup> (stretching of -OH groups), 2920 cm<sup>-1</sup> (C-H stretching vibrations), 1369 cm<sup>-1</sup> (C-H bending), 1160 cm<sup>-1</sup> (C-O-C asymmetrical bridge stretching), 1050 cm<sup>-1</sup> (C-O stretching vibration) and 890 cm<sup>-1</sup> (C-H stretching vibration). [42,78,79]. Yet, the appearance of a band near 1640 cm<sup>-1</sup> corresponding to C=C stretching of aromatic rings, attributed to lignin and a band around 1700-1725 cm<sup>-1</sup> attributed to ester and acetyl groups in residual hemicellulose [42,78,79] was observed corroborating XPS findings.

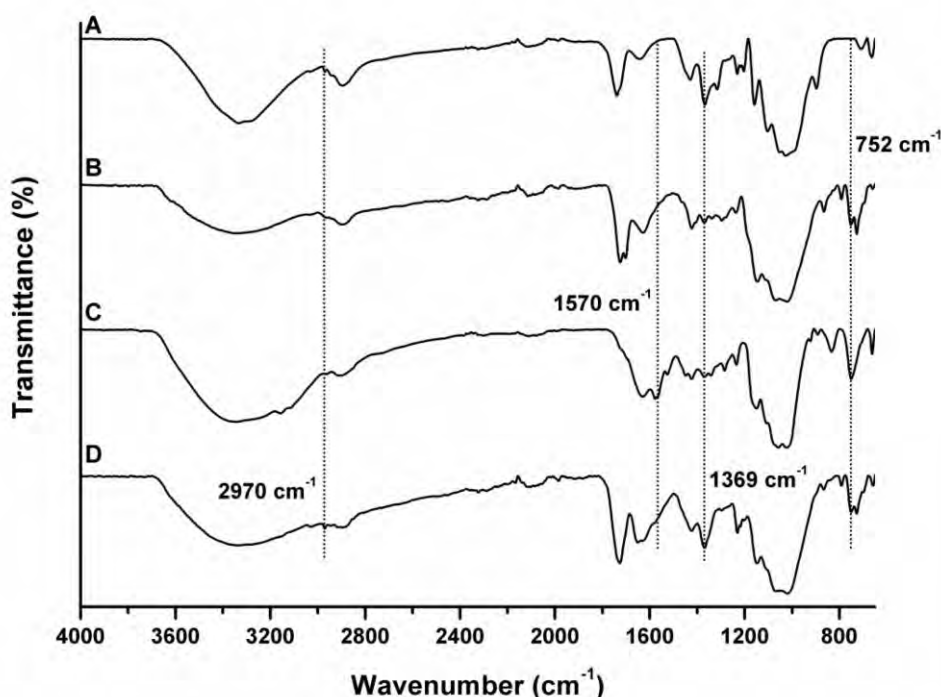


Figure 4. FT-IR spectra of the materials: (a) cellulose, (b) CDC, (c) [Celmim][Cl] and (d) [CelEt<sub>3</sub>N][Cl].

New bands were observed after the cellulose chemical modification (Figure 4). The CDC spectrum (Figure 4b) showed the appearance of two new bands near 752 e 723  $\text{cm}^{-1}$  related to stretching vibration of C-Cl bond [44–46]. The C-Cl appearing in CDC was also confirmed by XPS and <sup>13</sup>CNMR analysis. The spectrum obtained for imidazole based PILs revealed bands at 1570  $\text{cm}^{-1}$  and 1060  $\text{cm}^{-1}$  attributed to C=C stretching and C–N of imidazolium ring, respectively (Figure 4c).[44] For ammonium based PILs one can observe peaks intensification at 1369  $\text{cm}^{-1}$  (C-H bending), 2970 (C–H stretching of -CH<sub>2</sub>) and 2940  $\text{cm}^{-1}$  (C–H stretching of -CH<sub>3</sub>) associated to ammonium cation alkyl chains[69] (Figure 4d). The exchange of [Cl]<sup>-</sup> anion for [PF<sub>6</sub>]<sup>-</sup> or [Tf<sub>2</sub>N]<sup>-</sup> provided the appearance of new bands: near to 840  $\text{cm}^{-1}$  attributed to hexafluorophosphate anion[80,81], around 790  $\text{cm}^{-1}$  (C–S) and 846  $\text{cm}^{-1}$  (N–S) attributed to (trifluoromethanesulfonyl) imide anion[34]. The band near to 1050  $\text{cm}^{-1}$  of tetrafluoroborate anion [BF<sub>4</sub>]<sup>-</sup> [81] is in overlay with the band of C-H of cellulose structure.

Structural level changes may be assessed by X-ray diffraction (Figure 5 and 6). Figure 5 shows the diffractograms for cellulose extracted from rice husk (a), CDC (b) and cellulosic PILs [Celmim][Cl] (c) and [CeEt<sub>3</sub>N][Cl] (d). Cellulose showed typical peaks of microcrystalline cellulose around  $2\theta = 16^\circ$ ,  $22.6^\circ$  and  $35^\circ$  [42,44]. Changes in X-ray diffractograms profile for all samples were observed after functionalization (Figure 6). This behavior was also verified by Chen et al., 2015 [44]. Drastic crystallinity reductions were observed after cellulose functionalization (Figure 6). This indicates that cellulosic based PILs and CDC have an amorphous structure [44].

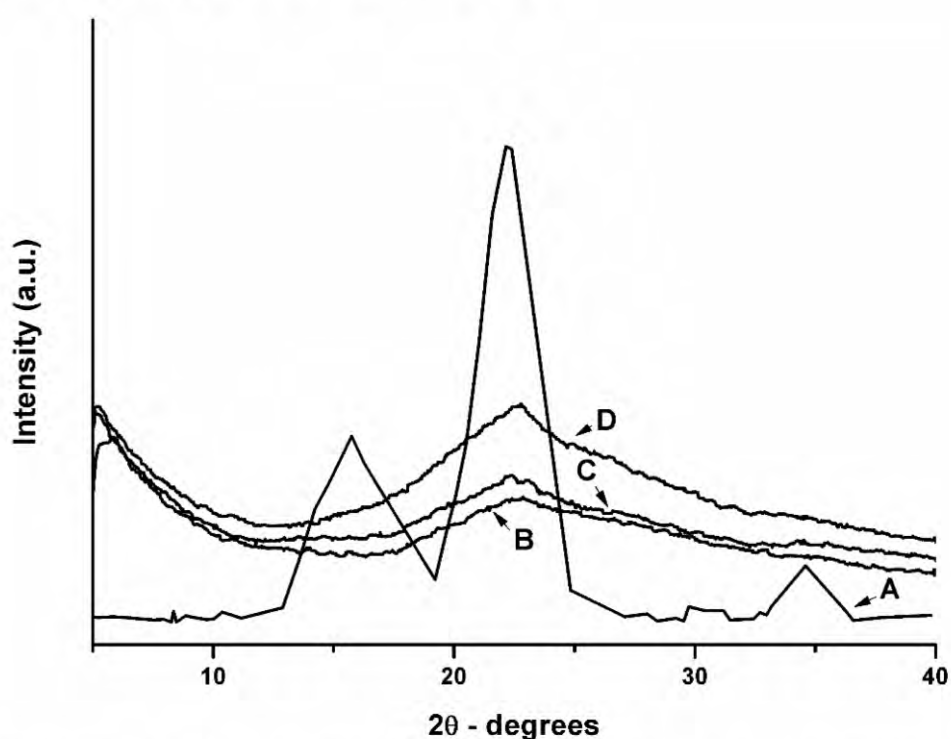


Figure. 5. XRD pattern of cellulose and modified cellulose samples: (a) cellulose, (b) CDC, (c) [CeEt<sub>3</sub>N][Cl] and (d) [Celmim][Cl].

The effect of cation side chain size and anion type on structural conformation of cellulosic PILs may be seen in Figure 5 and 6. It may be highlighted that the longer the side chain size of the inserted cation, the lower is the crystalline peak intensity. PIL [CeEt<sub>3</sub>N][Cl] evidenced lower crystalline peak intensity once compared to [Celmim][Cl] (Figure 5). When

the  $[\text{Cl}]^-$  anion exchange was performed by  $[\text{BF}_4]^-$ ,  $[\text{PF}_6]^-$  or  $[\text{TF}_2\text{N}]^-$ , a distinct behavior was evidenced for the different cations (Figure 6). Insertion of larger anions to  $[\text{Celmim}]^+$  further reduced the packaging degree of molecules, whereas for  $[\text{CelEt}_3\text{N}]^+$  cation the insertion of larger anions seems to contribute to a more ordered assembly (Figure 6).

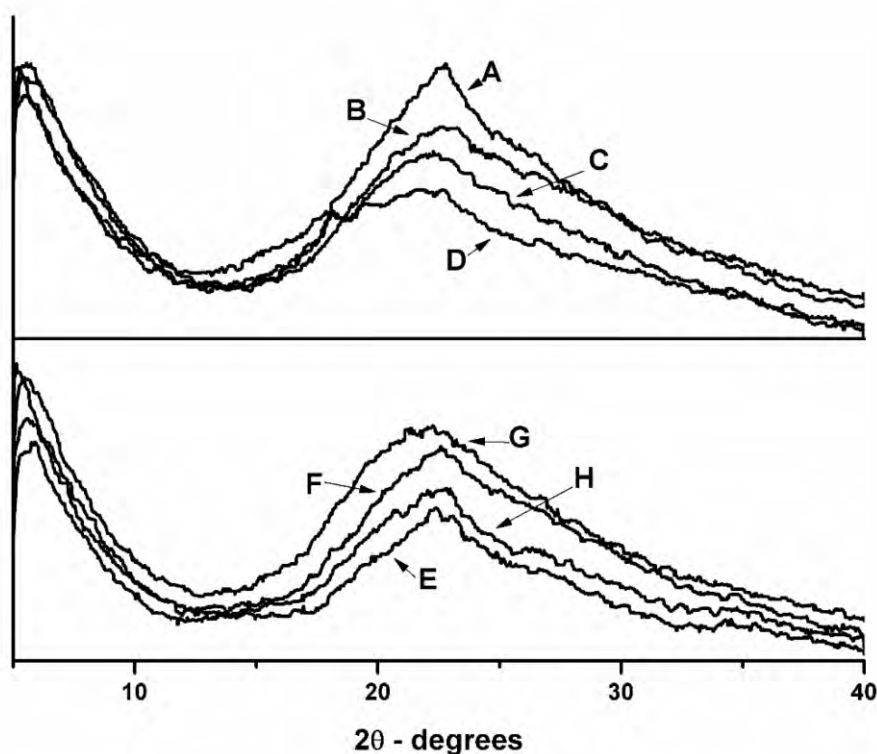


Figure. 6. XRD pattern of PILs: (a)  $[\text{Celmim}][\text{Cl}]$ ; (b)  $[\text{Celmim}][\text{BF}_4]$ ; (c)  $[\text{Celmim}][\text{PF}_6]$ ; (d)  $[\text{Celmim}][\text{TF}_2\text{N}]$ ; (e)  $[\text{CelEt}_3\text{N}][\text{Cl}]$ ; (f)  $[\text{CelEt}_3\text{N}][\text{BF}_4]$ ; (g)  $[\text{CelEt}_3\text{N}][\text{PF}_6]$ ; (h)  $[\text{CelEt}_3\text{N}][\text{TF}_2\text{N}]$ .

Cellulose, CDC and PILs morphology were investigated by FESEM (Figure 7). As expected the pure cellulose exhibited a typical fibrous-looking surface. This morphology was significantly altered in functionalized cellulose samples. Chemical modification resulted in fiber compaction and agglomeration, evidenced in all PILs and CDC (Figure 7).

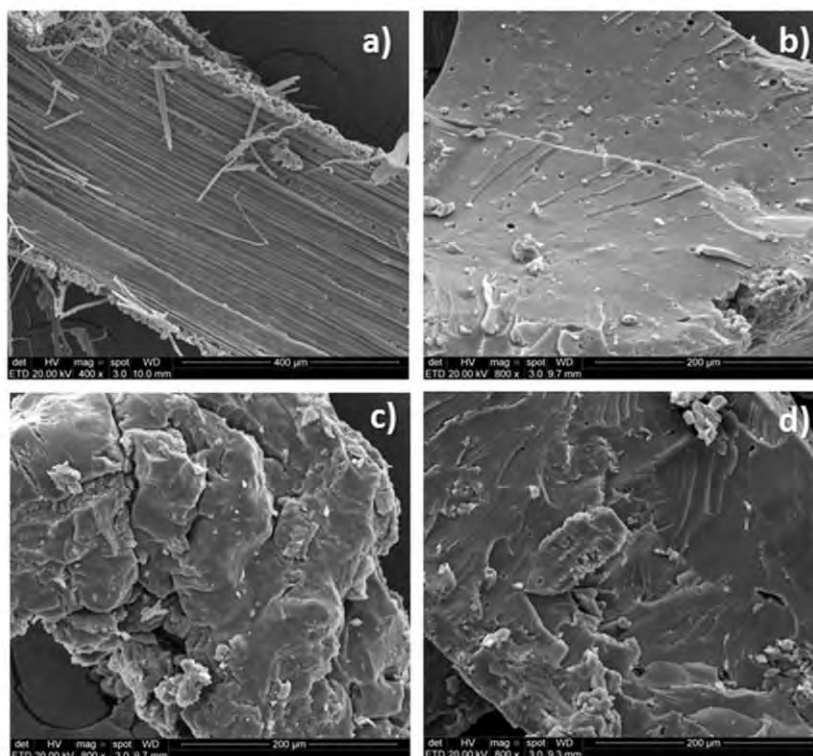


Figure 7. Micrographs of the materials: (a) cellulose, (b) CDC, (c) [Celmim][Cl] and (d) [CeEt<sub>3</sub>N][Cl].

Thermogravimetric curves of cellulose and PILs are shown in Figure 8. Cellulose presented an initial mass loss inferior to 5% ( $T_{1\text{onset}}$  of 44°C and  $T_{1\text{max}}$  of 63°C) attributed to physically absorbed water on surface[42]. Yet, a decomposition event with mass loss of 91% ( $T_{2\text{onset}}$  of 313°C and  $T_{2\text{max}}$  of 334°C), attributed to structure degradation was evidenced[42] (Figure 8). All PILs presented two thermal events: the first stage with mass loss of 38 % ( $T_{1\text{onset}}$  at 207°C and  $T_{1\text{max}}$  at 210°C) for [Celmim] [Cl] and 54% ( $T_{1\text{onset}}$  at 186°C and  $T_{1\text{max}}$  at 197°C) for [CeEt<sub>3</sub>N] [Cl] is attributed to imidazolium or ammonium group loss allied to condensation of the carbon -OH groups of carbons 2 and 3 [44,45,82]. The second stage with mass loss of 20% ( $T_{2\text{onset}}$  at 266°C and  $T_{2\text{max}}$  at 270°C) for [Celmim] [Cl] and 12% ( $T_{2\text{onset}}$  at 387°C and  $T_{2\text{max}}$  at 437°C) for [CeEt<sub>3</sub>N] [Cl] is attributed to cellulose fibers degradation [44,45] (Figure 8). Exchange of anion [Cl]<sup>-</sup> by fluorinated anions resulted in increased thermal stability for ammonium PILs ([BF<sub>4</sub>]<sup>-</sup> =  $T_{1\text{onset}}$  at 204°C and  $T_{1\text{max}}$  at 208°C; [PF<sub>6</sub>]<sup>-</sup> =  $T_{1\text{onset}}$  at 203°C and  $T_{1\text{max}}$  at

210°C;  $[\text{TF}_2\text{N}]^-$  =;  $T_{1\text{onset}}$  at 204°C and  $T_{1\text{max}}$  at 213°C) whereas for imidazolium PILs the thermal stability remained almost constant with the anion exchange ( $[\text{BF}_4]^-$  =  $T_{1\text{onset}}$  at 203°C and  $T_{1\text{max}}$  at 208°C;  $[\text{PF}_6]^-$  =;  $T_{1\text{onset}}$  at 198°C and  $T_{1\text{max}}$  at 200°C;  $[\text{TF}_2\text{N}]^-$  =  $T_{1\text{onset}}$  at 205°C and  $T_{1\text{max}}$  at 212°C) (Figure 8).

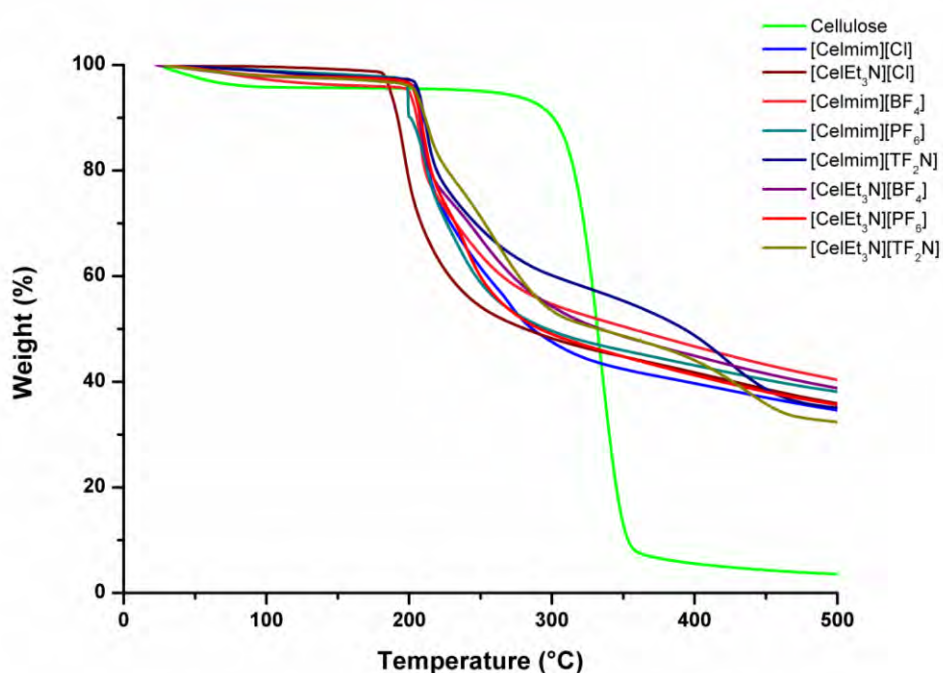


Figure. 8. TGA thermograms of cellulose and PILs

Experimental results of  $\text{CO}_2$  sorption for all synthesized PILs as well as for cellulose are shown in Figure 9. Table 3 presents a parallel for  $\text{CO}_2$  sorption values of PIL  $[\text{CelEt}_3\text{N}][\text{PF}_6]$  relative to PILs described in literature.

It may be evidenced from Figure 9 that cellulose structure ionization promotes increased  $\text{CO}_2$  affinity. This behavior was also observed in anionic cellulosic PILs[41]. Cellulose presented  $\text{CO}_2$  sorption capacity of 20 mg/g at low pressure (0.1MPa) and 88 mg/g at higher pressures (3 MPa). Cellulose sorption capacity is related to the  $\text{CO}_2$  affinity by polar groups (ether, ester and hydroxyl groups) of cellulose structure[41]. PILs presented superior sorption capacity once compared to cellulose (Figure 9). PILs  $[\text{Celmim}][\text{Cl}]$  and  $[\text{CelEt}_3\text{N}][\text{Cl}]$



presented similar CO<sub>2</sub> sorption capacity at low pressure (27 mg/g and 26 mg/g, respectively). At higher pressure a different behavior was observed. PIL [Celmim][Cl] presented sorption capacity of 113 mg/g and [CelEt<sub>3</sub>N][Cl] 89 mg/g. Anion effect for CO<sub>2</sub> sorption capacity may also be observed (Figure 9). The exchange of [Cl]<sup>-</sup> anion for fluorinated ones increased the CO<sub>2</sub> affinity. At higher CO<sub>2</sub> pressures (3 Mpa), sorption capacity increases in the following order for both cations: [Cl]<sup>-</sup> < [BF<sub>4</sub>]<sup>-</sup> < [TF<sub>2</sub>N]<sup>-</sup> < [PF<sub>6</sub>]<sup>-</sup>. Preferential affinity of CO<sub>2</sub> for ionic liquids containing the anion [TF<sub>2</sub>N] rather than other anions with lower fluorination ([BF<sub>4</sub>]<sup>-</sup>, [PF<sub>6</sub>]<sup>-</sup>) is well known [83]. For cationic cellulosic PILs a different behavior was evidenced since the anion [PF<sub>6</sub>]<sup>-</sup> provided higher performance. It is probable due to the size of [TF<sub>2</sub>N]<sup>-</sup> causing steric effect and making it more difficult to CO<sub>2</sub> interact with polar groups present in cellulose structure, resulting in CO<sub>2</sub> sorption capacity reduction. This was also observed using semi-empirical molecular dynamics simulations (Figure 10).

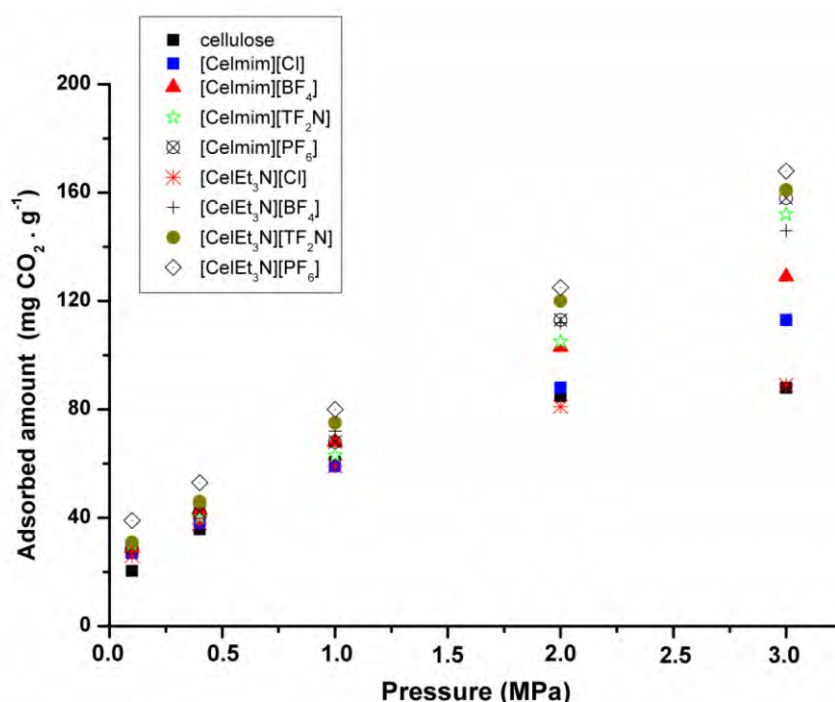


Figure 9. CO<sub>2</sub> sorption for cellulose and PILs.

To corroborate experimental observations, competitive atom-atom coordination (Figure 10) was investigated by the PM7-MD simulations, as described in the methodology. The simulations suggest that different CO<sub>2</sub> sorption capacities of [Celmim][PF<sub>6</sub>] and [Celmim][NTF<sub>2</sub>] are driven by the cation-anion coordination peculiarities. Due to its bulky structure and significantly polar moieties, the NTF<sub>2</sub> anion is located near the most CO<sub>2</sub>-philic groups of imidazole and cellulose. This effect can also be seen for the PF<sub>6</sub> anion, in which case it is, however, less pronounced (Figure 10). The corresponding peaks for NTF<sub>2</sub> are somewhat higher and sharper. In particular, the acidic hydrogen atom of the imidazole ring and two hydroxyl groups of cellulose are blocked. The distance between NTF<sub>2</sub> and the hydroxyl groups of cellulose, 0.21 nm, suggests formation of a strong hydrogen bond. According to the PM7-MD simulations, coordination of CO<sub>2</sub> by these moieties is not feasible. Note that, unlike fluorine atoms of PF<sub>6</sub>, are well distributed in space and can approach several CO<sub>2</sub>-philic sites of the cationic cellulose simultaneously.

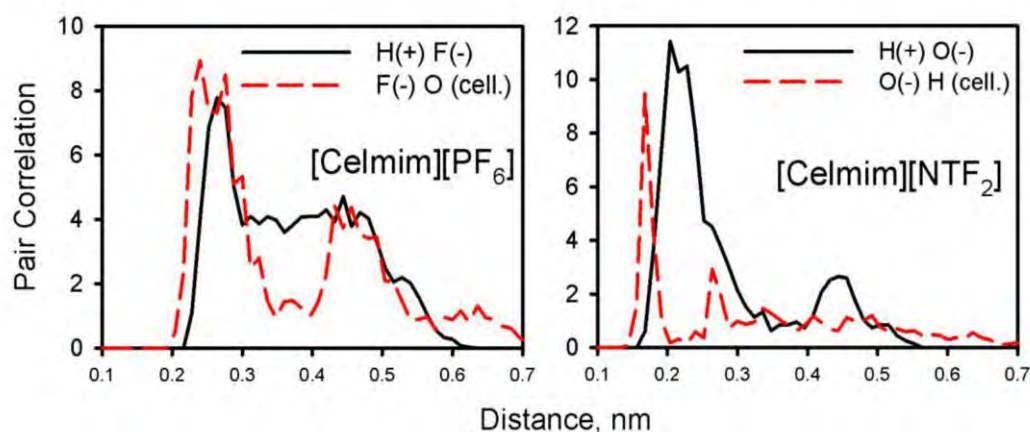


Figure 10. Selected pair correlation functions obtained from the PM7-MD simulations. The strongest spatial correlations between the anion (-), cationic (+) and neutral (cellulose) moieties are presented for comparison.

Table 3 presents CO<sub>2</sub> sorption values for PILs described in literature and PIL [CeEt<sub>3</sub>N][PF<sub>6</sub>]. As one may see PIL [CeEt<sub>3</sub>N][PF<sub>6</sub>] exhibited higher sorption value when compared to PILs obtained via direct radical polymerization (P[VBIT][BF<sub>4</sub>]; P[VBIH][PF<sub>6</sub>]; P([AMIM]BF<sub>4</sub>-AN); P6 [BIEMA][Br]), as well as polyurethane based PILs (PIL-8.1.BF<sub>4</sub>; PU-TAB). It can also be verified in Table 3 a comparison between the CO<sub>2</sub> sorption performance of [CeEt<sub>3</sub>N][PF<sub>6</sub>] and the anionic cellulosic PIL CL-TBA at different pressures (0.1 MPa and 3 MPa). At low pressure the [CeEt<sub>3</sub>N][PF<sub>6</sub>] exhibited sorption capacity comparable to CL-TBA and at high pressure higher CO<sub>2</sub> sorption capacity.

Table 3. Comparison of [CeEt<sub>3</sub>N][PF<sub>6</sub>] with different poly(ionic liquids) in terms of CO<sub>2</sub> sorption.

PIL	CO <sub>2</sub> sorption (mg/g)	Conditions (P, T)	Ref.
P[VBIT][BF <sub>4</sub> ]	3.05	0.79 bar, 295.15K	[84]
P[VBIH][PF <sub>6</sub> ]	3.22	0.79 bar, 295.15K	[84]
P([AMIM]BF <sub>4</sub> -AN)	14.30	0.1 MPa, 273.15K	[85]
P6 [BIEMA][ Br]	3.34	0.1 MPa, 278.15K	[86]
P[VBTEA][PF <sub>6</sub> ]	14.04	0.1 MPa, 278.15K	[87]
PIL-8.1.BF <sub>4</sub>	24.76	0.1 MPa, 273.15K	[88]
PU-TAB	16.10	0.082 MPa, 303.15K	[33]
PU-TAB	123.0	3 MPa, 303.15K	[33]
CL-TBA	44	0.1MPa, 298.15 K	[41]
CL-TBA	71	3 MPa, 298.15 K	[41]
[CeEt <sub>3</sub> N][PF <sub>6</sub> ]	38	0.1MPa, 298.15 K	This study
[CeEt <sub>3</sub> N][PF <sub>6</sub> ]	168	3 MPa, 298.15 K	This study

In order to evaluate the reuse capacity of PIL [CeEt<sub>3</sub>N][PF<sub>6</sub>] five CO<sub>2</sub> sorption/desorption cycles were performed (Figure 11). Sorption was evaluated at 298,15 k and 0.1 MPa and desorption under vacuum at 298,15 k and 1h. Sorption/desorption values evidenced that CO<sub>2</sub> was completely removed in 1h and sorption values were constant in all five cycles (Figure 11). This result evidenced the capacity of this material to be used in CO<sub>2</sub> capture processes.

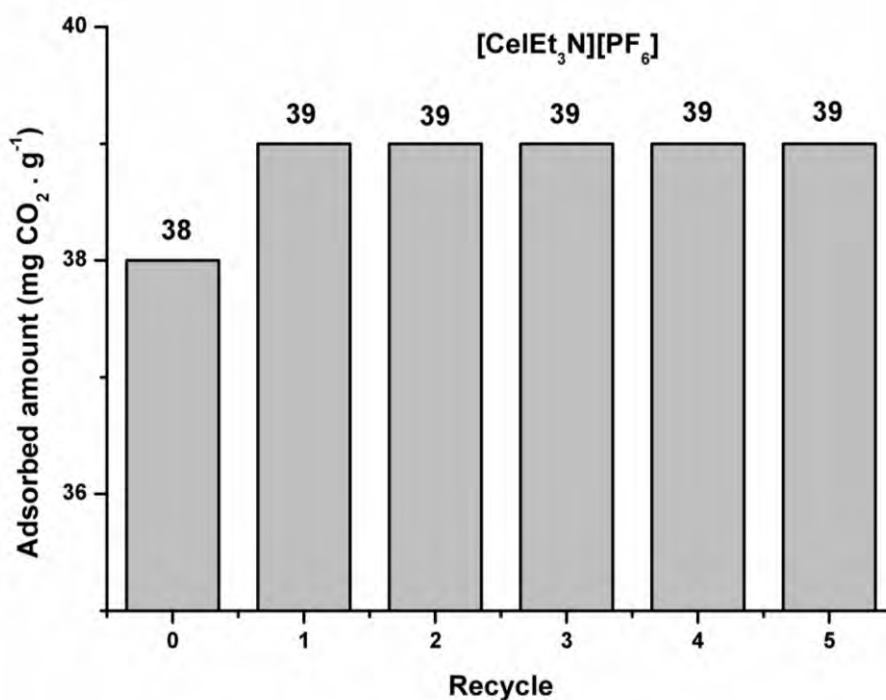


Figure 11. CO<sub>2</sub> sorption/desorption tests for [CeEt<sub>3</sub>N][PF<sub>6</sub>].

#### 4. Conclusions

Cationic cellulosic PILs obtained from rice husk cellulose extraction showed a high CO<sub>2</sub> sorption capacity when compared to anionic cellulosic PILs and other PILs reported in literature. The anion fluorination promotes CO<sub>2</sub> sorption capacity increase, however the use of voluminous fluorinated anions may impair the CO<sub>2</sub> interaction with polar groups of cellulose structure. The best sorption result was found for PIL [CeEt<sub>3</sub>N][PF<sub>6</sub>].

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#### 4.5. Capítulo V: Aminas e polieteramínas suportadas em celulose

O estudo relatado no artigo “*Development of Inexpensive Cellulose-Based Sorbents for Carbon Dioxide*” e submetido para o Brazilian Journal of Chemical Engineering (comprovante de submissão Anexo C), descreve o potencial para captura de CO<sub>2</sub> de fibras de celulose extraídas da casca de arroz, modificadas quimicamente com diferentes aminas e polieteramínas. Os compostos obtidos foram caracterizados por diferentes técnicas (FTIR, DRX, TGA, BET e MEV-FEG). Concentrações baixas de aminas e polieteramínas ( $2 \times 10^{-6}$  mol/mg) foram suportadas nas fibras de celulose. A capacidade de sorção de CO<sub>2</sub> foi avaliada gravimetricamente por uma balança de suspensão magnética. As simulações mecânicas quânticas e os resultados experimentais revelaram que a quimisorção é favorecida por grupos -NH- e -NH<sub>2</sub>. O melhor resultado de sorção de CO<sub>2</sub> foi alcançado para a celulose modificada com amina CL-D-400 (409 μmol CO<sub>2</sub> / g a 1 bar e 1091 μmol CO<sub>2</sub> / g a 10 bar).

## Development of Inexpensive Cellulose-Based Sorbents for Carbon Dioxide

Franciele L. Bernard <sup>a,b</sup>, Daniela. M. Rodrigues <sup>c</sup>, Barbara B. Polesso <sup>c</sup>, Vitaly V. Chaban <sup>d</sup>, Marcus Seferin <sup>a,b</sup>, Felipe Dalla Vecchia <sup>c</sup> and Sandra Einloft <sup>a,b\*</sup>

*a) Post-Graduation Program in Materials Engineering and Technology. Pontifical Catholic University of Rio Grande do Sul – PUCRS.*

*b) School of Chemistry. Pontifical Catholic University of Rio Grande do Sul – PUCRS.*

*c) School of Engineering, Pontifical Catholic University of Rio Grande do Sul – PUCRS. Av. Ipiranga, 6681, Partenon, Porto Alegre Brazil, CEP: 90619-900.*

*d) Institute of Science and Technology (ICT), Federal University of São Paulo (UNIFESP); São José dos Campos, SP – Brazil.*

\* Corresponding author: [einloft@pucrs.br](mailto:einloft@pucrs.br)

### Abstract

Aqueous amine solutions are benchmark solvents for CO<sub>2</sub> capture and their operational drawbacks are well-known. In order to overcome these problems, the support of amines on solid materials appears as an option for CO<sub>2</sub> capture. Cellulose is a versatile and low-cost material that can be used as a support. This study reports chemical modification of cellulose fibers extracted from rice husk with different amines and their potential for CO<sub>2</sub> capture. The obtained compounds were characterized by different techniques. The CO<sub>2</sub> sorption capacity was gravimetrically assessed in a Magnetic Suspension Balance. Quantum mechanical simulations and experimental results revealed that -NH- and -NH<sub>2</sub> represent major working sites of the employed compounds. The best result for CO<sub>2</sub> sorption was attained for the amine-modified cellulose CL-D-400 with a sorption capacity of 409 μmol CO<sub>2</sub>/g at 1 bar and 1091 μmol CO<sub>2</sub>/g at 10 bar with amine concentrations as low as 2 × 10<sup>-6</sup> mol/mg.

Keywords: cellulose, rice husk, CO<sub>2</sub> capture, amines, simulation.

## 1. Introduction

The CO<sub>2</sub> concentration in the atmosphere is increasing every year, as well as global warming and its effect on climate change (Barbosa, Damasceno e Hori, 2016; Chan *et al.*, 2016; Hussain, Farrukh e Minhas, 2015; Nouri e Ebrahim, 2016; Seo *et al.*, 2014). Preventing the growth of carbon dioxide emissions in atmosphere is essential for achieving the global warming reduction target (Dutcher, Fan e Russell, 2015). Different strategies have been considered and adopted by several countries aiming to reduce the CO<sub>2</sub> emissions, such as the improvement of energy efficiency, increasing the use of low carbon fuels or nuclear energy, renewable energy development, geo-engineering approaches and CO<sub>2</sub> capture and storage (CCS) (Garcés *et al.*, 2013; Leung, Dennis Y C, Caramanna e Maroto-Valer, 2014). Among these different strategies, CCS is considered crucial because it reduces the CO<sub>2</sub> emission from large emission sources (Arias *et al.*, 2016; Garðarsdóttir *et al.*, 2015; Kamarudin e Alias, 2013; Leung, Dennis Y.C., Caramanna e Maroto-Valer, 2014).

Chemical absorption using amine aqueous solutions is a benchmark technology for CO<sub>2</sub> capture (Arias *et al.*, 2016; Bachelor e Toochinda, 2012; Kamarudin e Alias, 2013). However, it has some operational drawbacks, such as high power consumption required for solvent regeneration, equipment corrosion (Gray *et al.*, 2005; Kamarudin e Alias, 2013), degradation/evaporation of amines [10], the need of a large-volume absorber (Yu, Huang e Tan, 2012), besides additional operating costs due to substitution of solvents (Seo *et al.*, 2014).

To overcome the drawbacks of chemical absorption, the impregnation and grafting of amines on solids have been suggested in literature (Bachelor e Toochinda, 2012; Gray *et al.*, 2005; Knowles *et al.*, 2005; Lima, Gomes e Lucena, 2015; Yu, Huang e Tan, 2012). Unfortunately, many of these commercial sorbents remain prohibitively expensive (Bachelor e Toochinda, 2012). This generates the necessity of developing inexpensive sorbents for CO<sub>2</sub> capture. As alternatives for obtaining low-cost sorbents, impregnation of agricultural wastes with amines has been suggested. Some examples are

bagasse, rice straw, industrial residues, such as mullite, fly ash and other oxygen rich sorbents (Bachelor e Toochinda, 2012; Gray *et al.*, 2004). Strategies to use residues for CO<sub>2</sub> capture will foster emergence of green technologies and sustainable development.

Brazil is among the ten largest rice producers in the world. By 2017, it produced about 11 759 096 tons of rice (IBGE- Grupo de Coordenação de Estatísticas Agropecuárias, 2017). Rice husk (RH) is a major agricultural waste generated as a byproduct during the rice milling stage (Rosa *et al.*, 2012) while cellulose constitutes about 33% of rice husk (Johar, Ahmad e Dufresne, 2012).

Cellulose is the most abundant natural polymer (Zhang *et al.*, 2016). It is a low-cost versatile material, thermodynamically stable, presenting crystalline structure and numerous hydrogen bonds (K. Shukla *et al.*, 2013). Cellulose undergoes functionalization primarily through the hydroxyl groups (Cuadro, de *et al.*, 2015; K. Shukla *et al.*, 2013).

Previous studies have been conducted on nanofibrillated cellulose modification using aminosilanes for CO<sub>2</sub> capture from the air (Gebald *et al.*, 2011, 2013). Commercially fibrillated cellulose suspension was used in that work. Hereby, we investigated chemical modification of cellulose fibers, extracted from the rice husk, with different amines. The potential of the newly developed compounds for CO<sub>2</sub> capture was characterized. Quantum mechanical simulations were employed to elucidate the impact of different functional groups on CO<sub>2</sub> sorption.

## **2. Experimental Section**

### **2.1 Materials**

The rice husks were donated by Cooperativa Arrozeira Extremo Sul Ltda. The material was washed with distilled water and dried in an oven at 100°C for 8h. The dried husk was grounded in a knife mill. The fraction that passed through the 20 mesh sieve (0.841 mm) was collected. Anhydrous citric acid (CA, 99.5% Synth), sodium hydroxide (NaOH, 97% Vetec), hydrogen peroxide

(H<sub>2</sub>O<sub>2</sub>, 35%, Neon), N,N-dimethylformamide (DMF, 99.5%, Merck) ethanol (CH<sub>3</sub>CH<sub>2</sub>OH, 99%, Vetec), hydrazine (32.04 g/mol, HYD, aqueous solution 65%), ethylenediamine (60.10 g/mol, EDA, 99%, Vetec) JEFFAMINE® M-2005 (Mn = 2000 g/mol, Huntsman), JEFFAMINE® D-4000 (Mn = 4000 g/mol, Huntsman), JEFFAMINE® D-400 (Mn = 430 g/mol Huntsman), JEFFAMINE® EDR-148 (Mn = 148 g/mol, Huntsman) were used as purchased.

## 2.2 Cellulose extraction

The cellulose extraction was adapted from literature procedures (Johar, Ahmad e Dufresne, 2012; Teodoro *et al.*, 2011). The rice husk was treated with 4 wt% NaOH at 90°C for 2h to remove lignin and hemicellulose of the rice husk fibers. The acid hydrolysis treatment was conducted on the fibers after alkali treatment using 10.0 mol L<sup>-1</sup> H<sub>2</sub>SO<sub>4</sub> at 50°C for 40 min under continuous stirring. Afterwards, bleaching was carried out by addition of 16% (v/v) H<sub>2</sub>O<sub>2</sub> and 5 wt% NaOH solution in a 1:1 proportion (v/v) at 55°C for 2h to remove the remaining lignin. The rice husk:solution ratio was 0.05 g/mL. Each step was repeated for five times. The solid was filtered and washed with distilled water until neutral pH was achieved and dried at 50°C in the oven after each treatment.

## 2.3 Cellulose chemical modification

Cellulose chemical modification was performed in two stages. The first was modification with citric acid (Figure 1). The second step was functionalization with amines (Figure 2). The modification with citric acid was performed by a similar procedure described elsewhere (Rodrigues *et al.*, 2006; Zhu, Fan e Zhang, 2008). The cellulose was poured into a flask containing 1.2 mol/L aqueous solution of citric acid to obtain 100g/L cellulose solution. This solution was stirred for 0.5h at room temperature. The temperature was increased up to 120°C and kept constant 1.5h under stirring. After cooling, the cellulose was washed with distilled water in the proportion of 200 mL/g for several times to remove any excess of citric acid and dried at 55°C for 24h. The



concentration of carboxylic groups ( $\text{mol mg}^{-1}$ ) introduced into the modified cellulose was determined by back titration (Gurgel *et al.*, 2008; Karnitz *et al.*, 2007; Shang *et al.*, 2016), by treating the material (0.1 g) with 100 mL of an aqueous solution of NaOH (0.02 mol/L) for 1h under constant stirring. The materials were separated by filtration and the solution was titrated with HCl (0.02 mol/L).

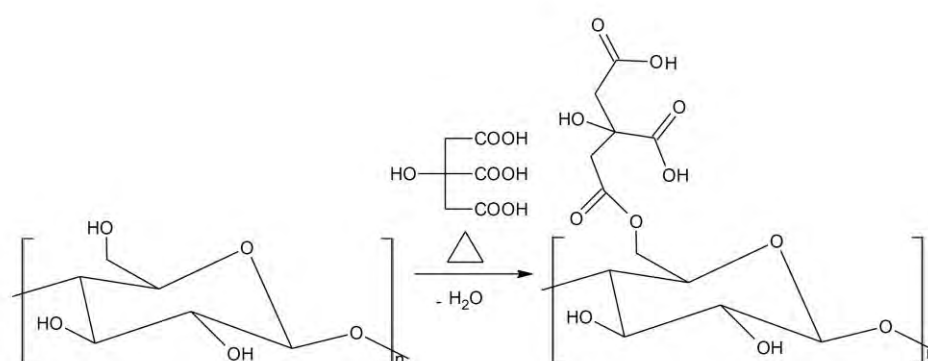


Figure 1. Schematic representation of the carboxylic acid functions introduced on the cellulose surface.

The amine functionalization reaction is described elsewhere (Bernard *et al.*, 2016; Magalhaes *et al.*, 2014). In a Schlenk flask, under  $\text{N}_2$  atmosphere 1g of citric acid modified cellulose was added to the amine (molar ratio  $\text{COOH}/\text{AMINE} = 1:1$ ) dissolved in 15 mL of dimethylformamide (DMF). The mixture was stirred at  $40^\circ\text{C}$  for 2h. The materials were separated by filtration, washed repeatedly with distilled water and ethanol and dried at  $55^\circ\text{C}$  for 24h. The concentration of amine ( $\text{mol mg}^{-1}$ ) introduced into the modified cellulose was determined by back titration, by treating the material (0.1 g) with 100 mL of an aqueous HCl solution (0.02 mol/L) for 1 hour under constant stirring. The materials were separated by filtration and the solution was titrated with NaOH (0.02 mol/L) (Gurgel *et al.*, 2008; Karnitz *et al.*, 2007) (see Figure 2).

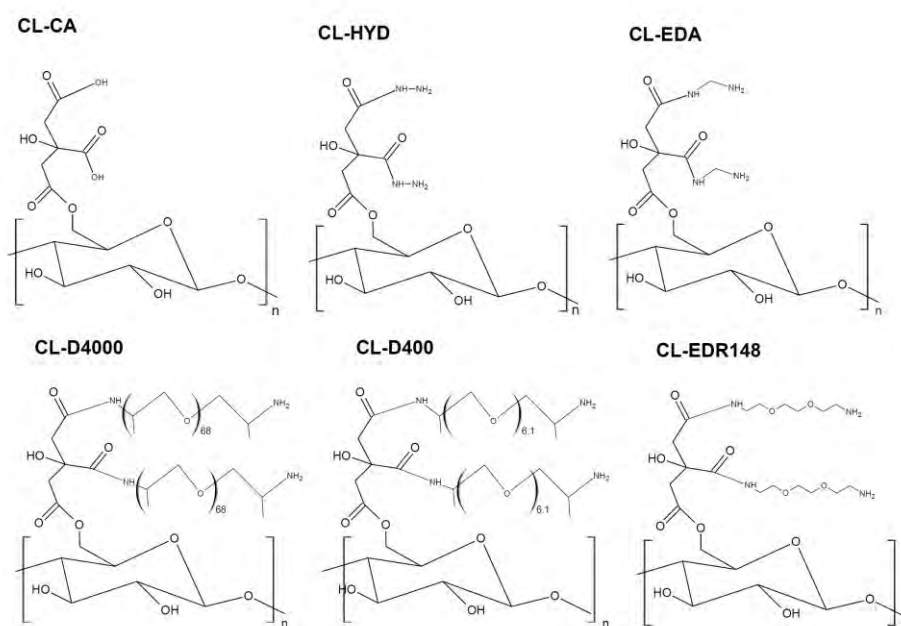


Figure 2 – Schematic representation of amines-modified cellulose.

## 2.4 Characterization

The synthesized materials were characterized by Universal Attenuated Total Reflectance sensor (UATR-FTIR) using Perkin-Elmer Model 100 FTIR Spectrum, in the range  $4000\text{--}650\text{ cm}^{-1}$ . Field emission scanning electron microscopy (FESEM) analyses were performed in FEI Inspect F50 equipment in secondary electrons (SE) mode. Thermogravimetric Analysis (TGA) was performed using TA Instruments SDT-Q600 between  $25^{\circ}\text{C}$  and  $600^{\circ}\text{C}$  with a heating rate of  $10^{\circ}\text{C}/\text{min}$  in nitrogen atmosphere. Specific surface area and pore size were calculated from nitrogen sorption data using Brunauer-Emmett-Teller (BET) at  $-196^{\circ}\text{C}$  (Micromeritics, ASAP 2420). Powder X-ray diffraction (XRD) pattern was recorded on Bruker-AXS D8 ADVANCE diffractometer operated at 40 kV and 20 mA using Cu K $\alpha$  radiation ( $\lambda = 1.5406\text{ \AA}$ ) in the range  $3\text{--}40^{\circ}$  with a step of  $0.02^{\circ}$  and scanning time of 1.0 min. The Segal method (Eq. 1) was used

to calculate the sample crystallinity index (CrI) of the samples (Segal *et al.*, 1959).

$$\text{CrI} = \frac{I_{002} - I_{AM}}{I_{002}} \times 100 \quad (1)$$

where  $I_{002}$  is the maximum intensity of the 002 peak and  $I_{AM}$  is the intensity scattered by the amorphous part of the sample. The diffraction peak for plane (002) is located at diffraction angle around  $2\theta = 22^\circ$  and the intensity scattered by the amorphous part is measured as the lowest intensity at a diffraction angle around  $2\theta = 18^\circ$  (Johar, Ahmad e Dufresne, 2012).

## 2.5 Ab Initio Calculations

Thermodynamics of the reactions was estimated using molecular partition functions obtained from the numerical ab initio calculations and subsequent vibration frequency analysis. The Gaussian-X composite schemes proposed by Curtiss and coworkers (Curtiss *et al.*, 1998; Curtiss, Redfern e Raghavachari, 2007) were used. For an interested reader, we compared G-3 with its simplified modification G-3-MP2, as well as a newer method, G-4-MP2. The term 'MP2' in the method designation means that the MP4 stage in the original composite method (G-3, G-4) was substituted by the MP2 stage (Curtiss *et al.*, 1999). The reported results correspond to a

n ideal gas approximation. Gibbs free energy, enthalpy and entropy of each process, computed at given external conditions, are based on the Hess law of constant heat summation and known relations between these thermodynamic quantities.

Binding energies were computed using the Becke-3-LYP hybrid density functional (Becke, 1988; Lee, Yang e Parr, 1988) with the 6-311+G\* atom-centered split-valence triple-zeta basis set. Both polarization and diffuse functions are included in this basis set for higher accuracy of the theoretical results. The basis set superposition error was excluded through the counterpoise approach. The geometries of the molecules were appropriately optimized prior to computing binding energies at the requested level of theory.

## 2.6 Gas adsorption measurements

Pure gas sorption ( $\text{CO}_2$ ) in the samples was gravimetrically assessed in a Magnetic Suspension Balance (MSB), (Rubotherm Präzisionsmesstechnik GmbH, 35MPa and 400°C) equipped with a single sinker device for absorbate density determination and thermostated with an oil bath (Julabo F25/± 0.1°C). The apparatus details are described elsewhere (Blasig *et al.*, 2007; Dreisbach e Lösch, 2000). When compared to other gravimetric sorption methods, the MSB device has the advantage of allowing high-pressure sorption measurements since the sample can be potted into a closed chamber coupled to an external precise accurate balance (accuracy of ±10 µg). The samples (0.06 to 0.09 g) were weighted and transferred to the MSB sample container, and the system was subjected to  $10^{-7}$  MPa vacuum at the temperature of the sorption measurement, 25°C, for 24 hours (constant weight was achieved in this time). The gas ( $\text{CO}_2$ , Air Liquide / 99.998 %) was admitted into the MSB pressure chamber up to the desired pressure, 0.1-3 MPa. In this study, pressure gauge with an accuracy of  $10^{-3}$  MPa was used to control the system pressure. The gas solubility in the samples for each isotherm and pressure considered was measured during 3-4 hours until no more weight increasing for gas sorption was observed. At this step of gas solubility in the samples, the weight reading from the microbalance at pressure P and temperature T is recorded as  $[W]_t(P, T)$ . The mass of the adsorbed gas in the sample (W) was calculated using the following equation (2):

$$W = [W_t(P, T) - W_{sc}(P, T) + \rho_g(P, T) \cdot (V_{sc}(T) + V_s(T))] - W_s(\text{vac}, T) \quad (2)$$

where  $W_{sc}(P, T)$  is the weight of sample container,  $\rho_g(P, T)$  stands for gas density, directly measured with the MSB coupled single-sinker device, dismissing the application of any equation of state to calculate the gas sorption,  $V_{sc}(T)$  is the volume of the sample container, determined from a buoyancy experiment when no sample is charged into the sample container,  $V_s(T)$  the specific solid sample volume,  $W_s(\text{vac}, T)$  is the weight of samples under vacuum and the term  $\rho_g(P, T) \cdot (V_{sc}(T) + V_s(T))$  represents the buoyancy force. In this work, the solubility in the sample was treated as the excess solubility.

### 3. Results and discussion

### 3.1 Specific surface area, amine load and cristallinity

Table 1 shows surface area data, amine loading and crystallinity for each sample. It can be seen that the amine functionalization caused a small increase in surface area with the exception of CL-M-2005 and CL-D-4000 samples. The determined concentration of carboxylic functions of CL-CA sample was  $2.15 \times 10^{-6}$  (mol/mg). Our result is close to the previous report [38]. The concentration of the loaded amines ranged from  $2.00 \times 10^{-6}$  to  $2.04 \times 10^{-6}$  (mol/mg) indicating that almost all supplied carboxyl groups were reacted. The corresponding crystallinity values are discussed in section 3.2.

Table 1: Specific surface area, amine loading and crystallinity.

Sample	SBET (m <sup>2</sup> /g)	Amine loading (mol/mg)	Crystallinity (%)
CELLULOSE	0.2398	-	87
CL-CA	0.4575	-	84
CL-HYD	0.6579	$2.00 \times 10^{-6}$	91
CL-EDA	0.5753	$2.04 \times 10^{-6}$	88
CL-M2005	ND	$2.02 \times 10^{-6}$	-
CL-D400	0.3113	$2.03 \times 10^{-6}$	63
CL-D4000	ND	$2.01 \times 10^{-6}$	-
CL- EDR 148	0.5831	$2.03 \times 10^{-6}$	86

\*ND (Not Determined).

### 3.2 X-ray diffraction (XRD)

Figure 3 shows the X-ray diffraction profile of the extracted cellulose and modified cellulose samples. All samples exhibit typical cellulose peaks around  $2\theta = 16^\circ$ ,  $22.6^\circ$  and  $35^\circ$  (Johar, Ahmad e Dufresne, 2012; Zhang *et al.*, 2016). The patterns indicated that the obtained cellulose presented a typical form of cellulose I since there was no doublet in the main peak at  $2\theta = 22.6^\circ$  (Li *et al.*, 2009). One can also observe a change in the profile of the x-ray diffraction patterns after the chemical modification. The samples that were modified with amines containing higher molecular weight side chain presented less defined peaks indicating a lower organization once compared with cellulose and the samples obtained with amines containing lower molecular weight side chains. The crystallinity index values (Table 1) highlight that cellulose modified with low-

molecular-weight amines (CL-HYD; CL-EDA, CL-EDR-148) present higher crystallinity values. The Compound CL-D-400 functionalized with JEFFAMINE® D-400 ( $M_n = 430$  g/mol) showed considerable decrease in crystallinity.. The samples synthesized with higher-molecular-weight amines (CL-M-2005 and CL-D-4000) exhibited skewed peaks, which indicate that new assembly is formed via inserted side chain. These results indicate that the bigger the size of the inserted side chain the smaller the crystallinity of the sample, probably due to the reduction in the packing degree of cellulose molecules caused by side chains.

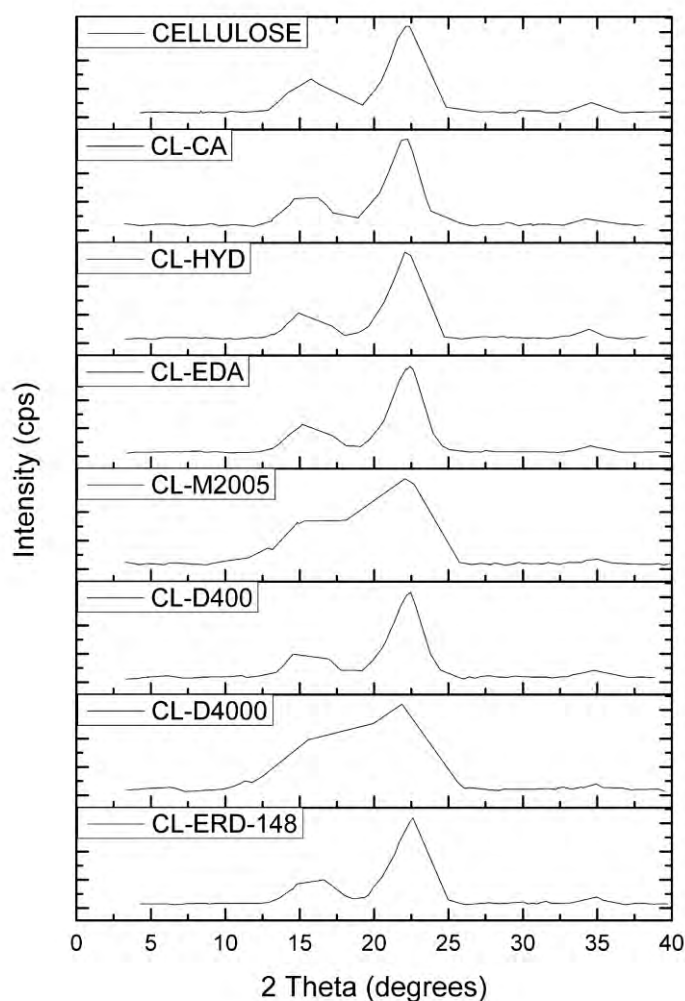


Figure 3. XRD pattern of cellulose and modified celluloses samples.

### 3.3 Spectroscopic characterization

The infrared spectra of cellulose in all samples (Figure 4-a, b and c) showed a broadband located at  $3330 - 3340 \text{ cm}^{-1}$ , attributed to the stretching of the OH groups, band near  $2920 \text{ cm}^{-1}$  related to C-H stretching, band at  $1160 \text{ cm}^{-1}$  corresponding to C-O-C asymmetrical bridge stretching, broadbands around  $1050 \text{ cm}^{-1}$  and  $890 \text{ cm}^{-1}$  related to the C-H and C-O stretching vibration of the cellulose structure (Johar, Ahmad e Dufresne, 2012; Rosa *et al.*, 2012). Having compared spectra for cellulose (Figure 4a) and citric acid modified cellulose (Figure 4b), we identified a strong band at  $1740 \text{ cm}^{-1}$  attributed to the carboxylic acid (X-COOH). This evidences introduction of citric acid into the cellulose chain (Cuadro, de *et al.*, 2015). Figure 4c presents the FTIR for CL-400 sample. A reduction in intensity of the carboxyl group absorption band ( $1740 \text{ cm}^{-1}$ ) due to the chemical modification with amines was observed. A new band at  $1454 \text{ cm}^{-1}$ , characteristic of the symmetric bending of primary amines ( $\text{NH}_2$ ) and at  $2869 \text{ cm}^{-1}$  relative to the  $\text{CH}_2$  stretching modes of the amine chains (Yu, Le e Cheng, 2012). A shoulder at  $3286 \text{ cm}^{-1}$  is evidenced which may be attributed to amine N-H stretching vibrations (Yu, Le e Cheng, 2012).

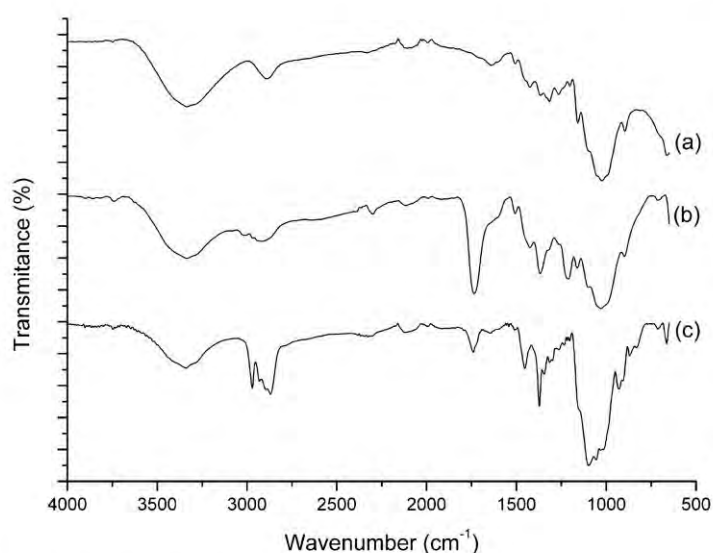
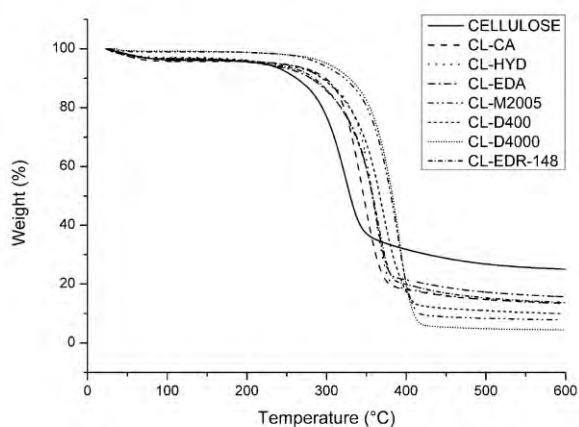


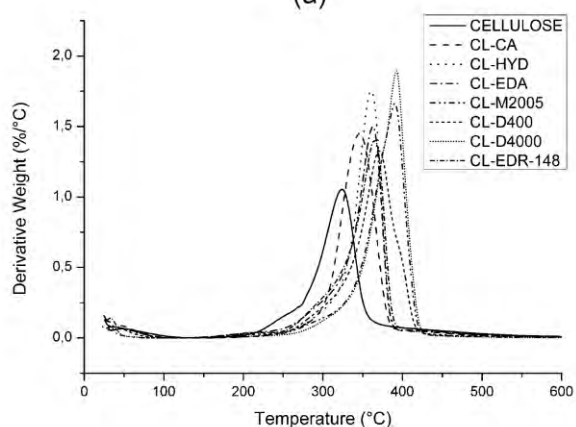
Figure 4. FTIR spectra of: (a) cellulose, (b) citric acid modified cellulose and (c) CL-D400.

### 3.4 Thermogravimetric Analysis (TGA)

Figure 5 presents TG and DTG curves obtained for all materials. All samples showed two thermal events. The earlier weight loss was ca. 4% and  $t_{\text{onset}}$  below 100°C, attributed to water vaporization [17]. The later weight loss attributed to cellulose degradation ( $t_{\text{onset}}$  of 289°C and  $T_{\text{max}}$  of 324°C) being 71%. The modified samples presented a higher thermal stability than cellulose. Degradation temperatures varied from  $t_{\text{onset}}$  318°C to 360°C and  $T_{\text{max}}$  of 346°C to 392°C with a weight loss around 83%.



(a)



(b)

Figure.5 TG (a) and DTG (b) curves for cellulose and modified celluloses samples



### 3.5 Field emission scanning electron microscopy (FESEM)

Figure 6 exemplifies different effects on rice husk surface after cellulose extraction as well as chemical modification. The cellulose extraction in the form of bundles of fiber (Figure 6b) can be highlighted when comparing with the smooth compact surface of the untreated rice husk (Figure 6a). Citric acid modified cellulose (Figure 6c) showed similar morphology to that shown by the sample after bleaching treatment (Figure 6b). Amine modification of cellulose samples led to a separation of fiber bundles, especially sample CL-D-400 (Figure 6g). This may contribute to the increase of contact surface area and, consecutively, to CO<sub>2</sub> sorption.

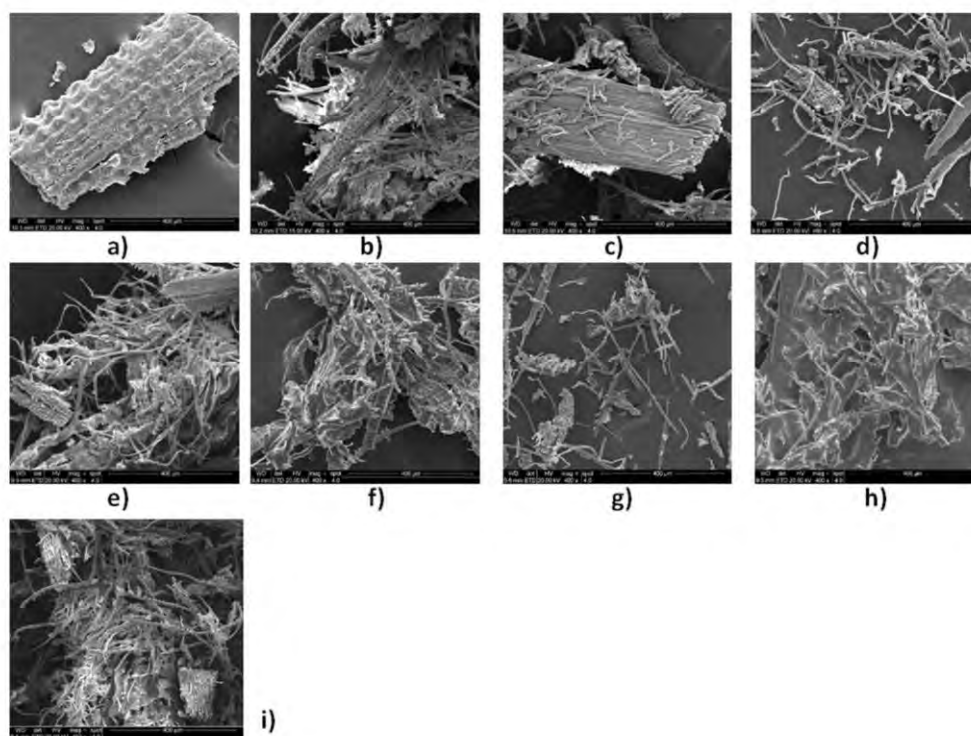


Figure 6. Micrographs of the materials: (a) rice husk fibers, (b) CELLULOSE, (c) CL-CA, (d) CL-HYD, (e) CL-EDA, (f) CL-M2005, (g) CL-D-400, (h) CL-D-4000, (i) CL-EDR-148.

### 3.6 Gas sorption measurements

Table 2 presents the experimental results for CO<sub>2</sub> sorption at 1 bar and 10 bar for the synthesized materials. Note that at both pressures all amine-functionalized samples presented higher sorption values when comparing with the citric acid modified cellulose (CL-CA).

The experimental results corroborate the thermodynamic potentials obtained by simulation (see section 3.7), which indicated that for CL-CA sample occurs only via physisorption, CO<sub>2</sub> is adsorbed due to electrostatic interactions. However, for samples functionalized with NH and NH<sub>2</sub> groups, chemisorption occurs. It is well-known that the interaction between the basic surface and acidic CO<sub>2</sub> molecules results in the formation of surface ammonium carbamates under anhydrous conditions (Yu, Le e Cheng, 2012). Among the amines modified cellulose samples, the best CO<sub>2</sub> sorption result was obtained for the sample functionalized with CL-D-400, whereas the worst result was recorded in the case of CL-D-4000. The micrograph images (Figure 6g) showed a better separation of fiber bundles for the sample CL-D-400. This can improve the interaction of CO<sub>2</sub> with the cellulose chain groups. For the CL-D-4000 sample, (Figure 6h) a higher compaction of the fibers is observed, which can hinder the interaction with CO<sub>2</sub>. More detailed discussion regarding the interaction of the compounds with CO<sub>2</sub> is available in section 3.7.

Table 2. Sorption values for cellulose and amine modified cellulose samples at 25°C.

Sample	10 bar μmolCO <sub>2</sub> /g	1 bar μmolCO <sub>2</sub> /g
CL-CA	243	0.227
CL-HYD	864	287
CL-EDA	1045	346
CL-EDR-148	961	330
CL-D-400	1091	409
CL-D-4000	777	264
CL-M-2005	961	334

The CO<sub>2</sub> sorption capacity of the amine modified cellulose described in this study was lower when compared with commercial solid adsorbents (1800-2000 μmol/g) of surface areas of 1000 to 1700 m<sup>2</sup>/g (Gray *et al.*, 2004). However, the cellulose used in this study originates from an agricultural residue and presents a low amount of amine ( $2 \times 10^{-6}$  mol/mg) when compared to literature (concentrations of 10-20% (w/w)) (Bachelor e Toochinda, 2012). For example, the sorption results at 1 bar and 30°C found were 351 μmol CO<sub>2</sub>/g (for bagasse enriched with 20% of DEA) and 329 μmol CO<sub>2</sub>/g (for rice chaff). These results are similar to the obtained for CL-EDA (346 μmol CO<sub>2</sub>/g) and lower than the sorption values obtained for the sample CL-D-400 (409 μmol CO<sub>2</sub>/g) using lower amine concentrations of around  $2 \times 10^{-6}$  (mol/mg).

In order to examine the reusability of the best material, CL-D-400 was subjected to five additional runs. The sorption was carried out at 25°C under 1 bar of CO<sub>2</sub> pressure. The CO<sub>2</sub> desorption was carried out at 100 °C. At the end, a loss of 5% in the sorption capacity was observed.

### 3.7 Quantum mechanical simulations

Our experimental results (Table 2 with gas-capture results) indicate that all employed compounds, except CL-CA, exhibit a similar performance. Pressure increase leads to better CO<sub>2</sub> capture, while the order of the compounds efficacies persists. A poor performance of CL-CA is due to absence of chemisorption in this case and weak attraction of the CO<sub>2</sub> molecule to the protonated acetate group. Indeed, the computed binding energy for this case amounts to just 7 kJ mol<sup>-1</sup>. The other results can be rationalized via computation of thermodynamic potentials for chemisorption of the molecular fragments (Figure 7), which represent the major working sites, -NH- and -NH<sub>2</sub>, of the employed compounds.

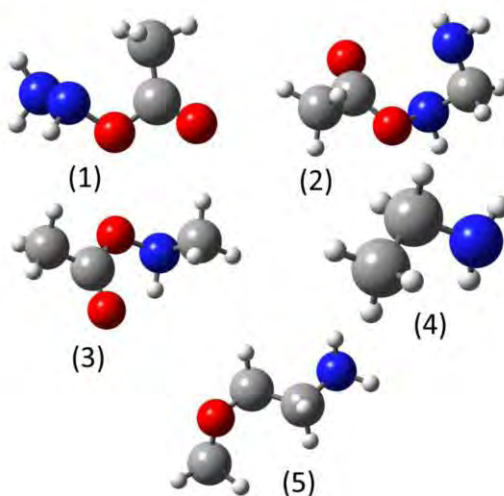


Figure 7. Optimized structures of molecular fragments, which are responsible for CO<sub>2</sub> capture in the modified cellulose species.

Thermodynamic potentials (Figure 8) were used to rationalize the experimentally observed performances of the cellulose based compounds. The depicted potentials correspond to the chemisorption reactions, but omitting subsequent solvation and ionization in aqueous media. This is done for simplicity of discussion and straightforward comparison. For instance, a total free energy of ethylamine is 20 kJ mol<sup>-1</sup>, due to hydrogen bonds between the products of chemisorption and media. Johnson and coworkers (Xie *et al.*, 2010) thoroughly investigated the mechanism of CO<sub>2</sub> chemisorption by the primary amine and presented extensive evidence that the two-step pathway via a zwitterion intermediate is most energetically favorable and, thus, most expected at the working conditions.

The trend of enthalpy completely repeats the trend of free energy (Figure 8). The entropic contribution is similar in all cases. Therefore, the chemical environment (in particular, non-covalent binding energy) is responsible for the difference between the considered molecular fragments. The best performance was found in CL-D-400 and CL-EDA. This is confirmed theoretically by the thermodynamics of fragment 2, whose free energy and enthalpy are lowest. The chemisorption is conducted both through the amide and amine groups and their high concentrations in the compounds. The performance of CL- D-4000 is

worse, since that compound contains lots of ether groups, which are not efficient for the CO<sub>2</sub> fixation. The ether group itself attracts CO<sub>2</sub> weakly. The binding energy corresponding to this process is 8 kJ mol<sup>-1</sup>. The impact of a few available carboxylate groups is even smaller, 4 kJ mol<sup>-1</sup> in terms of binding energy. The performance of CL-HYD is inferior to that of CL-EDA, because the amide group cannot bind CO<sub>2</sub> in the -NH-NH<sub>2</sub> configuration. The resulting compound is unstable (according to the stationary-point-search simulations) and, therefore, is never formed. In turn, separating -NH- and -NH<sub>2</sub> by one or two methylene groups allows to achieve a more decent CO<sub>2</sub> capture.

It is noteworthy that the gas-phase thermodynamics provides a correct account of the reactions occurring in the condensed phase allowing to rate large molecules with multiple binding sites. This fact also suggests that intra-molecular interactions are more important than inter-molecular ones upon chemical adsorption. All methods, G-3, G-3-MP2, G-4-MP2, for thermodynamic potentials provide very similar results.

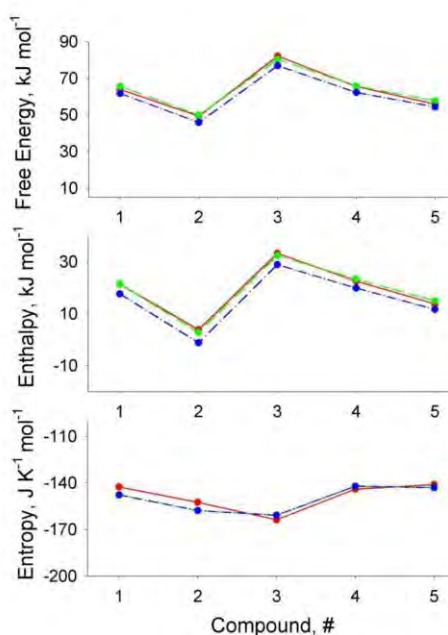


Figure 8. Free energy (top), enthalpy (center) and entropy (bottom) for CO<sub>2</sub> capture through chemisorption at -NH<sub>2</sub> by the molecular fragments depicted in Figure 7. The results from the G-4-MP2 method are red solid lines, from the G-

3-MP2 method are green dashed lines, from the G-3 method are blue dash-dotted lines.

Since a total number of molecules decreases upon chemisorption, pressure increase shifts this reaction rightwards. Figure 9 depicts an effect of pressure up to 1000 MPa. The latter brings the  $34 \text{ kJ mol}^{-1}$  free energy increase. An approximate shift of the chemisorption equilibrium constant can be estimated from the following relationship:  $K^i = \exp(-G^i/R \times T)$ , where  $T$  is temperature,  $R$  is gas constant,  $G^i$  is free energy at given conditions. These calculations allow to determine to which extent it is economically feasible to invest in the high-pressure apparatus. The calculated effect of pressure is in concordance with our experimental results. In all systems, higher pressure results in a significantly better  $\text{CO}_2$  fixation.

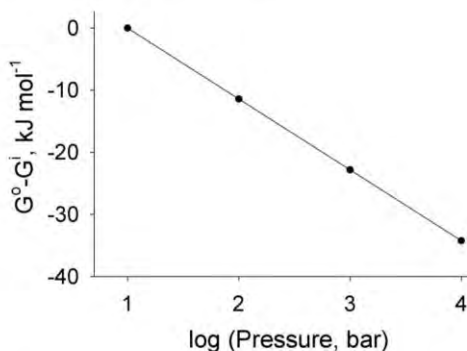


Figure 9. Free energy decrease due to elevation of external pressure. The thermodynamics of the  $\text{CO}_2$  capture reaction by ethyl amine was considered for an example.

#### 4. Conclusion

The chemical modification of cellulose fibers, extracted from the rice husk with different amines was performed. The potential of the obtained compounds for use as sorbents for  $\text{CO}_2$  capture evaluated. The quantum mechanical simulations and the experimental results revealed the role of each moiety of the new compounds in the gas capture process. A poor performance of CL-CA was noticed due to the absence of chemisorption in this case and a relatively weak

attraction of the CO<sub>2</sub> molecule to the protonated acetate group. The best performance was found in CL-D-400 (CL-D - 400 (409 μmol CO<sub>2</sub>/g at 1 bar). This result is confirmed theoretically by the thermodynamics of fragment 2, whose free energy and enthalpy are lowest. The chemisorption is conducted both through the amide and amine groups, therefore, their high concentrations in the compounds are important. On the other side, when a higher-molecular-weight amine was used (CL-D-4000) the performance was worse, since that compound contained lots of ether groups, which are not efficient for the CO<sub>2</sub> fixation. The performance of CL-HYD is inferior to that of CL-EDA, because the amide group cannot chemically bind CO<sub>2</sub> in the -NH-NH<sub>2</sub> configuration. In turn, separating -NH- and -NH<sub>2</sub> by one or two methylene groups allows to achieve a more decent CO<sub>2</sub> capture. Higher pressure results in a significantly better CO<sub>2</sub> fixation. The use of an industrial residue, cellulose, as a material for CO<sub>2</sub> capture is a promising environmental solution, besides being a low-cost option.

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## 5. DISCUSSÕES GERAIS

### 5.1. Comparativo entre os melhores materiais de cada grupo

A partir dos resultados e discussões apresentados na seção 4 foram selecionados os melhores materiais obtidos de cada grupo: CL-TBA, [CeIEt<sub>3</sub>N][PF<sub>6</sub>], CL-D400; PU-TBA, PU-TBP e HPIL-02-TBA. A capacidade de sorção de CO<sub>2</sub> foi utilizada como fator principal para seleção destes materiais. A aparência física dos materiais é ilustrada na Figura 5.1 e um comparativo entre os principais resultados é mostrado na Tabela 5.1. Os resultados experimentais obtidos para o teste de dessorção utilizando temperatura ou vácuo, também foram incluídos na Tabela 5.1. Neste teste, após a realização do ensaio de sorção na célula de decaimento de pressão, a amostra foi submetida a vácuo ou aquecimento. Em seguida, esta amostra foi re-submetida a um ensaio de sorção e a partir da diferença entre o CO<sub>2</sub> capturado no primeiro e no segundo ensaio, a quantidade de CO<sub>2</sub> dessorvido da amostra foi determinado.

Filmes de Poli (líquidos iônicos) translúcidos foram obtidos a partir de PU ou PU-imida (Fig. 5.1d-f). Os PLLs base PU-imida apresentaram coloração levemente amarela, resultante do dianidrido utilizada na síntese (Fig. 5.1f). Todos os materiais sintetizados a partir da celulose apresentaram-se sob a forma de pó, as amostras de celulose modificadas com o ácido cítrico (Fig. 5.1a e Fig. 5.1c) exibiram coloração amarela e as modificadas com cloreto de tionila, coloração marrom escura (Fig.5.1b).



Figura 5.1. Imagens ópticas obtidas para os melhores materiais de cada grupo: a) CL-TBA, b)  $[\text{CeEt}_3\text{N}][\text{PF}_6]$ ; c) CL-D400; d) PU-TBA; e) PU-TBP e f) HPIL-02-TBA.

A Tabela 5.1 apresenta um comparativo entre os resultados de estabilidade térmica, módulo de Young, sorção e dessorção de  $\text{CO}_2$  obtidos para as melhores amostras de cada grupo. Todas as amostras demonstraram boa estabilidade térmica. O PLI base PU-imida mostrou módulo de Young superior aos obtidos com os PLIs base PU. Os valores de sorção de  $\text{CO}_2$ , obtidos a baixa pressão ( $\sim 0.1\text{MPa}$ ) demonstraram o desempenho superior dos PLIs celulósicos (CL-TBA e  $[\text{CeEt}_3\text{N}][\text{PF}_6]$ ). Por outro lado, com o aumento da pressão (1MPa), os melhores resultados de sorção foram alcançados com o PLI celulósico  $[\text{CeEt}_3\text{N}][\text{PF}_6]$  e com o PLI base PU-imida HPIL-02-TBA.

Os testes de dessorção realizados com vácuo demonstraram a capacidade de reciclo dos PLIs. Este comportamento não foi evidenciado para a amostra CL-D400, onde a reação entre o grupo  $-\text{NH}_2$  e o  $\text{CO}_2$  pode produzir carbamato de amônio (ABOUDI; VAFAEEZADEH, 2015), reduzindo assim a capacidade de sorção deste material ao longo do processo. Para todos os PLIs a substituição de vácuo por

temperatura durante os testes de dessorção gerou uma retenção de CO<sub>2</sub> na amostra (10,5% a 17%). A diminuição da temperatura de dessorção de CO<sub>2</sub> da amostra CL-D400 de 100°C para 50°C reduziu a capacidade de sorção de CO<sub>2</sub> em 58,5% (Tabela 5.1). Em geral, os PLIs demonstraram maior potencial para captura quando comparados com o CL-D400.

Tabela 5.1. Principais resultados obtidos para as amostras selecionadas em cada grupo.

Amostra	T <sub>(onset)</sub> °C	Módulo de Young	Sorção de CO <sub>2</sub> (mg/g) ~0,1MPa	Sorção de CO <sub>2</sub> (mg/g) 1 MPa	CO <sub>2</sub> dessorvido (%)		Material pertencente ao Capítulo
					Condições: vácuo; T. amb; Tempo 1h	Condições: 50°C e Tempo 1h	
CL-TBA	324	-	44,0*	53,0*	100	84,5	III
[CeIEt <sub>3</sub> N][PF <sub>6</sub> ]	203	-	38,0*	80,0*	100	89,5	IV
PU-TBA	190	6,3	16,1**	49,1**	100	84,6	I
PU-TBP	210	11,4	15,7**	43,7**	100	86,5	I
HPIL-02-TBA	175	56,4	33,1**	61,0**	100	83,0	II
CL-D400	318	-	18,0*	48*	95***	41,5	V

\*Temperatura = 25°C; \*\*Temperatura = 30°C; \*\*\* condição de dessorção diferente = 100°C sem uso de vácuo.

Apesar dos PLIs sintetizados neste trabalho terem demonstrado valores de sorção de CO<sub>2</sub> superiores a diversos PLIs relatados na literatura (TANG et al., 2005a; ZHU et al., 2011; PRIVALOVA et al., 2013; MOROZOVA et al., 2017), a capacidade de sorção destes PLIs não superou a alcançada com o uso da solução aquosa de 30% m/m de MEA (0,45 mol/mol = 324mg CO<sub>2</sub>/g MEA á 1 atm e 25°C) (RINPRASERTMEECHAI et al., 2012), solvente tradicionalmente empregado em plantas de captura. Entretanto, o uso destes PLIs oferece algumas vantagens em comparação às aminas, tais como a ausência da formação de produtos corrosivos, a reciclabilidade, e a baixa necessária para a regeneração. Também cabe ressaltar que a comparação destes PLIs com aminas deve considerar outros fatores que não

foram abordados neste trabalho, como por exemplo o custo de fabricação destes sorventes, a capacidade seletiva, entre outros.



## 6. CONCLUSÕES

Nesta secção, as conclusões serão divididas em três blocos: PLIs aniônicos base PU e PU-imida, PLIs celulósicos aniônicos e catiônicos e aminas suportadas em celulose.

Os resultados obtidos para os PLIs aniônicos base PU e PU-imida evidenciaram que o contra-cátion desempenha um papel importante sobre a solubilidade do CO<sub>2</sub>, estabilidade térmica, propriedades mecânicas e morfologia dos PLIs. Ensaio experimentais de sorção/dessorção de CO<sub>2</sub> demonstraram a capacidade de reutilização destes materiais. Estudos computacionais revelaram que o grupo carboxílico desprotonado (COO<sup>-</sup>) é um importante sítio de interação com o CO<sub>2</sub>, e que os cátions TBA e TBP exercem uma coordenação mais fraca sobre este sítio comparada com o imidazólio. A coordenação fraca favorece a interação entre o CO<sub>2</sub> e o grupo carboxilato (COO<sup>-</sup>), contribuindo para o aumento da capacidade de sorção de CO<sub>2</sub>. A comparação dos resultados de sorção de CO<sub>2</sub> obtidos para os PLIs base PU e PU-imida mostraram que o aumento da presença de grupos polares na cadeia polimérica promove a melhora da capacidade de sorção de CO<sub>2</sub>. Para os PLIs base PU-imida observou-se que concentrações menores dos grupos silanos favorece a interação entre o PLIs e o CO<sub>2</sub>.

O uso da celulose extraída da casca de arroz como base para o desenvolvimento de sorventes para captura de CO<sub>2</sub>, demonstrou ser uma opção promissora, pois permite a redução de resíduos agroindústrias e pode contribuir para a obtenção de sorventes de baixo custo.

Os PLIs celulósicos aniônicos e catiônicos demonstraram-se estáveis termicamente e reutilizáveis no processo de captura de CO<sub>2</sub>. Para os PLIs aniônicos o método empregado para obtenção destes compostos permite uma capacidade de

sorção elevada, com o uso de uma pequena quantidade de líquido iônico. A investigação realizada por simulação computacional revelou que o CO<sub>2</sub> interage de forma mais efetiva com o grupo carboxilato, e, em menor grau com os grupos polares da estrutura da celulose. Os cátions BMPYRR, TBP e TBA são usados como contra-íons de coordenação fraca, mantendo os locais de atração do CO<sub>2</sub> parcialmente ou totalmente livres. Por sua vez, o cátion BMIM coordena o CO<sub>2</sub>, devido a uma ligação de hidrogênio fraca proveniente do átomo de hidrogênio ácido do anel imidazólio com o átomo de oxigênio do CO<sub>2</sub>. A capacidade de sorção de CO<sub>2</sub> dos PLIs catiônicos é influenciada pelo contra-ânion, os resultados mostraram também que a fluoretação do ânion pode proporcionar o aumento da sorção de CO<sub>2</sub>. Porém, o uso de ânions fluorados volumosos pode prejudicar a interação do CO<sub>2</sub> com os grupos polares da estrutura e reduzir a capacidade de sorção de CO<sub>2</sub>.

A modificação química da celulose com aminas e polieteraminas promoveu resultados de sorção de CO<sub>2</sub> menores que os obtidos com os PLIs celulósicos e PLIs aniônicos base PU. A introdução de grupos amidas e aminas proporcionaram o aumento da estabilidade térmica e da capacidade de solubilidade de CO<sub>2</sub> da celulose. Estudos de simulação revelaram que a quimisorção é favorecida por grupos -NH- e -NH<sub>2</sub>. Aliado a isso, os testes experimentais de sorção de CO<sub>2</sub> evidenciaram que a concentração destes grupos pode influenciar a solubilidade.

Para os PLIs, os ensaios de dessorção efetuados com o uso de temperatura (50°C) demonstraram a possibilidade de utilização de calor residual de processo para a realização da etapa de dessorção. Isto pode representar uma vantagem em relação a soluções aquosas de aminas que necessitam de alta quantidade de calor para etapa de regeneração.

Dentre todos os materiais sintetizados neste trabalho, os PLIs celulósicos CL-TBA e [CeEt<sub>3</sub>N][PF<sub>6</sub>] e com o PLI base PU-imida HPIL-02-TBA demonstraram maior afinidade pelo CO<sub>2</sub>, aliado a boa estabilidade térmica e reciclabilidade.

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## ANEXO A

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Manuscript Type: Article

Manuscript Title: Hybrid alkoxysilane-functionalized urethane-imide-based poly(ionic liquids) as a new platform for carbon dioxide capture

Author(s): Bernard, Franciele L.; Polesso, Barbara; Cobalchini, Fabiana; Chaban, Vitaly; do Nascimento, Jailton; Dalla Vecchia, Felipe; Einloft, Sandra

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Title: Cellulose based poly(ionic liquids): Tuning cation-anion interaction to improve carbon dioxide sorption

Authors: Franciele Bernard; Rafael Duczinsky; Marisol Rojas; Maria Fialho; Luz Angela Carreno; Vitaly Chaban; Felipe Dalla Vecchia; Sandra Einloft

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barbara.polesso@acad.puhrs.br <barbara.polesso@acad.puhrs.br>; vvchaban@gmail.com <vvchaban@gmail.com>; Marcus Seferin  
<seferin@puhrs.br>; Felipe Dalla Vecchia <felipe.vecchia@puhrs.br>; Sandra Mara Oliveira Einloft <Einloft@puhrs.br>;

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